

Third structure determination by powder diffractometry round robin (SDPDRR-3)

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The results from a third structure determination by powder diffractometry (SDPD) round robin are discussed. From the 175 potential participants having downloaded the powder data, nine sent a total of 12 solutions (8 and 4 for samples 1 and 2, respectively, a tetrahydrated calcium tartrate and a lanthanum tungstate). Participants used seven different computer programs for structure solution (ESPOIR, EXPO, FOX, PSSP, SHELXS, SUPERFLIP, and TOPAS), applying Patterson, direct methods, direct space methods, and charge flipping approach. It is concluded that solving a structure from powder data remains a challenge, at least one order of magnitude more difficult than solving a problem with similar complexity from single-crystal data. Nevertheless, a few more steps in the

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direction of increasing the SDPD rate of success were accomplished since the two previous round robins: this time, not only the computer program developers were successful but also some users. No result was obtained from crystal structure prediction experts. © 2009 International Centre for Diffraction Data. [DOI: 10.1154/1.3200881]

Key words: powder diffraction, crystal structure determination, round robin, blind test

I. INTRODUCTION

Two structure determinations by powder diffractometry (SDPD) round robins (RRs) were organized in 1998 and 2002 (Le Bail and Cranswick, 2001, 2003). In both cases, materials (raw powder diffraction patterns) were distributed worldwide on the internet and the competition was opened to all. The number of potential participants (counting the data downloads) was 70 for the SDPDRR-1, only two of them sent a solution to the second of two problems (one inorganic and one pharmaceutical). Both participants were the conceivers/developers of the computer programs they applied (DASH and CSD). The SDPDRR-2 was proposed in two steps, indexing and then structure solution. Again, only two participants completed the second step, sending the solutions for the first two of three samples (one inorganic, one organometallic, and a fullerene compound). And again, these two participants were the conceivers/developers of the computer programs (FOX and TOPAS). Because no external user of the structure solution computer programs could provide any solution, it was concluded that the SDPD of these compounds was not attainable in a routine manner. As a consequence of the one-dimensional character of powder diffraction data, inducing considerable overlap of diffraction peaks, the complexity of a problem is mainly dependent on two causes, instrumental (or instrumental and sample dependent if the crystallinity is low via contribution to the broadening) and structural. High instrumental resolution can reduce peak overlap problems, favouring the use of synchrotron radiation since more complex problems can in principle be solved. The number of degrees of freedom (DOFs) at the solution stage is the second parameter that determines the complexity of a SDPD problem. This DOF number will depend on the number of atoms in the asymmetric unit or on the number of independent molecules and torsion angles if the molecular geometry is known (one molecule corresponding to six DOFs, three positional and three orientational; one atom in general position corresponding to three DOFs). The effective number of DOF at the solving (sDOF) stage is the central value that we will consider here, being the smallest number of unknown parameters necessary to estimate the initial structural model that will allow for the completion and (Rietveld) refinement of the final crystal structure. At the refinement stage, rDOF can be defined, represented by the number of atomic coordinates which should be refined. The sDOF can be considerably smaller than the rDOF depending on the level of chemical knowledge about the sample which will determine the choice of the structure solution method (classical methods such as Patterson or direct methods or the direct space approach). We are now close to 1500 published structures determined from powder diffraction data, 200/year. Since the previous SDPDRR-2 in 2002, many new computer programs appeared for the purpose of SDPD, and the access to them (academic or commercial) is also much

broader. Recently, the charge flipping approach (Oszlányi and Sütő, 2004) was adapted to powder data (Baerlocher *et al.*, 2007). Moreover, structure prediction has progressed after a series of blind tests (Day *et al.*, 2005). It was thus considered timely to try to verify by a SDPDRR-3 if we are now closer to the expected “routine SDPD.” Moreover it was explicitly requested by the Chairman of the IUCr Commission on Powder Diffraction: “*It is a decade since Armel Le Bail and Lachlan Cranswick issued a challenge to the powder diffraction community to solve two crystal structures from powder diffraction data alone. Despite a generous time scale of around 6 weeks, there were less than a handful of correct solutions. Would the situation be different today? I think so—but I do have a number of caveats. (...) With all the diversity of methodologies presented in this newsletter, are we nearer to saying that structure determination from powder diffraction data is routine. My personal view is that for the general user, it is not. (...) Perhaps then, Armel and Lachlan, we will all be ready for another SDPD round robin.*” (David, 2007).

II. SDPD

The possibilities to completely solve complex structures from powder diffraction data alone are described in many review articles and a few recent books (David *et al.*, 2002; Pecharsky and Zavalij, 2003; Clearfield *et al.*, 2008; Dinnebier and Billinge, 2008). The topic is young since less than 300 published SDPDs were realized ten years ago and less than 2000 today (now running at 200/year). It is thus a quite small niche when compared to structure determination from single-crystal data (~40 000/year), mainly gathered in the repositories of organic crystal structures, the Cambridge structural database (CSD) (Allen, 2002), and of inorganics (ICSD) (Belsky *et al.*, 2002). However, during the last 20 years from several research teams, considerable efforts have been made in order to improve our abilities in that domain. For such a young topic, it is desirable from time to time to offer the community some problems to solve in order to compare the efficiency of various approaches and to provide guidelines to the users, as was frequently the case for the powder diffraction community in the past (surveys of crystallographic program packages, interlaboratory intercomparisons of procedures) or more recently to give a few references: the Rietveld RR (Hill and Cranswick, 1994), the Rietveld refinement guidelines (McCusker *et al.*, 1999), the quantitative phase analysis RR (Madsen *et al.*, 2001), the indexing (Bergmann *et al.*, 2004), and the size strain RR (Balzar *et al.*, 2004). Because SDPD attempts are either succeeding or failing, we do not expect to be able to provide many new and meaningful recommendations but at least the target is to show the state of the art, how experts may approach and eventually solve differently some typical and

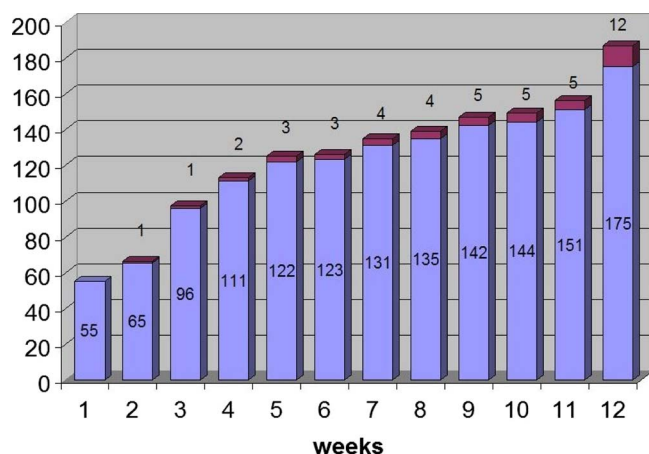


Figure 1. (Color online) Data downloads and solutions received along the 12-week duration (2008) of the round robin.

relatively difficult problems, using the same data. Forming an opinion by only reading the computer program manuals gives a relatively false impression that SDPD can be easily performed in a routine manner.

III. ROUND-ROBIN ORGANIZATION, SAMPLES, AND TIMETABLE

The SDPDRR-3 was proposed for two indexed powder patterns (similarly to the RR-1), a new tetrahydrated calcium tartrate polymorph and a lanthanum tungstate. Because so few solutions were obtained during the first two RRs, more time was allocated, exactly 3 months from February 1st to April 30, 2008 (Figure 1). The information was distributed worldwide through the SDPD and Rietveld mailing lists (having respectively more than 700 and 1000 subscribers), the sci.techniques.xtallography newsgroup, and personal electronic mails were sent to the conceivers/developers of SDPD computer programs and to the previous participants of the crystal structure prediction blind tests. A probably non-exhaustive list of published dedicated SDPD software from which structure solution could be expected for one or both samples is as follow (alphabetical order): DASH (David *et al.*, 2006), EAGER (Harris *et al.*, 1998), ENDEAVOUR (Putz *et al.*,

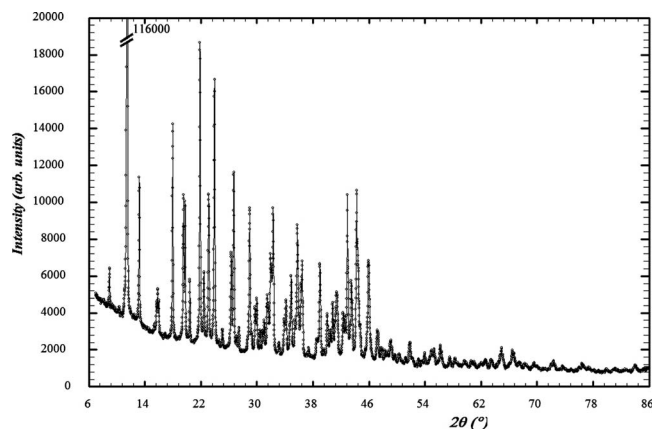


Figure 2. X-ray powder pattern for sample 1 (calcium tartrate tetrahydrate). The second reflection at low angle culminates at ~116 000 counts.

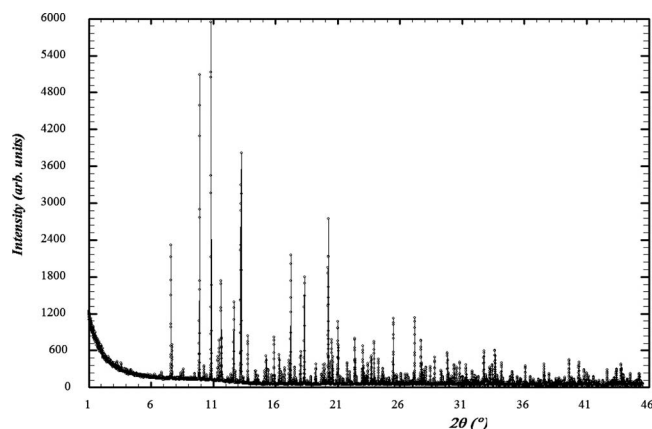


Figure 3. Synchrotron powder pattern for sample 2 (lanthanum tungstate).

1999), ESPOIR (Le Bail, 2001), EXPO (Altomare *et al.*, 2004), FOX (Favre-Nicolin and Cerny, 2002), GEST (Feng and Dong, 2007), OCTOPUS (Harris *et al.*, 1994), ORGANA (Brodski *et al.*, 2005), POWDERSOLVE (Engel *et al.*, 1999), PSSP (Stephens and Huq, 2002), SAFE (Brenner *et al.*, 2002), SA (Andreev *et al.*, 1997), SIMPEL (Jansen *et al.*, 1993), SUPERFLIP (Palatinus and Chapuis, 2007), TOPAS (Coelho, 2000), and XLENS (Rius, 2004). A list of computer programs for the prediction of the packing of molecular structure, which could have been used for the structure solution of sample 1, can be found in Day *et al.*, 2005. The powder patterns for the two samples were experimental, supplied in various standard formats (Figures 2–4). The participants were warned about the possibility of impurity presence as well as systematic zero-point error or even some preferred orientation. The additional details provided are gathered in Table I.

For the new calcium tartrate tetrahydrate form, a CIF provided the tartrate molecular formula as established from two previous single-crystal structure determinations (Hawthorne *et al.*, 1982; Boese and Heinemann, 1993). For the lanthanum tungstate, the formula and cell were provided according to the published literature mentioning composition variations. If the cell parameters given for sample 1 could allow directly for a Pawley (1981) or Le Bail (2005) fit, this was not the case for sample 2. Moreover, the space groups were not defined, either $P1$ or $P-1$ for sample 1, whereas the list of possibilities was longer for sample 2. Those looking

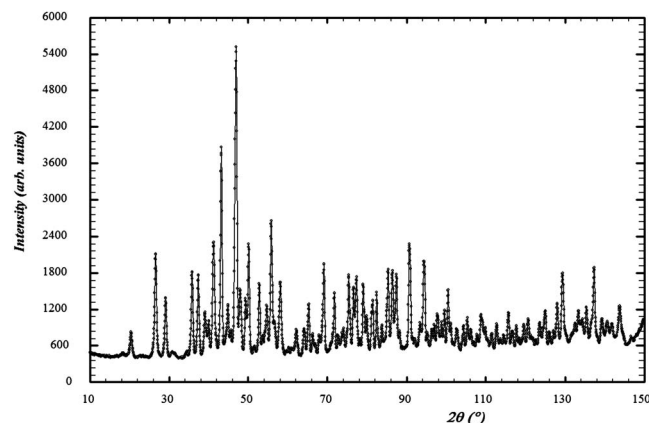


Figure 4. Neutron powder pattern for sample 2 (lanthanum tungstate).

TABLE I. Data provided for samples 1 and 2.

			Sample 1	Sample 2
Probable formula			CaC ₄ H ₄ O ₆ ·4H ₂ O	La ₁₄ W ₈ O ₄₅ or La ₈ W ₅ O ₂₇
Cell parameters	(Å)	<i>a</i> =	8.222	9.039
		<i>b</i> =	10.437	9.039
		<i>c</i> =	6.249	32.60 to 33.65
	(deg)	α =	105.97	90
		β =	107.51	90
		γ =	94.94	120
Symmetry			Triclinic	Hexagonal
ICDD-JCPDS cards			None	032-0502 and 032-0503
Geometry			Reflection	Transmission
			Laboratory instrument	Synchrotron/neutron
			Bragg Brentano	Capillary/vanadium can
Instrument			PANalytical	ESRF BM01/ILL D2B
Wavelength			Cu <i>K</i> α	0.600 44/1.5944 (Å)

accurately at the publications mentioned in the ICDD-JCPDS cards could have found more details, but they were contradictory: “possible space groups *P6₃/mmc*, *P6₃nmc*, and *P-62c*,” favouring the centrosymmetrical group (negative result at the generation of second harmonic)” (Yanovskii and Voronkova, 1975); “hexagonal with *R-62c* space group” (Kovalevsky *et al.*, 1999). Table II provides the sDOF and rDOF corresponding to the various samples of the SDPDRR-1, 2, and 3. However, they are estimated as if the final structure and space group were known by the participants, which was not always the case.

IV. CRYSTAL STRUCTURES AS SOLVED BY THE ORGANIZERS

A. Sample 1: Calcium tartrate tetrahydrate

Large crystals (100 to 200 μm) were extracted from rat kidney. The chemical analysis suggested a hydrated calcium tartrate. The crystal structure determination revealed a new tetrahydrated form. Crystals were found in massive intergrowths and first resisted to the characterization attempts due to the difficulties to separate a real single crystal. Indexing in a triclinic cell was realized from powder diffraction data by

TABLE II. Supposing the space group and structure known, minimal number of degrees of freedom at the structure solution (sDOF) stage from X-ray data for various methods, and total number of refinable (rDOF) atomic coordinates (non-H atoms). DM: direct methods, Patt: Patterson method, DS: direct space method, References: SDPDRR-1, samples 1 (Zhu *et al.*, 1999) and 2 (Clegg and Teat, 2000); SDPDRR-2 samples 1 (Adil *et al.*, 2007) and 2 (Le Bail, 2003); and SDPDRR-3 samples 1 (Le Bail *et al.*, 2009) and 2 (Chambrier *et al.*, 2009).

	SDPDRR-1	SDPDRR-2	SDPDRR-3
	Sample 1	Sample 1	Sample 1
	Sample 2	Sample 2	Sample 2
Chemical formula	[Co(NH ₃) ₅ CO ₃]NO ₃ ·H ₂ O C ₂₂ H ₂₄ N ₂ O ₈ HCl	Al ₂ F ₁₀ [C ₆ N ₄ H ₂₀] Sr ₅ V ₃ (F/O/H ₂ O) ₂₂	CaC ₄ H ₄ O ₆ ·4H ₂ O La ₁₈ W ₁₀ O ₅₇
Space group	<i>P2₁</i> <i>P2₁2₁2₁</i>	<i>P2/c</i> <i>P2₁/c</i>	<i>P-1</i> <i>P-62c</i>
<i>Z</i>	2 4	2 4	2 2
Total independent non-H	15 33	12 30	15 20
Sites to find by DM/Patt	~10 ~20	~10 ~15	~10 to 15 6W+4La
sDOF DS:	6×3+3=21	6×3=18	6+5×3=21
mol+atoms	6+3=9	6×8=48	10×3=30
rDOF	44 99	30 90	45 43
Data quality	Low Medium	Low High	Low High

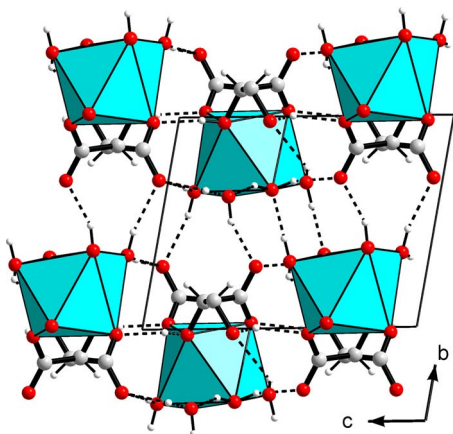


Figure 5. (Color online) Calcium tartrate tetrahydrated crystal structure drawing.

using the MCMAILLE software (Le Bail, 2004). The structure was solved from the powder data by direct space method using the ESPOIR software (Le Bail, 2001) applied to intensities extracted by the Le Bail (2005) method in space group $P-1$. The tartrate molecule was rotated and translated together with the calcium and remaining water oxygen atoms up to find an optimum by a Monte Carlo process. From these convincing results, more efforts were taken with the data collected from a selected “single crystal,” producing a final refinement of much higher quality than from the powder data; this in spite of the relatively high R factor due to peak overlapping from several intergrown domains (Figure 5) (Le Bail *et al.*, 2009).

B. Sample 2: Lanthanum tungstate

In a previous study (Yanovskii and Voronkova, 1975) (from single-crystal data though the structure determination was not completed because it was not possible to obtain good crystals), the commonest six-layered polytype was said to belong to the space group $P6_3/mmc$ (no piezoeffect detected) with cell parameters $a=9.04(1)$ Å and $c=32.60$ to 33.65 Å depending on the composition of the crystal. This cell was confirmed here by a satisfying whole powder pattern fit by using the Le Bail method through the FULLPROF software (Rodríguez-Carvajal, 1993). The extracted intensities from the synchrotron powder pattern were then used for attempting the structure solution by direct space methods as embedded in the ESPOIR software, searching for the heavy W and La independent atoms by a Monte Carlo process. Nothing better than $R_p > 35\%$ could be obtained during various tests in the $P6_3/mmc$ or $P6_3mc$ space groups. Direct or Patterson methods failed as well to provide a satisfying starting model. Then, instead of trying directly the other possible space groups ($P-62c$, $P-31c$, and $P31c$), the search for a solution was made in the $c/6$ subcell in spite of the fact that very intense reflections had to be excluded (scaling the most intense 206 at $I=100$, the 207 is at $I=13$, and the 217 is at $I=14$). Trying various space groups without extinction, a promising model leading to $R_p=22\%$ on 220 remaining peaks was finally obtained from the ESPOIR software in the $P-62m$ space group, corresponding to a La/W=2 ratio (La_2WO_6 formula). No extension of that model in the large

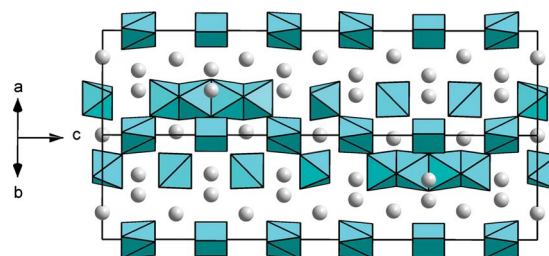


Figure 6. (Color online) Lanthanum tungstate crystal structure drawing.

cell could be obtained in the $P6_3/mmc$ or $P6_3mc$ space groups. Then the other space groups compatible with the $hh-2hl$, $l=2n$ reflection condition were examined ($P-62c$, $P-31c$, and $P31c$). The small initial model could be finally extended in the large cell by using the acentric space groups, for instance with four La and five W independent atom sites in the general or special positions of $P-62c$. Introducing these atomic coordinates into a Rietveld (1969) refinement led then to $R_B=19.7\%$ and $R_F=11.1\%$ when the thermal parameters were refined (most having negative values because of the absence of absorption correction at this stage). From a Fourier difference map, an additional W atom site was detected as well as all the oxygen atoms in ten independent sites. Further refinements suggested that this new W site had to be half occupied, leading to the $La_{18}W_{10}O_{57}$ formula with $Z=2$. A part of the W atoms are found in octahedral coordination but the majority of them are in an unusual trigonal prismatic coordination (Figure 6). This unusual trigonal prismatic coordination was previously observed for the WO_6 group in the X-ray studies of $Pr_3WO_6Cl_3$ (Polyanskaya *et al.*, 1969) and $La_3WO_6Cl_3$ (Brixner *et al.*, 1982), the latter structure being then confirmed from neutron powder diffraction data (Parise and Brixner, 1983). Tests in order to see if reducing the symmetry would allow the half occupied W site to become fully ordered were made in $P31c$ and in various subgroups of $P-62c$ and $P31c$ ($Ama2$, Cc) with no convincing result: the number of atomic coordinates to be refined becoming prohibitive (Chambrier *et al.*, 2009).

V. CRYSTAL STRUCTURES AS SOLVED BY THE PARTICIPANTS

The results for the structure solution step are summarized in Table III (complete reports are available as supplementary materials (<http://www.cristal.org/SDPDRR3/>)). From the 175 potential participants having downloaded the data, nine sent a total of 12 solutions (8 and 4 for samples 1 and 2, respectively). They used seven different computer programs for structure solution. This third round robin received more results compared to first and second round robins and for the first time, contributions were also obtained from nonsoftware developers. This implies that solving structures from powder diffraction data is becoming more accessible to nonexperts compared to previous round robins.

The person-time required for solving from powder data a crystal structure can be almost as short as for single-crystal data, providing that the correct solution is recognized at the first try. Computing time is in general longer for the programs working in direct space methods.

TABLE III. Summary of the results from the SDPDRR-3 contributors for the two samples (S). Note that C1 to C9: contributor number; CF: charge flipping; D: developer/conceiver; De: density analysis; DS: direct space; E: extraction of intensities; E1: extraction of intensities by the Le Bail method; E2: extraction of intensities by the Pawley method; FD: Fourier difference; P: Patterson; R: restraints on bond lengths and angles used at the Rietveld refinement stage; S: Sample; and U: User. Dates for download submission in 2008 are given as day/month.

Contributor (user or developer)	S	Dates for download submission	Programs used	Solved with	Rietveld software	Comments
C1 (U)	1	1/2 to 8/2	GSAS(E1) SHELXS SIR2002 EXPO FOX	FOX(DS)	GSAS(R)	SG: $P-1$. A few different attempts. (1) Tartrate treated as a molecule, calcium, and four oxygen atoms all treated independently of each other; DOF: 24, torsions 3. (2) Only with two or three oxygen atoms bonded to calcium atoms; DOF:18, torsions 3. (3) as (1) with more than four not bonded oxygen atoms; DOF: around 30, torsions 3. This tartrate is a racemate. Sample 2 declared solvable but not tried.
C2 (D)	2	22/2 to 24/2	JANA2000(E) SUPERFLIP EDMA(De)	SUPERFLIP(CF)	JANA2000	SG: $P-62c$. Ten atoms located (all La and W positions), then all oxygen atoms located by a Fourier difference synthesis performed after a Rietveld refinement. One La changed into a W half occupied. Symmetry $-62c$ established by the “derive symmetry” option of SUPERFLIP. The whole process took about 2 h.
C3 (U)	1	14/2 to 5/3	TOPAS	TOPAS(DS)	GSAS	SG: $P-1$. A $\text{CaC}_4\text{O}_6\text{H}_2$ unit was taken from one of the single-crystal structures supplied (six DOFs). The torsional angle around the tartrate central C-C bond was treated as an additional DOF. Four O atoms were also placed in the unit cell as independent entities, giving 12 additional DOF (19 total). When we were confident we had found the solution with lowest R value, we also released all other torsional angles as well as a bond length and a bond angle that defined the position of Ca with respect to the tartrate moiety.
	2	14/2 to 3/4	TOPAS(E)	SUPERFLIP(CF)	GSAS	SG: $P-62c$. Charge flipping (SUPERFLIP) to locate the heavy atom and direct space method (TOPAS) to locate the O atoms (no Fourier difference).
C4 (U)	1	28/2 to 20/3	GSAS(E) SHELXS(P) SHELXL(FD) FORCITE	SHELXS(P)	GSAS	SG: $P-1$. Overlapping peaks were filtered (keeping 150 of 800 after extraction. Patterson procedure using SHELXS allowed to recognize Ca atoms. One of the COO groups was located in a series of Fourier syntheses using different subsets of the originally extracted Fobs's. Then the first Rietveld refinement was performed revealing the water oxygen atoms around Ca. Positions of H atom from geometry optimization with molecular mechanics module forcite (Material Studio).

TABLE III. (Continued.)

Contributor (user or developer)	S	Dates for download submission	Programs used	Solved with	Rietveld software	Comments
C5 (U)	1	2/2 to 21/4	FOX	FOX(DS)	GSAS	SG: <i>P</i> -1. Started as <i>Z</i> matrix from Pb-tartrate example. Start with one rigid body tartrate molecule plus five free atoms: 1Ca and 4O. Later on: release 11 torsion angles one by one within window of 10° around starting model angle. Next <i>Z</i> matrix converted to molecule description for finding final solution.
	2	2/2 to 21/4	FOX	FOX(DS)	GSAS	SG: <i>C</i> 2 <i>cm</i> . Started with set of La atoms and WO ₆ octahedra. Fourier map shows missing heavy atoms. FOX merges atoms when too many atoms have been entered. Sometimes La atoms and W octahedra had to be interchanged. Initially only the X-ray pattern was used to find heavy atom positions. Later on the neutron pattern was added to position O atoms. The final FOX solution had 11 La atoms and 5 W octahedra.
C6 (D)	1	2/2 to 29/4	FULLPROF(E1) FOX	FOX(DS)	TOPAS	SG: <i>P</i> -1. 1 semirigid molecule 2 <i>R</i> , 3 <i>R</i> C ₄ H ₄ O ₆ with internal DOF and restraints as used in FOX. Four rigid molecules H ₂ O. One free atom of Ca. A total of 33 DOFs+internal DOF of the C ₄ H ₄ O ₆ molecule.
	2	2/2 to 29/4	FULLPROF(E1)	FOX(DS)	TOPAS	SG: <i>P</i> 31 <i>c</i> . 6 free atoms of La, 9 free atoms of W, 18 free atoms of O, 1 octahedron WO ₆ . A total of 105 DOFs. Both data sets used jointly for structure solution and refinement.
C7 (D)	1	7/2 to 30/4	EXPO200X	EXPO200X	EXPO200X	SG: <i>P</i> -1. Solved by direct method and simulated annealing DM: whole structure (non-H atoms) recognized at the third phasing trial. SA: Six independent molecules/atoms: one tartrate, four water molecules, one Ca atom. Total number of DOFs=38 (33 external DOFs and 5 torsion angles). Sample 2 declared solvable but probably additional chemical information is needed.
C8 (U)	1	4/2 to 30/4	FULLPROF(E1) ESPOIR	ESPOIR(DS)	FULLPROF	SG: <i>P</i> -1. One CaO ₈ polyhedron, a C ₄ molecule with free torsion angle and 2O atoms. Best solution with <i>R</i> =0.205 at test 91.
C9 (D)	1	20/4 to 1/5	TOPAS(E2) PSSP	PSSP(DS)	TOPAS	SG: <i>P</i> 1. Based on description of the sample as tartrate (rather than mesotartrate), space group <i>P</i> 1 was selected. Two independent tartrates, each with Ca tethered in a likely location to two oxygen atoms. eight independent oxygen atoms. The Ca and carboxyl torsions were varied over a limited range relative to planar. So each Ca tartrate was described by three translations, three rotations, and three torsions. Eight independent oxygens have three translations each for a grand total of 42 parameters. Questionable refinement was ascribed to low resolution data. Sample 2 not really tried.

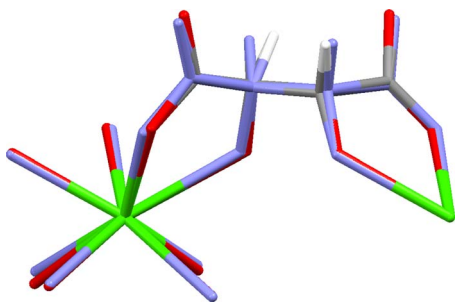


Figure 7. (Color online) Projection along the c axis of the superposition of the tartrate structure as obtained before and after Rietveld refinement (contributor C1).

The fact is that participants may end in multiple solutions ($P1$ or $P-1$ for sample 1 and $P-62c$, $P31c$, or $C2cm$ for sample 2). We can only suppose that these differences would have vanished at the Rietveld structure refinement after more efforts and symmetry-checking ultimate stages (Spek, 2003), which were not included into the SDPDRR-3 targets and so not always realized (this was not a Rietveld round robin). For instance, applying the PLATON symmetry check to the $P31c$ model suggests that it fits at 91% in $P-62c$ (the disordered W atom in $P-62c$ can be ordered in $P31c$); for the $C2cm$ description, PLATON suggests the hexagonal symmetry with $\gamma = 119.98^\circ$; for the $P1$ tartrate description, PLATON does not find the inversion center, saying “no obvious space group change needed.” For these reasons, no metrics will be presented for comparing the structure models with the “known” solutions. Moreover, if the tartrate structure, finally refined from single-crystal data, can be considered as hardly disputable, the lanthanum tungstate structure could be certainly discussed, and some participants could well have tried to publish their own results sometimes differing a bit from the structure determined by the organizers, and even some modern crystallographers would possibly have applied some four-dimensional algorithm to this disordered and clearly modulated (but commensurable) structure. Anyway, given the low level of success, it appears too early to propose a round robin which would address the accuracy issue, comparing the models obtained at the solution stage with the final structures. Nevertheless, when available from the participants, the comparison between the atomic coordinates at the structure solution and at the Rietveld stage shows little differences. For instance in the case of the tartrate (participant C1, see Table III) on a plot of the before/after refinement coordinates (Figure 7), the largest discrepancy in position is observed for the H atoms involved in C-H bonds (other H atoms not located). This round robin is limited to give answers to simple questions “can this be done?” or “is it routine?” Moreover, if the minimal model allowing then to refine and complete the structure in the calcium tartrate case is close to 80% of the atomic positions (including Ca, excluding H), it is limited to the finding of the heavy atoms (W, La) in the tungstate case from the synchrotron data even if some computer programs are able to locate most of the oxygen atoms altogether by combining the synchrotron and the neutron data.

VI. DISCUSSION AND CONCLUSIONS

One should note that for crystallographers, either from powder or single-crystal data, structure solution is realized by expert algorithms embedded in computer programs not by human brains. Human intervention is limited to the choice of the computer program (looking adapted to the problem type) and to the preparation of the data and instructions required by the program. The question why 166 of the 175 participants (~95%) did not send any result may be answered in a simple way: most of the powder diffraction computer programs for structure solution have not attained the level of automatism of the single-crystal software which benefit generally of high-quality three-dimensional intensity data. Very probably, the same problems as those presented in this round robin would have been solved by >95% of the participants from single-crystal data. The monodimensionality of a powder pattern implies that this routine character of a single-crystal approach (unless twinning or other difficulties) is lost because the solution is much less easily recognized, consequently to the data overlapping, and even the recording of an optimum data set requires special care. We could first advice to the software developers to include more efficiency and user friendliness in their computer programs than was done up to now. Second the advice to the powder diffraction expert is to increase his training level in crystallography and this would perhaps improve the current poor SDPD success rate. The SDPDRR-3 establishes that when properly applied by capable users, a variety of algorithmic methods is effective for solving structures from powder diffraction data as provided by round-robin organizers. In the previous SDPDRR-1 and SDPDRR-2, software developers were the only contributors. In comparison with previous RR, solving round-robin structures was found to be feasible by users of existing packages, though developers of structure solving packages predominated (some being able to find a solution in a few hours using their own software). This round robin does not pretend to the generalization to any SDPD from a sample of two structures, and there is no new scientific knowledge to extract from it. It just shows that more science or more user friendliness has to be included in SDPD software if one expects to see more (complex) crystal structures solved by scientists moderately trained in crystallography interested in such problems.

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