Lawrence Berkeley National Laboratory

Recent Work

Title

Tailoring Copper Nanocrystals towards C2 Products in Electrochemical CO2 Reduction.

Permalink https://escholarship.org/uc/item/9kk4j52t

Journal Angewandte Chemie (International ed. in English), 55(19)

ISSN 1433-7851

Authors

Loiudice, Anna Lobaccaro, Peter Kamali, Esmail A <u>et al.</u>

Publication Date

2016-05-01

DOI

10.1002/anie.201601582

Peer reviewed

Tailoring Cu nanocrystals towards C₂ products in electrochemical CO₂ reduction

Anna Loiudice,^[a] Peter Lobaccaro,^[b,c] Esmail A. Kamali,^[a] Timothy Tao,^[d] Brandon Hung,^[c] Joel W. Ager, ^[b,d] Raffaella Buonsanti*^[a]

ChemSusChem

Communications Abstract: Favoring the CO₂ reduction reaction (CO2RR) over the hydrogen evolution reaction and controlling the selectivity towards multicarbon products are currently major scientific challenges in sustainable energy research. It is known that the morphology of the catalyst can modulate catalytic activity and selectivity, yet this remains a relatively underexplored area in electrochemical CO2 reduction. Here, we exploit the material tunability afforded by colloidal chemistry to establish unambiguous structure/property relations between Cu nanocrystals and their behavior as electrocatalysts for CO2 reduction. Our study reveals a non-monotonic size-dependence of the selectivity in cube-shaped copper nanocrystals. Among 24 nm, 44 nm and 63 nm cubes tested, the cubes with 44 nm edge length exhibited the highest selectivity towards CO2RR (80%) and faradaic efficiency for ethylene (41%). Statistical analysis of the surface atom density suggests the key role played by edge sites in CO2RR.

Converting carbon dioxide into useful chemicals through an electrochemical process represents a promising path towards establishing a carbon-neutral cycle.^[1] High energy density multicarbon products, such as hydrocarbons and alcohols, are much sought after as potential sustainable fuels.^[1] Copper is the only metal electrocatalyst with the ability to form substantial amounts of hydrocarbons at the moment.^[2] Yet, its efficiency and selectivity are still too low to allow practical implementation of this technology.^[1,2] Previous work has shown that the product selectivity depends on the Cu structure.^[3-13] For example. Cu (100) increases the selectivity for ethylene while methane is the main hydrocarbon product on Cu (111).^[9-12] Recently it has been shown that the selectivity for ethylene can be further increased by utilizing Cu cubes, with around 100 nm edge length, whigh predominantly show the (100) facets, and nearly comple suppress methane generation.^[7] While the (100) surface single crystals is the most selective for ethylene, it still produces a significant amount of methane.^[9-12] The increase of the ethylene yield in the cubes was hypothesized to stem from the convolution of a high density of ethylene-selective active sites. yet to be identified, and local pH changes.^[7]

- Dr. A. Loiudice, E. A. Kamali, Prof. R. Buonsanti* [a] Department of Chemical Sciences and Engineering École Polvtechnique Fédérale de Lausanne Sion, CH-1950, Switzerland *E-mail: raffaella.buonsanti@epfl.ch
- P. Lobaccaro, Dr. J.W. Ager [b] Joint Center for Artificial Photosynthe Lawrence Berkeley National Laboratory Berkeley, California 94720, United States

document.

- P. Lobaccaro, B. Huang [C] Department of Chemical and Biomolecular Eng ring University of California, Berkeley Berkeley, California 94720. Ur
- [d] J. W. Ager, T. Thao, Department of Materials Science a Engineering, University of California, Berkeley, California 94720 ited States

Supporting inform via a link at the end of the ticle is a

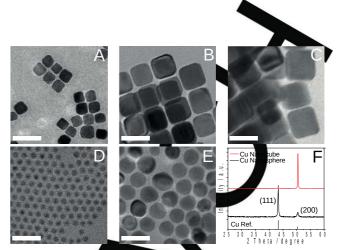


Figure 1. (A-C) TEM imag average edge length of 24, 44, 63 nm respectively; (D-E) TEM Image us spheres with an average diameter of 7.5 and 27 nm, respectively. Scale bar 50 nm. (F) Typical XRD patterns of the Cu spheres (black line) and Cu cubes (red line).

ge in ethylene, the production rate of hydrogen Unlike the cha on the cubes vas not significantly changed from that of per.^[7] Size-dependent selectivity of spherical polycrystalline c Cu catalysts tow d CO2 reduction has been investigated but ^[3,4] Manthiram et al. reported enhanced with on with well dispersed 7 nm Cu eleg mical m nanoparticles compared to Cu foil.^[4] In contrast, Reske et al. und that Cu nanoparticles smaller than 30 nm produced vdrogen as the main product.^[3] Unambiguous conclusions cannot be made as these catalysts were synthesized by different ved.^[3,4] different measurement conditions appro were emp

xplore the effect of both size and shape of Cu Here, we NCs) on their CO2 electroreduction activity and ocrystals sele livity. Two different sizes of Cu NC spheres (7.5 nm and three different sizes of Cu NC cubes (24 nm, 44 and nm, and 63 nm) were synthesized by the same colloidal chemistry-based method. Within the same morphology, smaller whibited higher activity; however overall, the cube-shaped NCS were more intrinsically active than the spheres, as NC ssed below. An unexpected non-monotonic selectivity trend di observed, with an 80% selectivity for carbon products served, 50% of which corresponded to ethylene, for the best case Cu NC cubes with 44 nm side length.

Figure 1A-E shows the transmission electron microscopy (TEM) images of the synthesized Cu NC cubes and spheres which exhibit uniformity of size and shape (Figure S1). Figure 1F compares the typical X-ray diffraction (XRD) patterns for the assynthesized Cu NC cubes and spheres. In the Cu cubes, the (200) peak is more pronounced than that of the bulk fcc Cu reference, which indicates the predominance of {100} facets.

The Cu NC catalysts were spin-coated on glassy carbon, which served as the working-electrode in a three-electrode setup containing CO2-saturated 0.1 M potassium bicarbonate as the electrolyte. Figure 2 reports the current densities, normalized by the electrochemically active surface area (ECSA), corresponding to the different Cu catalysts at -1.1 V vs. RHE while the linear

COMMUNICATION

WILEY-VCH

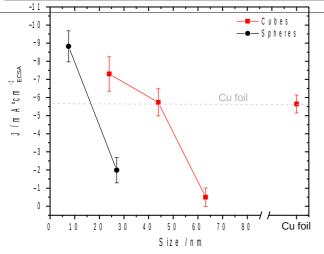


Figure 2. Current density at -1.1 V vs RHE are plotted against the size of the Cu NC catalysts. The complete linear sweep voltammogramms are reported in Figure S3. The error bars indicate the standard deviation of five independent measurements. High purity Cu foil was used as a control-working electrode and showed current density comparable to Kuhl et al.^[2]

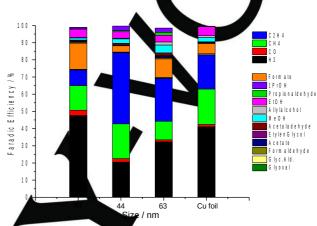
sweep voltammograms are reported in Figure S3. When comparing electrodes with different areas, it is important to know their relative number of active surface sites, as a larger surface area can lead to a larger apparent activity, without fundamentally changing the underlying catalytic ability of the material. Among the different techniques, we chose to normalize the current density by the ECSA measured using the Randles-Sevcik equation (Figures S2). A discussion about different methods to estimate surface area is reported in Figure S3 and Table S1. In figure 2 it can be seen that the smallest spheres (7.5 nm) and the cubes of 24 nm and 44 nm exhibited larger current density than Cu foil. Instead, the 27 nm spheres and the 63 nm cubes were less active (lower current density) than Cu foil. Within the particles of the same shape, there is a monotonic sizedependent trend, where smaller NCs are more active than bigger ones. Such trends are similar to what was observed by Reske et al. for their Cu spheres.^[3] Interestingly, the sizedependence does not hold when Cu cubes and Cu spheres are compared, with the 44 nm cubes showing a higher curren density than the 27 nm spheres, indicating a higher intring activity for the cubes.

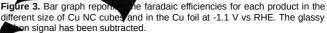
To gain insights into the CO2RR and the competing hydrogen evolution reaction (HER) of the synthesized Cu NCs, faradaic efficiency (FE) measurements were performed at -1.1V vs. RHE, where HER is less favorable (Figure S5)., Herein, the focus is on the size-dependent activity of the cubes it has not been investigated before. Figure 3 reports adaic efficiency for each product detected by gas chromatog HPLC for the various Cu cubes compared to Cu foil. Each represents the average of independer point hree measurements. For comparison, the FE n rements relative to the spheres are reported in Figure S6 a Stable S3. currents were measured during the ne hour ele Figure S7). Importantly, scanning electron croscopy con lat the morphology of the Cu NC cube was preserved ina electrolysis (Figure S8).

When aiming to maximize the per mance of CO2RR catalysts, higher selectivity for CO2RR over R and for higher hydrocarbons are much souch ave decided to focus on the selectivity for C Eco2RR), the 2H4) and the selectivity for faradaic efficiency for ethylene ethylene versus methane (C_2H_4/C_2 as figures of merit to discuss the catalytic rformance of r Cu NC cubes. Table 1 hat the smaller cubes summarizes the r _{CO2RR}=50%), with C1 show equal activity for HER a

products, and specifically methane and formate being the major carbonaceous species ($C_2H_4/CH_4=0.63$, Table S3). As the size of the cubes increases, the three figures of merit improve. The cubes with 44 nm edge length exhibited the best performance with the highest selectivity toward CO2RR (FE_{CO2RR}=80%) and with ethylene as the major reduction product (FE_{C2H4}=41%, $C_2H_4/CH_4=2.03$). The largest cubes showed a slight increase in C_2H_4/CH_4 ; however, the overall CO2RR activity decreased and the ethylene production was lower (FE_{C2H4}=25%).

The surface metal atomic coordination a prresponding surface electronic structure, along with the surface H, have been identified as crucial parameters to explain the 2RR selectivity in nanostructured and NC-based Cu elect ^{16]} In agreement with Reske et al., we exclude a ificant rol of local pH changes to explain the set nges in our catalysts.^[3] In fact, in the Cu NC cubes with edge length, which produce larger current densities would ct the pH increase to lead to lower hydroger ction, which tot the case. [3,7] Instead, the analysis of e active site plane-, edgeand corner-type) on the surface ionalize the aids better to tivity trend. [3,17,18 non-monotonic nature of the se







 $\label{eq:table_to_constraint} \begin{array}{l} \textbf{Table 1.} \ \mbox{Figures of merit to describe the CO2RR catalytic behavior of the Cu} \\ \mbox{NCs cubes compared with Cu foil.} \end{array}$

| Size M | FE _{CO2RR} (%) ^[a] | FE _{C2H4} (%) | $C_2H_4/CH_4^{[b]}$ |
|---------|--|------------------------|---------------------|
| 24 | 43 | 9 | 0.63 |
| 44 | 80 | 41 | 2.03 |
| 63 | 63 | 25 | 2.33 |
| Cu foil | 56 | 20 | 1 |

[a] FE_{CO2RR} = 100*FE_{CO2RR}/(FE_{CO2RR}+FE_{HER}), where FE_{CO2RR} is the sum of the FEs of all the C-products. [b] C₂H₄/CH₄ = FE_{C2H4}/FE_{CH4}.

As discussed in the introduction, the exposed crystalline planes have an influence on the CO2RR selectivity, with the (100) and the (111) surfaces favoring ethylene and methane formation, respectively.^[7,9-12] In the Cu NC cubes, the collected XRD patterns indicate that the {100} facets predominate on the

WILEY-VCH

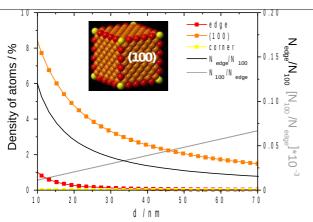


Figure 4. On the left axis: Density of adsorption sites in Cu NC cubes reported versus the edge length (d). On the right axis: trend of N_{edge}/N_{100} and N_{100}/N_{edge} versus the edge length (d), where N_{edge} is the number of atoms at the edge and N_{100} is the number of atoms on the (100) plane.

surface. We used a simple model to derive the statistics of surface atoms for Cu NC cubes (Figure 4 and Table S4). As the NC size increases, the relative number of atoms on the corners and on the edges decreases while the number on plane increases. Thus we move towards a surface configuration closer to one single crystal with all the surface atoms laying on (100) planes. The sizes of the smaller Cu NC cubes (edge length \leq 24 nm) is on the upper end of the size range studied by Reske et al.^[3] Based on their findings, the higher density of lowcoordinated sites (corners, steps and kinks) might account for their high activity (higher current density) and the HER selectivity compared to the other samples. The trend of the active sites with the size suggests that the unique reactivity of the 44nm Cu NC cubes derives from an optimal balance between plane- and edge- sites. Thus, we highlight the importance of atoms on the edges as CO2RR and ethylene-selective active sites in the Cu NC cubes. The role of edge sites has been longer recognized in catalysis.^[18,19] Furthermore, this hypothesis is consistent with the size-dependence studies on Au NCs as CO₂ electroreduction catalysts.^[18] Using density functional theory, edge sites were identified as key sites which facilitate the adsorption and stabilization of CO2RR reaction intermediates (i.e. COOH*) and inhibits HER. The dominance of such sites on 8 nm Au NG made them more active and selective for CO2RR than 4, 6, nm ones.[18]

Catalysts which are selective for CO2RR and favor C-C coupling are much sought after. Through a careful investigation of Cu NCs with different sizes and shapes produced by the same colloidal approach, we confirm the increased lectivity for ethylene on Cu NC cubes, in agreement with pr rks.^[7] In addition, we reveal a unique size-dependent selection the highest selectivity for CO2RR and C2 products found nm Cu cubes. This result emphasizes that an optimal ratio of edge sites over (100) plane-sites (N_{edge}/N₁₀ 25 for d=44 nm) is crucial to maximize CO2RR and ethyler /. Our work represents a step further towards Itimately it which structural parameters in CO2 reduce electrocata **n** the selectivity towards multicarbon p ducts. Future w on ultrathin structures and nanoframes, ch possess a high density of edge atoms, might be desirable further improve the selectivity towards CO2RP and towar C-C coupling. Considering the possibility to access a wide compositional space, ndational in demonstrating the use of atomic defined collocal NCs as a eduction.[20] materials platform for advancing CC

Experimental

NCs synthesis. Cu NCs spheres and cubes were synthesized according to modifications of literature methods.^[21] For the synthesis of Cu cubes of 24 nm, CuBr (3 mmol) and TOPO (25 mmol) were dissolved into oleylamine (10 mL) in a three-necked flask by vigorous magnetic stirring at 80 °C for 15 min. Then the resulting solution was heated up to 210°C guickly and refluxed at this temperature for 1 h before cooling down to room temperature naturally. In order to obtain Cu nanocubes with a larger edge length (43.8 nm), the content of TOPO was reduced to 7.5 mmol and the content of CuBr to 1.5 mmol dissolved in 35 mL of oleylamine. Then the resulting solution was heated up to 220°C quickly and refluxed at this temperature for 1 h. In order to obtain Cu nand with a larger edge length (63 nm), the content of TOPO was redu 2.5 mmol, while the temperature and the aging time was increased to and 2.5 h respectively. For the synthesis of Cu spheres of 27 ng (35 mL) and CuBr (3 mmol) were mixed in a three-necked by st magnetic stirring at 80°C for 15 min, then TOF injected into mmol) the obtained green solution, which immediate olorless. After an additional stirring at 80 °C for 5 min, the mixed s was heated up to 260°C quickly and kept at this temperature for x3hbe ling down to room temperature naturally. Excess was added bright out by itated reddish solution, and the Cu were pre crystals were centrifugation. The volume of hexa in which the suspended was adjusted in order t ave a concentrati of 1.5 mg/mL for each solution. Cu NCs sphere vere obtained by owing similar procedure. Electrochemical CO2RR experimen ectrochemig neasurements were carried out using a Biol or gas product SF analysis, a gas chromatograph instru as used. The liquid

products were collected from the whode and anode chambers after electrolysis and analyzed by high-performance liquid chromatography (HPLC) on an UltiMate 3000 instrume from Thermo Scientific. More detailed experimental performance are apported in the Supporting Information.

Acknowledgement

The authors th k Tracy Mattox, at the Molecular Foundry, for elemental analy of the samples by ICP-AES and Dr. Kendra Kuhl and Dr. Etos Cave for helpful discussions. The synthetic deve terial integration was supported by Laboratory Directe search and Development (LDRD) funding from Berkeley ab, provided by the Director, Office of ence, of the U.S. Department of Energy under Contract No. E-AC02-05CH11231. Electrochemical measuraments and product analysis were supported by the Joint Center for Artificial thesis, Photo DOE Energy Innovation Hub, supported of Science of the U.S. Department of Energy thro h the Offi Number DE-SC0004993. Faradaic efficiency Award suremen were supported by the California Energy under agreement 500-11-023. Work at the sio Mole oundry was supported by the Office of Science, Basic Energy Sciences, of the U.S. Department of Office of under Contract No. DE-AC02- 05CH11231. B. H. Energy vledges the California Energy Corps program for finalncial ackr sup

Eywords: electrochemical CO₂ reduction• colloidal chemistry• nanocrystals • nanocubes • selectivity

- a) C. Graves, S.D. Ebbesen, M. Mogensen, K.S.Lackner, *Renewable Sustainable Energy Reviews* 2011, 15, 1-23; b) X. Lim, *Nature* 2015, 526, 628-630.
- [2] K. P. Kuhl, E.R. Cave, D.N. Abram, T.F. Jaramillo, *Energy Environ. Sci.* 2012, 5, 7050-7059.
- [3] R. Reske, H. Mistry, F. Behafarid, B. R. Cuenya, P. Strasser, J. Am. Chem. Soc. 2014, 136, 6978-6986.
- [4] K. Manthiram, B.J. Beberwyck, A.P. Alivisatos, J. Am. Chem. Soc. 2014, 136, 13319-13325.
- [5] W. Tang, A.A. Peterson, A.S. Varela, Z.P. Jovanov, L. Bech, W.J. Durand, S. Dahl, J.K. Norskov, I. Chorkendorff, *Phys. Chem. Chem. Phys.* **2012**, *14*, 76-81.
- [6] C.W. Li, M.W. Kanan, J. Am. Chem. Soc. 2012, 134, 7231-7234.

COMMUNICATION

WILEY-VCH

- [7] F.S. Roberts, K.P. Kuhl, A. Nilsson, Angew. Chemie. Int. Ed. 2015, 54, 5179-5182.
- [8] C.S. Chen, A.D. Handoko, J.H. Wan, L. Ma, D. Ren, B.S. Yeo, Catal. Sci. Technol. 2015, 5, 161-168.
- [9] Y. Hori, I. Takahashi, O. Koga, N. Hoshi, J. Mol. Catal. A 2003, 199, 39-47.
- [10] K. J.P. Schouten, E. Perez Gallent, M.T.M. Koper, ACS Catal. 2013, 3, 1292-1295.
- [11] K.J.P. Schouten, Z. Qin, E. Perez Gallent, M.T.M. Koper J. Am. Chem. Soc. 2012, 134, 9864-9867.
- [12] K. J. P. Schouten, Y. Kwon, C. J. M. van der Ham, Z. Qin, M. T. M. Koper, *Chem. Sci.* **2011**, *2*, 1902-1909.
- [13] R. Kortlever, J. Shen, K.J.P. Schouten, F. Calle-Vallejo, M.T.M. Koper, J. Phys. Chem. Lett. 2015, 6, 4073-4082.
- [14] R. Kas, R. Kurtlever, H. Yilmaz, M.T.M. Koper, G. Mul, Chem. Electro. Chem. 2015, 2, 354-358.
- [15] K.J.P. Schouten, E. Perez Gallent, M.T.M. Koper, J. Electroanal. Chem. 2014, 716, 53-57.
- [16] W. J. Durand, A. A. Peterson, F. Studt, F. Abild-Pedersen, J.K. Norskov, *Surface Sci.* 2011, 605, 1354-1359.

- [17] a) M. Crespo-Quasada, A. Yarulin, M. Jin, Y. Xia, L. Kiwi-Minsker, J. Am. Chem. Soc. 2011, 133, 12787-12794. b) T. Chen, S. Chen, Y. Zhang, Y. Qi, Y. Zhao, W. Xu, J. Zeng, Angew. Chemie. Int. Ed. 2015, 54, 1-6.
- [18] a) W. Zhu, R. Michalsky, O. Metin, H. Lv, S. Guo, C.J. Wright, X. Sun, A.A. Peterson, S. Sun, J. Am. Chem. Soc. 2013, 135, 16833-16836. b)
 W. Zhu, Y.-J. Zhang, H. Lv, Q. Li, R. Michalsky, A.A. Peterson, S. Sun, J. Am. Chem. Soc. 2014, 136, 16132-16135. c) A. A. Peterson, ECS Transactions 2015, 66, 41-52.
- [19] a) B. Ni, X. Wang, *Adv. Sci.* 2015, *2*, 15006 (1900) 500107. b) T.F. Jaramillo, K.P. Jørgensen, J. Bonde, J.H. Nielsen, Horch, I. Chorkendorff, *Science* 2007, *317*, 100-102. c) J. Kibsgar, J. Ben, B.N. Reinecke, T.F. Jaramillo, *Nature Mater.* 2012, *11*, 94969.
- [20] M.V. Kovalenko, L. Manna, A. Cabot, Z. Mens; D. Yalapin, C. R. Kagan, V.I. Klimov, A.L. Rogach, P. Reiss, and Mirron, P. Guyot-Sionnest, G. Konstantatos, W.J. Parak, T. Hyee, A. Korgel, C.B. Murray, W. Heiss, ACS Nano 2015, 9 (2017) 1057.
- [21] H. Guo, Y. Chen, M. B. Cortex, Liu, Q. Xie, X. Weig, D.-L. Peng, J. Phys. Chem. C 2014, 19, 9801-9808.

[3]

Entry for the Table of Contents

