University of Wollongong

Research Online

Australian Institute for Innovative Materials -**Papers**

Australian Institute for Innovative Materials

2014

Ultra-small fluorescent inorganic nanoparticles for bioimaging

Zhen Li

University of Wollongong, zhenl@uow.edu.au

Qiao Sun

University of Queensland

Yian Zhu

University of Queensland

Bien Tan

Huazhong University of Science and Technology

Zhi Ping Xu

University of Queensland

See next page for additional authors

Follow this and additional works at: https://ro.uow.edu.au/aiimpapers



Part of the Engineering Commons, and the Physical Sciences and Mathematics Commons

Recommended Citation

Li, Zhen; Sun, Qiao; Zhu, Yian; Tan, Bien; Xu, Zhi Ping; and Dou, S X., "Ultra-small fluorescent inorganic nanoparticles for bioimaging" (2014). Australian Institute for Innovative Materials - Papers. 1038. https://ro.uow.edu.au/aiimpapers/1038

Research Online is the open access institutional repository for the University of Wollongong. For further information contact the UOW Library: research-pubs@uow.edu.au

Ultra-small fluorescent inorganic nanoparticles for bioimaging

Abstract

The novel optical, electrical, and magnetic properties of ultra-small inorganic nanoparticles make them very attractive in diverse applications in the fields of health, clean and renewable energy, and environmental sustainability. This article comprehensively summarizes state-of-the-art fluorescence imaging ultra-small nanoparticles as probes, including quantum dots, metal nanoclusters, carbon nanomaterials, up-conversion, and silicon nanomaterials.

Keywords

inorganic, nanoparticles, bioimaging, fluorescent, small, ultra

Disciplines

Engineering | Physical Sciences and Mathematics

Publication Details

Li, Z., Sun, Q., Zhu, Y., Tan, B., Xu, Z. & Dou, S. Xue. (2014). Ultra-small fluorescent inorganic nanoparticles for bioimaging. Journal of Materials Chemistry B, 2 (19), 2793-2818.

Authors

Zhen Li, Qiao Sun, Yian Zhu, Bien Tan, Zhi Ping Xu, and S X. Dou

Ultra-small fluorescent inorganic nanoparticles for bioimaging

Zhen Li,*a Qiao Sun,b Yian Zhu,b Bien Tan,c Zhi Ping Xu,*b and Shi Xue Doua

Received (in XXX, XXX) Xth XXXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX DOI: 10.1039/b000000x

⁵ The novel optical, electrical, and magnetic properties of ultra-small inorganic nanoparticles make them very attractive in diverse applications in the fields of health, clean and renewable energy, and environmental sustainability. This article comprehensively summarizes state-of-the-art fluorescence imaging using ultra-small nanoparticles as probes, including quantum dots, metal nanoclusters, carbon nanomaterials, up-conversion, and silicon nanomaterials.

10 1 Introduction

When the size of inorganic materials is reduced to the nanoscale range, they exhibit unusual optical, electrical, magnetic, mechanical, and chemical properties, distinctly different from those in their bulk analogues. For example, semiconductor 15 nanocrystals (usually referred to as quantum dots (QDs)) exhibit strong size-dependence of their optical properties when their size is smaller than the Bohr exciton radius. 1 Magnetic iron oxide nanoparticles become superparamagnetic when their size is reduced below the critical size where they behave as individual 20 magnetic domains.² Carbon nanotubes show remarkable tensile strength,3 and graphene exhibits remarkably high electron mobility.⁴ Their novel properties make these nanomaterials very attractive in diverse applications, ranging from energy conversion and storage to biomedical imaging. In this article, we summarize 25 the recent advances in ultra-small inorganic nanoparticles for fluorescence imaging (Table 1), especially those smaller than 10 nm as they are easily taken up and excreted, and show longer blood circulation time in comparison with larger ones.

For fluorescent materials, there are two kinds of 30 photoluminescence mechanisms, i.e. down conversion and up conversion.⁵ The down-conversion process normally absorbs one high energy photon and emits a low energy photon, e.g. a Stokesshift emission. In contrast, up-conversion is an anti-Stokes process that converts the absorbed low energy light into higher 35 energy emission via multiple absorptions or energy transfer processes. The fluorescence generated by both processes has long been used in molecular imaging to visualize cell biology at many levels.6,7 The first fluorescence imaging could be dated back to 1924 when Policard observed red fluorescence from endogenous 40 porphyrins in tumours illuminated with an ultraviolet light. Since then, advances in molecular biology, organic chemistry and material science have revealed several classes of promising probes for fluorescence imaging, which include small organic dyes, fluorescent proteins, and fluorescent inorganic 45 nanoparticles. 6 Compared with organic dyes and fluorescent proteins, fluorescent inorganic nanoparticles have several distinct advantages. For example, QDs have high absorbance, high QY, narrow emission, large Stokes shifts, and high resistance to

photobleaching. These properties render them robust for fluorescent probes for biolabelling and bioimaging. 1 In recent years, other fluorescent nanomaterials, such as ultra-small metal nanoclusters, fluorescent carbon and graphene dots, upconversion nanocrystals, and silicon nanoparticles have been exploited as alternatives to conventional QDs. In the following sections, we introduce these fluorescent nanomaterials from viewpoints of preparation and functionalization to satisfy the requirements for routine labelling and imaging of cells and tissues. Advanced applications of fluorescent nanomaterials in living systems as sensors for enzyme, oxygen, metal ions, and pH, have readily been described elsewhere. 13-16

For bioimaging, fluorescent nanoparticles should have watersolubility, biocompatibility, chemical- and photo-stability. They should also have uniform size and high quantum yield (QY) for optimized brightness, narrow and symmetric emission for 65 multiplexing and colour saturation, and minimized blinking for light output stability. In the second part, we introduce the development in synthesis and surface modification of fluorescent ODs (especially CdSe- and CdTe based II-VI ODs) to result in water-soluble, biocompatible and highly stable QDs with high 70 QY, together with their routine bioimaging applications and their toxicity. In the third section, we describe extremely small metal nanoclusters (usually smaller than 2 nm) as an emerging fluorescent probes, and address the difficulties in their synthesis, characterization, modification, and imaging application. In the 75 fourth part, we bring in carbon-based fluorescent nanoprobes including carbon dots and graphene quantum dots (GODs), which are usually smaller than 10 nm. These carbon-based nanoprobes have excellent biocompatibility and unique properties (e.g. both up-conversion and down-conversion emissions). In the fifth 80 section, we briefly introduce lanthanide-based up-conversion nanocrystals, which have attracted considerable attention in recent years. Most of up-conversion nanoprobes have a large size (>10 nm) and are out of the scope of this article. In the sixth part, we discuss fluorescent silicon nanoparticles, which have excellent 85 biocompatibility and stability. In the last part, we highlight the major challenges and perspectives of ultra-small fluorescent

nanoparticles and fluorescence bioimaging.

Table 1. Comparison of different types of fluorescent inorganic nanoparticles.

Types	Representatives	Preparation	Size (nm)	Advantages	Disadvantages	Applications
QDs	CdSe, CdTe, InP, CuInS ₂ , CuInSe ₂ , and their core- shell nanostructures		< 10	Tuneable size and fluorescence, high QY and relatively stable	toxicity	Fluorescent labels and sensors
Metal NCs	Au, Ag, Cu, Pt, Po	Reduction of metal salts or etching of large metal nanoparticles	< 2	Ultra-small size, easily taken up and excreted.	sensitive fluorescence, difficult for modification and functionalization	Imaging of cells and tissues. Sensors
Carbon-based materials	C-dots, GQDs	Solvothermal and hydrothermal methods	<10	Tunable fluorescence, excellent biocompatibility	Instability and unclear of fluorescence mechanism	Fluorescent biomarkers and sensors
UCNs	Yb ³⁺ /Ln ³⁺ -doped NaYF ₄ , GdYF ₄ , and their core- shell nanostructures	Solvothermal and hydrothermal methods	most >10, few <10	1	Low QY, potential toxicity	Multimodal imaging agents and drug carriers
Si NPs	Si	Etching annealed SiO_x , reduction of $SiCl_4$, and reaction of Na_4Si_4 with NH_4Br	< 5	Small size, Ultrahigh stability	Laborious synthesis, Low QY, Difficult to tune fluorescence	Long-term imaging and labeling

QDs: quantum dots; UCNs: up-conversion nanocrystals; NPs: nanoparticles; GQDs: graphene quantum dots; QY: quantum yield

2 Semiconducting Fluorescent QDs

5 2.1 Synthesis of monodisperse QDs with high QYs

Fluorescent QDs include semiconducting nanoparticles from Groups IV (Si and Ge dots), ¹⁷⁻²⁰ II-VI (CdE and ZnE, E = S, Se, and Te), III-V (InP), and I-III-VI (CuInS₂, CuInSe₂), ²¹⁻²³ in which the II-VI QDs (especially CdSe and CdTe based QDs) have been 10 extensively investigated as prototypes of semiconductor QDs due to their strong quantum confinement effects and high fluorescence QYs. II-VI colloidal fluorescent QDs can be prepared in organic solvents or aqueous solutions. Organic routes are usually selected to prepare monodisperse and highly 15 fluorescent QDs. Discovered in 1981, QDs did not receive intensive attention until 1993, when a breakthrough in preparation of colloidal QDs in solution was achieved.²⁴ Monodisperse cadmium chalcogenide (CdE, E = S, Se, and/or Te) QDs were prepared by fast injection of a solution of 20 precursors (organometallic Cd and Se/S/Te dissolved in trioctylphosphine (TOP)) into a high-boiling-point (~ 300 °C) coordinating solvent trioctylphosphine oxide (TOPO).²⁴ These QDs had a narrow particle size distribution with 10% standard deviation, which was reduced to 5% after size-25 selective precipitation. Their fluorescent QY was about 10%.

The key in this "TOPO-TOP" approach is a burst of nucleation which can be effectively separated from the growth process.²⁵ The use of highly flammable and toxic dimethylcadmium, however, limited the applicability of this approach at that time. Extensive efforts have been made to develop and optimize this approach by using various stable and low toxicity precursors²⁶ (e.g. cadmium oxide, CdO;

cadmium carboxylate, Cd(OOCR)₂; selenium oxide, SeO₂, etc.), non-coordinating solvents (e.g. 1-octadecene, ODE) and stabilizers²⁷ (e.g. octyldiphenylamine (ODPA), diethylenetriaminepentaacetic acid (DTPA), hexadecyl amine (HDA), etc.) to generate monodisperse QDs with high QY.

Monodisperse II-VI QDs with different sizes and shapes can be obtained by controlling their nucleation and growth processes 40 through optimization of monomer concentration and reactivity, molar ratio, reaction temperature, ligands, etc. (Figure 1).²⁸⁻³¹ It has been found that slight modification of reaction parameters can lead to a broad variety of particle sizes and shapes. For example, Peng et al. demonstrated that the size and size 45 distribution of CdSe dots can be manipulated by the monomer concentration.³² At high monomer concentrations, the smaller nanoparticles grow faster than larger ones, which results in the size distribution being "focused". If the monomer concentration drops below a critical threshold, the smaller particles are depleted 50 as larger ones grow (i.e. Ostwald ripening), and the size distribution gets broader or is "defocused". Controlling the nanoparticle growth kinetics can result in a narrow particle size distribution (5% standard deviation) without the sizeselective precipitation.³²

the maximum value and then decreased with the growth time. 33 Such a photoluminescence bright point indicates an optimal surface structure/reconstruction. Use of a large Se/Cd ratio (10/1) can result in very bright CdSe QDs with QY of 85% at room temperature. The high QY of these QDs is attributed to stabilization of organic ligands on the surface. Since these ligands can be chemically degraded and detached from the surface, the photo- and chemical stability of the core is

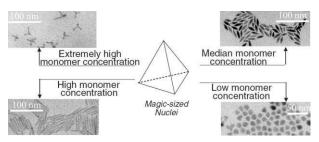


Figure 1. Transmission electron microscope (TEM) images and growth paths of CdSe nanocrystals with different morphology. Reproduced from Ref. 30.

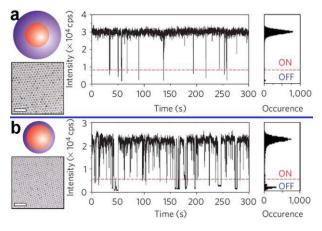


Figure 2. TEM images and blinking behaviour of core-shell CdSe@CdS nanoparticles: (a) 2.2 nm CdSe core with 2.4 nm CdS shell; (b) 2.2 nm CdSe core with 0.7 nm CdS shell. Reproduced from Ref. 36.

sometimes severely affected. In order to improve their luminescence and photostability, wide-band-gap shells (e.g. cadmium sulphide (CdS) and zinc sulphide (ZnS)) have been coated onto their surface to form core-shell QDs. 34, 35 Li et al. s developed a successive ion layer adsorption and reaction (SILAR) technique to epitaxially grow shells in a noncoordinating solvent.34 The resultant core-shell CdSe@CdS QDs had a QY of 40%. Xie and his co-workers further developed this approach to prepare CdSe-core-multishell QDs with QY 10 up to 85%. 35 Recently, Chen et al. used cadmium oleate and octanethiol as Cd- and S-precursors, and prepared nearly perfect core-shell CdSe@CdS ODs with the highest OY (97%) ever reported (Figure 2).³⁶ The slow continuous precursor infusion and the relatively low reactivity of 15 octanethiol provide optimal conditions for passivation of the CdSe surface and growth of the CdS shell. Compared with conventional core-shell CdSe@CdS QDs, these new QDs featured significantly suppressed blinking, with an average fluorescence on/off time ratio of 94:6 for single large core-20 shell nanocrystals (Figure 2). The blinking was gradually suppressed with increasing shell thickness.³⁶ In addition to Wurtzite core-shell CdSe@CdS QDs, zinc-blende core-shell analogues with suppressed blinking (>95% on time) were also prepared by Oin and co-workers.³⁷ These zinc blende 25 core-shell CdSe@CdS QDs exhibited a QY of 90%. It should be noted that non-blinking core-shell CdZnSe@ZnSe QDs, which exhibited complete suppression of blinking on the time scale from milliseconds to hours, were successfully prepared

by Wang *et al.* ³⁸ The high fluorescence QY and non-blinking ³⁰ QDs make them very useful in applications requiring a continuous output of single photons.

During the preparation of QDs, attempts to adjust the growth kinetics of the QDs incidentally led to the development of onedimensional (1D) nanorods.²⁹ By using very high precursor 35 concentrations and a defined admixture of alkylphosphonic acids and trioctylphosphine oxide (TOPO), 1D and even more complex structures such as arrows, teardrops, or tetrapods were synthesized (Figure 1).30 Recently, we demonstrated that doped and undoped 1D semiconductor nanostructures can be produced 40 by using a lower precursor concentration in the presence of bismuth nanoparticles. 39-44 This is in contrast to the synthesis of 1D nanostructures without nanocatalysts. These nanowires exhibited unusual optical, 45 , 46 electronic, 47 , 48 and magnetic 40 properties with potential diverse applications. 48, 49 The fast 45 growth process resulted in crystal twinning and defects in the nanowires, 50, 51 leading to a low fluorescence QY (< 1%) which could be improved by more than three times through coating with a wide-band-gap shell.⁵²

Despite monodispersity, high QY, and stability, these QDs generated in organic solvents are normally hydrophobic and have to be modified in order to be water-soluble and biocompatible for bioapplications. The modification leads to the decrease in fluorescence QY, e.g. the QY of above perfect core-shell CdSe@CdS QDs decreased from 94% to 55 77% after transferred into PBS solution with PEG-SH. Therefore, direct preparation of QDs in aqueous solution has been developed almost simultaneously.

The aqueous approach was firstly adopted by Henglein et al. to prepare CdS nanoclusters in 1982.53 They also reported 60 the first example of the preparation of CdTe QDs in aqueous solution. 54 The resultant CdTe QDs did not show fluorescence, however. Rogach et al. synthesized stable fluorescent CdTe QDs with a QY of 3% by using thioglycerol and mercaptoethanol as stabilizers.⁵⁵ Later on, many efforts 65 were made to improve OD fluorescence OY by using different stabilizers (thioglycolic acid (TGA); mercaptopropionic acid (MPA)), precursor ratios, and manners of heating (hydrothermal and microwave methods).⁵⁶ Under the optimal conditions, the QY of water-70 soluble CdTe-based QDs can reach as high as 84%, which is comparable to that of the above-mentioned hydrophobic QDs. High fluorescence QY also can be obtained by surface modification of as-synthesized QDs with illumination. For example, the fluorescence QY of CdTe QDs was drastically 75 improved from 8% to 85% after 28-day illumination, due to the formation of the core-shell structure (i.e. CdTe@CdS) with the assistance of illumination.⁵⁷

The above water-soluble QDs were normally prepared in strong basic solution (pH > 8), which limits their bio-applications, as most biological activities take place under neutral-pH conditions. Adjusting the solution pH to neutral could quench the fluorescence of the QDs. Therefore, it is of great interest to develop a novel approach for preparing highly fluorescent water-soluble QDs from stable precursors under neutral pH conditions. Recently, we have successfully synthesized highly fluorescent (84% QY) mercaptosuccinic acid

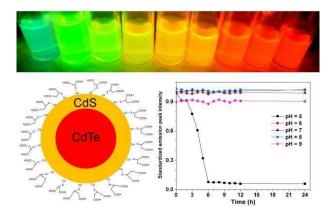


Figure 3. Tunable core-shell CdTe@CdS QDs with high stability. Reproduced from Ref. 58.

(MSA)-capped CdTe/CdS QDs using stable Na₂TeO₃ as the Te source via a one-pot reaction under neutral conditions (Figure 3).⁵⁸ A novelty of this approach is the use of MSA, which exhibits the features (p $K_{COOH1} = 3.30$, and p $K_{COOH2} = 4.94$) of ₅ both TGA (pK_{COOH} = 3.53) and MPA (pK_{COOH} = 4.32) in terms of acidity. MSA can effectively stabilize QDs in a wider pH range (pH = 6 - 9) with better protection because of its stronger interactions with the surface Cd2+ ions and its stronger steric effects. In addition, slow decomposition of MSA-Cd complexes 10 forms a thin layer of CdS on the surface of CdTe nanocrystals, decreasing the surface defects and leading to high fluorescence QY. Another novelty is the use of sodium citrate as buffer. The resultant QDs show higher fluorescence QY than those stabilized with TGA or MPA obtained from the conventional aqueous 15 method. They also show lower cytotoxicity at certain concentrations due to the unique structure of MSA and the formation of a CdS shell on the surface of the CdTe core.⁵⁸

In addition to organic and aqueous routes, QDs can be produced in living organisms. Stürzenbaum *et al.* demonstrated that the earthworm's metal detoxification pathway can be exploited to produce water-soluble and biocompatible CdTe QDs.⁵⁹ This bioapproach is time-consuming (11 days), however, and the resultant QDs have a low fluorescence QY (8%), so this method cannot be used for large-scale preparation.

25 2.2 Surface modification of QDs

From the viewpoint of bioapplications, QDs should have excellent water-solubility, biocompatibility, and stability. These properties not only depend on their particle size, shape and composition, but also rely on their surface structure and surface 30 charge. More importantly, the surface properties of QDs determine their bio-interface interactions, cellular endocytosis and intracellular distribution, in vivo biodistributions, metabolism, and fate. 60-63 Engineering surface of QDs therefore becomes highly important as this process can improve these properties and 35 introduce additional functions. 10 Medintz et al. recently summarized the strategies for surface modification and bioconjugation of QDs. 11 One popular strategy for hydrophobic QDs is ligand exchange, which not only transfers them from organic solvents into aqueous solution, but also provides 40 functional groups for further conjugation with biomolecules. 64 Small water-soluble molecules such as TGA, MPA, and

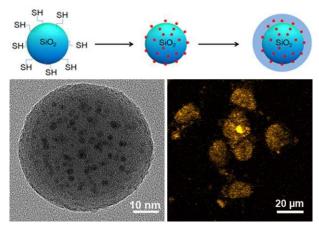


Figure 4. Preparation of sandwich-like SiO₂@CdTe@SiO₂ nanoparticles for cell labeling. Reproduced from Ref. 72.

dihydrolipoic acid (DHLA) are often used. Unfortunately, ligand-exchange can lead to a huge loss of fluorescence due to the changes in surface properties. These small molecules cannot prevent QDs from oxidation and degradation. Thereby a number of polymers (e.g. polyethylene glycol (PEG), and polyethylenimine (PEI)), proteins, peptides, and liposomes have been adopted to coat QDs. Similar to small-molecule modified QDs, these surface-coated flexible polymers and biomolecules are less resistant to oxygen and chemicals, and have little impact on the improvement of the photo- and chemical stability of QDs. Therefore, organic-modified QDs still face the issues of toxicity, instability and the loss of fluorescence, despite the significant progress achieved in recent years.

Compared with unmodified and organically modified QDs, QDs coated with an inorganic shell show higher stability in terms of both chemistry and fluorescence. Silica is one of the most popular inert materials used for surface modification, and has a few distinct advantages, 65 including: (1) a non-porous silica shell 60 can protect QDs from environmental damage and improve their stability; ⁶⁶ (2) the silica shell can effectively inhibit the release of toxic Cd²⁺ ions and thus reduce the QDs' toxicity; ⁶⁶ (3) the silica coating can provide a hydrophilic surface and functional groups for conjugating with biomolecules.⁶⁷ The silica shell can be 65 formed on the surface of the QDs by the Stöber method⁶⁸ or the reverse microemulsion approach.⁶⁹ Both methods have their own advantages and disadvantages, but one common challenge is the preparation of highly fluorescent QDs@SiO2 nanoparticles with tuneable size, as the fluorescence of QDs is drastically decreased 70 during silica coating.

In 2004, Nann *et al.* prepared single-dot@SiO₂ nanoparticles by the Stöber method.⁷⁰ Yang and his co-workers prepared similar CdTe@SiO₂ nanoparticles by the reverse microemulsion approach.⁶⁹ These single-dot@SiO₂ nanoparticles show a low fluorescence QY (< 10%), however. Later on, the fluorescence QY of CdTe@SiO₂ nanoparticles was improved to 47%. In comparison with incubated CdTe QDs (83%), nearly 40% of the fluorescence was still lost during silica coating.⁷¹ The formation of single-dot@SiO₂ nanoparticles is attributed to the electrostatic repulsion between QDs and silica intermediates.

In order to improve the fluorescence QY and the number of QDs in each SiO₂ nanoparticle, we successfully prepared sandwich-like SiO₂@CdTe@SiO₂ (SQS) nanoparticles using a

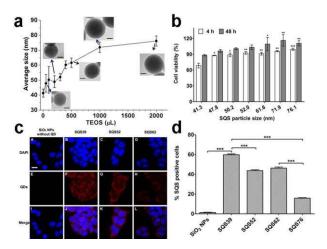


Figure 5. (a) Size-dependence of SiO₂@CdTe@SiO₂ nanoparticles on tetraethyl orthosilicate (TEOS) volume; (b-d) size-dependence of cytotoxicity and cell uptake. Reproduced from Ref. 74.

novel strategy (Figure 4).⁷² We started from the synthesis of the thiol-capped SiO₂ core. The surface thiol groups can tightly anchor CdTe QDs on the surface of SiO₂ nanospheres. Then, a silica layer was coated on the SiO₂@CdTe to form SQS nanoparticles. During the silica coating, it is important to add an appropriate amount of 3-mercaptopropyl-trimethoxysilane (MPS) for pre-coating in order to get highly fluorescent sandwich-like nanoparticles. Compared with other QDs@SiO₂ nanoparticles, our SQS nanopaticles show the highest fluorescence QY ever reported (up to 61%). They also show higher stability and lower toxicity in comparison with SiO₂@CdTe nanoparticles.

During the modification of QDs, the overall particle size has to be strictly controlled because it can dramatically influence the nanoparticle biological behaviour, such as cell internalization, 15 tumour targeting and penetration, in vivo systemic and lymphatic biodistribution, metabolism, and clearance. Nanoparticles with a size of 20-60 nm have shown distinct biodistribution, tumour penetration, and cellular tracking properties.⁷³ Therefore, we prepared a series of SQS nanopaticles with sizes in the range of 20 39 nm to 76 nm by controlling the reaction parameters, including the amount and the type of silica precursor, the ratio of silica precursor to ammonia, and the ratio of H₂O to surfactant.⁷⁴ These SQS nanoparticles exhibited strong size dependence of their stability, toxicity, and cellular uptake (Figure 5). Our findings 25 highlight the importance of controlling particle size and shell thickness during the preparation of fluorescent QDs@SiO2 coreshell nanoparticles.

2.3 Fluorescence imaging of QDs

The earliest bioapplications of fluorescent QDs were reported in 1998. ^{64, 67} Bruchez *et al.* coated core-shell CdSe@CdS QDs with a thin layer of silica and then conjugated them with biotin. ⁶⁷ The biotinylated QDs were successfully applied to label 3T3 mouse fibroblast cells. Chan *et al.* used small molecule TGA to transfer hydrophobic CdSe@ZnS QDs into water solution, and then ³⁵ conjugated them with transferrin proteins. ⁶⁴ The authors incubated TGA-modified QDs and transferrin-QD conjugates with HeLa cells, respectively, and found that no QDs could be observed inside the cell in the absence of transferrin. In contrast,

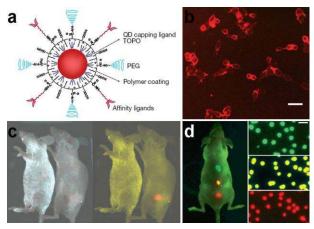


Figure 6. (a) Structure of a multifunctional QD probe; (b) C4-2 cells labelled with multifunctional QDs; (c) *In vivo* targeted imaging using multifunctional QDs; (d) multicolour capability of QD imaging in live mouse. Reproduced from Ref 76

QDs were internalized into the cells in the presence of transferrin due to the occurrence of receptor-mediated endocytosis.

Motivated by the above pioneering research, extensive nonspecific and targeted bio-labelling and imaging have been carried out at different levels, ranging from *in vitro* to *in vivo* models. 10-12 Nonspecific cellular labelling involves the use of 4s hydrophobic and electrostatic interactions between surface-capping molecules of QDs and biomolecules in the cell membrane. Thus, their surface ligand properties and the cell type largely determine the nonspecific adsorption and uptake of QDs. In most cases, such nonspecific adsorption is unwanted, as this reduces the selectivity and targeting efficiency. In order to overcome nonspecific adsorption, PEG and its derivatives have been used to modify the QD surface, as they can effectively minimize and prevent the nonspecific interactions of QDs with biomolecules, cells, and tissues.

Similar to the *in vitro* nonspecific adsorption of cells, non-targeted QDs can accumulate within tumours through the enhanced permeability and retention (EPR) effect. Such passive targeting is attributed to the leakiness of the tumour vasculature and the poor lymphatic drainage, which enables QDs or other nanoparticles to accumulate in tumours. The EPR effect could lead to more than 50 times as great nanoparticle accumulation in tumours compared with healthy tissues. It is difficult, however, to maximize nanoparticle accumulation through the EPR effect, as this effect varies from tumour to tumour, and strongly depends on the particle size and the surface charge. In addition, the EPR effect is not commonly observed in some types of cancers such as gastric and pancreatic cancers.

An alternative approach is active targeting, which can be achieved by conjugating QDs with targeting moieties such as small molecules (e.g. folic acid and hyaluronic acid), peptides (e.g. arginine-glycine-aspartic acid (RGD)), and proteins (e.g. antibodies, antibody fragments, transferrin, etc.). In 2004, Gao and colleagues reported a landmark work on *in vivo* cancer targeting with QDs (Figure 6). They first encapsulated hydrophobic CdSe@ZnS core-shell QDs with an ABC triblock copolymer (i.e. polybutylacrylate-polyethylacrylate-polymethacrylic acid) by using hydrophobic-hydrophobic

interactions between the capping ligands of QDs and the hydrophobic segments of the block copolymer. Then, they conjugated tumour-targeting ligands and drug-delivery functionalities with the polymethacrylic acid segment. The *in s vivo* study showed that these QD probes accumulated at the tumour site through the EPR effect, and the specific antibodyantigen interactions. It is worth mentioning that passive targeting is much slower and less efficient than active targeting.

Although targeted nanoparticles hold much promise, and the 10 concept was introduced more than 30 years ago, none of them has been clinically approved. 75 One possible reason is the huge gap between cost and benefit. Compared with expensive antibodies and other targeting ligands, cost-effective small molecules such as folic acid have been adopted. Folic acid and folate conjugates 15 can be specifically recognized by the folate receptor (FR), which is a glycosylphosphatidylinositol-anchored protein. The alpha isoform of FR (FR-α) is found to be overexpressed in many epithelial cancers, but not highly expressed in normal tissues except for the kidney. Since the affinity of FR to folic acid and ₂₀ folate conjugates is relatively high ($K_d \approx 100$ pM), FR- α has been extensively investigated for tumour targeting,77 including many studies focusing on QDs. For example, folic acid was conjugated to PEG and subsequently deposited onto N-acetyl-L-cysteine (NAC)-stabilised CdTeS QDs, which was demonstrated to be 25 able to target tumours in mouse models. 78 Another small targeting molecule is hyaluronic acid, which is widely distributed throughout connective, epithelial, and neural tissues. Hyaluronic acid, associated with tumour angiogenesis and progression, ⁷⁹ has been conjugated to QDs for tumour targeting, as it can 30 specifically bind with CD44, a cell-surface glycoprotein overexpressed in many tumour types. Therefore their conjugates have not only cancer targeting characteristics, but also the capability for imaging lymphatic vessels.80

In addition to the high cost, the low targeting efficacy and the unclear mechanism could also limit their clinical applications. This is because not all cancer cell types overexpress the same unique receptors, and the overexpressed receptors are often present on normal cells. Moreover, the density of the targeted receptors on tumour cells could be another factor influencing the targeting efficacy. For II-VI QDs, the biggest challenge for their clinical applications is their potential toxicity, as discussed in the following section.

2.4 QD toxicity

Most II-VI QDs consist of toxic elements such as cadmium,
⁴⁵ lead, mercury, etc. Their toxicity has always been of concern and
could limit the diversity of their applications, such as in solar
cells, light-emitting diodes, flat-screen televisions, and
biomarkers. ⁸¹ The bio-toxicity depends on multiple factors, ⁸²
which can be mainly classified into two groups: (1) the inherent
⁵⁰ properties of QDs, including QD size, charge, composition,
concentration, and outer-layer coating bioactivity (capping
material, functional groups); (2) environmental factors such as
oxidation, photolysis, and mechanical effects. A number of
studies show that appropriate surface modification, modulating
⁵⁵ the surface charge, and controlling the QD dosage can effectively
reduce QD cytotoxicity. Previously, we demonstrated that coating
QDs with silica shells can improve their stability and reduce the

toxicity (Figure 5).⁷⁴ Some research has shown that the release of Cd²⁺ and the oxidation products of anions are responsible for their bio-toxicity.⁶⁶ The QDs themselves (i.e. non-degraded QDs) are not acutely toxic, and they can be retained in the body for two years and remain fluorescent.

In 2007, Choi and co-workers studied the renal clearance of QDs. 83 They chose cationic, anionic, zwitterionic, and neutral 65 molecules to modify CdSe@ZnS core-shell QDs and tested their binding with serum proteins. They found that the QD surface charge has a profound effect on the adsorption of serum proteins and the hydrodynamic diameter. Cationic or anionic charge led to the hydrodynamic size increasing from around 5 nm to over 15 70 nm after incubation with serum. Neutral (PEGylated) QDs did not aggregate, but had a large size. Zwitterionic coatings prevented serum protein adsorption and produced the smallest hydrodynamic size. The biodistribution results show that a final hydrodynamic diameter < 5.5 nm resulted in rapid and efficient 75 urinary excretion and elimination of QDs. In their later report, the authors conjugated small targeting molecules on the surface of zwitterionic coatings of QDs.84 These targeted probes were also cleared by the kidneys when their hydrodynamic size was smaller than 5.5 nm, which sets an upper limit of 5-10 ligands per QD for 80 renal clearance. The animal models demonstrated their performance for in vivo targeted imaging and renal clearance within 4 h post-injection.

Recently, Ye *et al.* injected phospholipid micelle-encapsulated CdSe/CdS/ZnS QDs into rhesus macaques, and tracked the relevant markers in the next 90 days. Their results demonstrated that the acute toxicity of these QDs *in vivo* is minimal. Accumulation of an initial dose of Cd was found in the liver, spleen, and kidneys, however, even after 90 days, indicating slow breakdown and clearance of the QDs. Although QDs have not shown acute or short-term toxicity, comprehensive assessments of their long-term bio-toxicity are needed to confirm the ultimate fate of these heavy metals and the impact of their persistence in primates for potential clinical use.

95 3 Fluorescent metal nanoclusters

Since QDs have potential toxicity and long in vivo retention time, many efforts have been made to develop alternatives to them. An alternative is fluorescent metal nanoclusters, which have attracted considerable attention during the past several years. It is well 100 known that large nanoparticles of metals such as Au, Ag, and Cu possess the face-centred cubic (fcc) structure and the surface plasmon resonance (SPR) property. 86 Their SPR absorption is due to the collective oscillation of electrons on the surfaces, and it is strongly dependent on the particle size. When that size is smaller 105 than the electron mean path (e.g. 20 nm for Au nanoparticles), the conducting electrons in the ground states and excited states are confined.86 The large metal nanoparticles have very low fluorescence emission. Very interestingly, when their size is further reduced below 2 nm, the ultra-small nanoclusters possess 110 different crystal structures and exhibit strong photoluminescence while their unique SPR property disappears.

Nanoclusters bridge the gap between molecules and nanoparticles, and could simultaneously display the properties of both molecules and nanoparticles. Their novel optical, electronic,

and catalytic activities make them very useful in ultrasensitive detection, biolabelling, bioimaging, and catalysis. ⁸⁷⁻⁹⁰ The big challenge, however, is how to controllably synthesize metal nanoclusters with defined size, composition, crystal structure, and ⁵ surface properties. ^{88, 91}

3.1 Synthesis of fluorescent metal nanoclusters

Compared with large nanoparticles, metal nanoclusters are difficult to synthesize and functionalize because they only consist of a few to tens of metal atoms. They are very sensitive to slight variation of the environment, such as solution pH, ion strength, solvents, oxygen, temperature etc. They have very high surface-area-to-volume ratios and tend to aggregate into large particles. In general, fluorescent metal nanoclusters can be prepared by reduction of metal precursors or etching of large nanoparticles in the presence of strong stabilizers such as small thiol-molecules, polymers, and biomolecules.

Reduction of metal precursors such as salts and complexes is a straightforward way to produce fluorescent metal nanoclusters. Au nanoclusters are usually chosen as representative for 20 investigation due to their high chemical stability, easy preparation, and biocompatibility. The first observation of Au photoluminescence from its ingots, single-crystal slices, and films, with a QY of 10⁻¹⁰, was reported by Mooradian in 1969.⁹² The extremely low QY did not attract any attention until 25 Wilcoxon et al. observed fluorescence from colloidal Au nanoparticles with a OY of $10^{-5} - 10^{-4.93}$ The authors prepared colloidal Au nanoparticles through reduction of HAuCl₄ by citrate in water, or by metallic sodium dispersion or lithium trisamylborohydride in inverse micelles, and then used liquid 30 chromatography to fractionate the resultant Au nanoparticles. They found that only nanoparticles smaller than 5.0 nm showed a blue fluorescence at 440 nm under an excitation of 230 nm. Their results suggest that ultra-small nanoclusters could exhibit strong fluorescence.

A breakthrough in preparing fluorescent Au nanoclusters was achieved by Zheng and co-workers. 94-96 They synthesized a series of Au₅, Au₈, Au₁₃, Au₂₃, and Au₃₁ nanoclusters using poly(amidoamine) (PAMAM) dendrimers as stabilizers. By adjusting the molar ratio between Au³⁺ and PAMAM from 1:1 to 40 1:15, they tuned the emission of these Au nanoclusters from the ultraviolet (UV) to the near infrared (NIR) range with a QY from 10% to 70%. The latter experiments, however, proved that PAMAM dendrimers made a contribution to the solution fluorescence. The authors also used dendrimers as ligands to 45 prepare fluorescent Ag nanoclusters. 89, 97 In addition to dendrimers, some other polymers such as multiarm star polyglycerol-block-poly(acrylic acid) and DHLA functionalized PEG were used to stabilize metal nanoclusters. 98, 99

Recently, we used multidentate thioether-terminated poly(methacrylic acid) (PTMP-PMAA) (Figure 7(a)) as ligand to successfully prepare water-soluble fluorescent Au nanoclusters through reduction of HAuCl₄ with NaBH₄. Due to the strong steric effect, this polymer ligand has also been used to prepare ultra-small magnetic iron oxide nanoparticles. By controlling the polymer concentration and molecular weight, we obtained a series of Au nanoclusters with emissions between 540 – 800 nm and QYs of 2.6 – 4.8%. In contrast to dendrimers, our

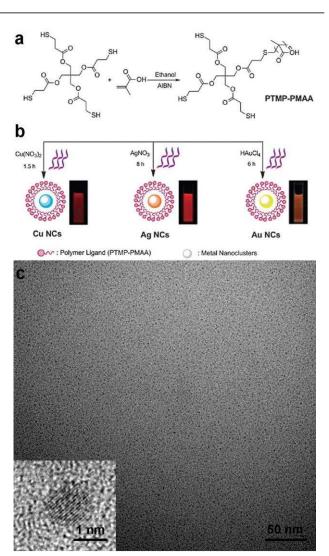


Figure 7. Schematic diagram of preparation of **(a)** polymer ligand PTMP-PMAA; **(b)** photoreductive synthesis of fluorescent Cu, Ag, and Au nanoclusters; **(c)** TEM image of Au nanoclusters. Reproduced from Ref. 100 and Ref. 105.

polymer ligands did not show fluorescence, and the observed fluorescence was only caused by the Au nanoclusters. The different emissions of Au nanoclusters are attributed to their different sizes. On the basis of this research, we prepared fluorescent Au, Ag, and Cu nanoclusters using photoreduction rather than chemical reduction (Figure 7(b)). Compared with conventional chemical reduction, photoreduction is clean and non-toxic as this method avoids the use of additional reducing agents. The QYs of the resultant Au, Ag, and Cu nanoclusters were 5.3%, 6.8%, and 2.2%, respectively. Using the photoreduction method, Shang *et al.* also prepared very highly fluorescent Ag nanoclusters (18.6% QY) in the presence of poly(methacrylic acid) (PMAA).

Compared with PMAA, our polymer ligands have a stronger steric hindrance effect. Figure 7(c) presents a typical transmission electron microscope (TEM) image of Au nanoclusters stabilized with PTMP-PMAA, clearly showing their ultra-small size (< 1.0 nm). In order to further investigate the polymer hindrance effect, we designed three types of tridentate thioether-terminated

polymer ligands, ¹⁰⁷ i.e. poly(methyl methacrylate) (PTMP-PMMA), poly(n-butyl methacrylate) (PTMP-PBMA), and poly(t-butyl methacrylate) (PTMP-PtBMA), which were used to synthesize fluorescent Au nanoclusters through the facile ⁵ photoreduction method. The resultant Au nanoclusters exhibited blue fluorescence instead of red fluorescence due to their small particle size. Their QYs were found to be 3.8%, 14.3%, and 20.1%, respectively, which increases with increasing polymer steric hindrance, i.e. PTMP-PMMA < PTMP-PBMA < PTMP-PBMA.

In addition to polymer ligands, small thiol molecules such as glutathione, tiopronin, MPA, DHLA, phenylethylthiolate, and thiolate α-cyclodextrin were also used to prepare fluorescent metal nanoclusters. ^{87, 108, 109} For example, Luo *et al.* used L-15 glutathione as ligands to prepare Au(0)@Au(I)-thiolate core-shell nanoclusters with a QY of 15%. ¹⁰⁹ They proposed that strong luminescence emission is attributed to aggregation-induced emission of Au(I)-thiolate complexes. The QYs of metal nanoclusters stabilized by small molecules are similar to those ²⁰ nanoclusters stabilized with polymer.

In order to improve the biocompatibility of fluorescent metal nanoclusters, several groups used biomolecules such as oligonucleotides, peptides, and proteins as stabilizers during preparation. 110-113 For example, the Dickson group took 25 advantage of the strong affinity of Ag⁺ to cytosine bases from single-stranded DNA, and prepared very small Ag nanoclusters using DNA as stabilizer. 110 In their later report, they used DNA microarrays for high-throughput analysis of 12-mer strands to identify optimized sequences for Ag encapsulation, and produced ₃₀ five distinct Ag emitters with QYs in the range of 16 – 34%. ¹¹¹ Compared with single-stranded DNA, proteins have abundant binding sites and offer better protection to metal nanoclusters. Xie et al. prepared Au₂₅ nanoclusters with a QY of 6.0% using bovine serum albumin (BSA) as the stabilizer and reducing 35 agent. 112 The reduction process was induced by adjusting the solution pH.

Similar to the QDs produced in living organisms, fluorescent metal nanoclusters can also be formed *in-situ* in cells. For example, Wang and co-workers found that fluorescent Au nanoclusters were spontaneously biosynthesized by cancer cells (human hepatocarcinoma cell line HepG2 and leukaemia cell line 562) rather than normal cells such as human embryo liver cells (L02) when the cells were incubated with chloroauric acid solution. 114 Au nanoclusters were formed by reduction of Au-45 precursor inside the cell cytoplasm and concentrated around their nucleoli. The selective formation of fluorescent Au nanoclusters by cancer cells can be exploited for *in vivo* self-bio-imaging of tumours.

Etching of large metal nanoparticles is an alternative approach to prepare fluorescent metal nanoclusters. The etching process can be performed by adding strong ligands or precursors into the nanoparticle solution. For example, Muhammed et al. synthesized fluorescent Au nanoclusters from MSA-stabilized Au nanoparticles by etching with excess glutathione. The etching process is pH-dependent and the obtained Au₈ and Au₂₅ nanoclusters have a QY of 0.015% and 0.19%, respectively. They also developed an interfacial etching process to prepare fluorescent Ag nanoclusters. First, they prepared MSA-

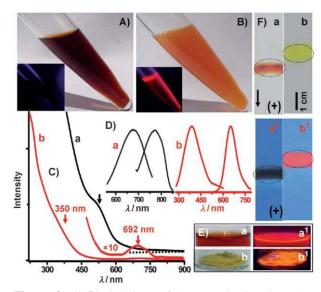


Figure 8. (**A-B**) Solutions of $Ag_{7,8}$ and alloyed Ag_7Au_6 nanoclusters; (**C-D**) their absorption and emission spectra; (**E**) alloyed Ag_7Au_6 nanoclusters in solution and in the solid state under visible and UV light; (**F**) comparison of the PAGE of $Ag_{7,8}$ and alloyed Ag_7Au_6 nanoclusters. Reproduced from Ref. 117.

stabilized Ag nanoparticles and then added them into an organic solvent (e.g. toluene, carbon tetrachloride, diethyl ether) containing excess MSA under magnetic stirring. A mixture of Ag₈ and Ag₇ nanoclusters with red and blue-green fluorescence was obtained. The QYs of the Ag₈ nanoclusters at room temperature and 273 K were calculated to be 0.3% and 9%, respectively. The authors used a similar approach to obtain alloyed Ag₇Au₆ nanoclusters (3.5% QY) by adding HAuCl₄ solution into the as-etched Ag nanocluster solution (Figure 8). 117

In addition to small molecules, multivalent polymers can also be used as etching agents. Duan *et al.* used multivalent polyethylenimine (PEI) to etch 8 nm Au nanoparticles, which were prepared by a two-phase approach and stabilized with dodecylamine. The resultant cluster solution surprisingly appeared to be in an oxidized electronic state with an emission at 505 nm. The emission was blue shifted to 445 nm with a QY of 75 10 - 20% after reduction with NaBH₄. ¹¹⁸

Similar to organic ligands, metal precursors can also induce the etching process. For example, Lin and co-workers extracted HAuCl₁ from aqueous solution didodecyldimethylammonium bromide (DDAB) toluene solution, 80 and then added the mixture into 5.6 nm Au solution to result in 3.2 nm particles. 119 After replaced DDAB with dihydrolipoic acid, these Au nanoparticles were further decreased to 1.6 nm and showed a red emission around 700 nm. Their fluorescence QY was 3.4% in methanol and 1.8% in water (pH = 9). Recently, 85 Yuan et al. developed a general etching approach to prepare fluorescent Au, Ag, Cu and Pt nanoclusters with a QY of 5.4%, 6.5%, 3.5% and 4.6%, respectively. 120 They started with glutathione-stabilized metal nanoparticles, and then transferred the metal nanoparticles into an organic phase by taking advantage 90 of the electrostatic interactions between negatively charged glutathione (carboxyl group) positively and cetyltrimethylammonium bromide (CTAB). The beauty of this

approach is that the resultant fluorescent metal nanoclusters can be shuttled back to the aqueous phase using hydrophobichydrophobic interactions upon addition of hydrophobic salts (e.g. tetramethylammonium decanoate) in chloroform.

Besides these methods, microwaves and ultrasound were also used to assist the synthesis of fluorescent metal nanoclusters in recent years. ^{121, 122} For example, Xu and Suslick adopted sonochemistry to prepare fluorescent Ag nanoclusters with a QY of 11% in the presence of PMAA. ¹²¹ Shang *et al.* synthesized fluorescent Au nanoclusters (2.9% QY) via a rapid microwave assisted method. ¹²² In all syntheses, ligands play a crucial role in obtaining these ultra-small fluorescent metal nanoclusters. Their ability to donate electrons drastically influences the fluorescence intensity, i.e. the stronger the electron donating capability is, the 15 higher the fluorescence intensity will be. ¹²³

3.2 Characterization and modification of fluorescent metal nanoclusters

Metal nanoclusters can be characterized by the techniques applied to nanomaterials and molecules. Compared with large 20 nanoparticles, metal nanoclusters have a smaller size and a "narrower" size distribution, so that size-selective precipitation is not suitable for their separation. They are usually fractionated by chromatography and electrophoresis techniques, which are usually applied to molecules. These separation methods include 25 high-performance liquid chromatography (HPLC), size exclusion chromatography (SEC), ion exchange chromatography (IEC), capillary electrophoresis, and polyacrylamide gel electrophoresis (PAGE). It is still very challenging to obtain monodisperse nanoclusters using these approaches. For example, Tsunoyama et 30 al. separated Au:SC_x nanoclusters into different fractions using gel permeation chromatography (GPC), 124 and then characterized them with laser-desorption ionization (LDI) mass spectroscopy. The results show that each fraction had a wide distribution of Au atoms although they were well separated with high resolution in 35 the GPC spectrum. Negishi and co-workers synthesized glutathione-protected Au nanoclusters and then fractionated them into 9 fractions, with the number of Au atoms ranging from 10 to 39 by PAGE analysis. 125 Among their Au nanoclusters, $Au_{25}(SG)_{18}$ is the most stable one.

TEM, and their molecular weight can be measured by mass spectroscopy. In principle, their crystal structures could be determined by X-ray diffraction (XRD). Metal nanoclusters are less ordered, however, and their powder XRD patterns are broad.

45 In comparison with metal complexes with defined molecular structure, it is very difficult to obtain single crystal clusters for structural characterization. So far, most structural investigations of metal nanoclusters are focused on "large" Au nanoclusters. 91. 108, 126-128 For example, Jadzinsky *et al.* determined the structure of a Au₁₀₂(*p*-mercaptobenzoic acid)₄₄ single crystal and found a core-shell structure, 127 in the which the Au₄₉ core is surrounded by two groups of Au atoms. Qian and co-workers characterized the crystal structure of Au₂₅(SR)₁₈ and Au₃₈(SR)₂₄ nanoclusters, and found a similar core-shell structure. 91 An Au₂₅(SR)₁₈ cluster 55 consists of an icosahedral Au₁₃ core and exterior 12 Au atoms in

the form of six -RS-Au-RS-Au-RS- motifs. 91, 108

The size of fractionated metal clusters can be characterized by

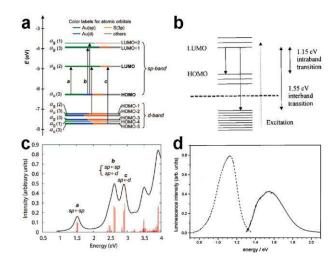


Figure 9. (a) Kohn-Sham orbital energy level diagram for a model compound Au₂₅(SH)₁₈; (b) Solid-state model for the origin of the two luminescence bands in (d); (c) theoretical absorption spectrum of Au₂₅(SH)₁₈; (d) two luminescence peaks observed in Au₂₅(SG)₁₆ clusters. Reproduced from Ref. 131 and Ref. 132.

The ultra-small size (limited atomic numbers) of metal nanoclusters makes it possible to predict their crystal structures through precise theoretical simulation. For example, Xiang *et al.* 60 developed a new genetic algorithm approach to search for the global lowest-energy structures of DMSA-stabilized Ag nanoclusters. 129 In combination with density functional theory (DFT), their genetic algorithm simulations show that the ground state of [Ag₇(DMSA)₄]⁻ has eight instead of four Ag–S bonds, 65 with a much lower energy than the structure based on the [Ag₇(SR)₄]⁻ cluster with a quasi-two-dimensional Ag₇ core. Their simulated X-ray diffraction pattern of the [Ag₇(DMSA)₄]⁻ cluster is in good agreement with the experimental results.

The optical properties of fluorescent metal nanoclusters can be characterized with UV-visible (UV-Vis) absorption and photoluminescence spectroscopy. As mentioned previously, metal nanoclusters have no SPR absorption, but they show molecular-like electronic transitions due to the quasi-continuous energy band structure and quantum confinement effects. Bakr et al. synthesized Ag nanoclusters through the reduction of Agprecursor in the presence of 4-fluorothiophenol, and investigated the evolution of their absorption from multiple bands into a single SPR band by heating the original nanocluster solution at 90 °C for different periods of time. Their results demonstrate the size dependence in UV-Vis absorptions of metal nanoclusters.

In order to demonstrate the origin of multiband absorption, Zhu and co-workers chose the Au₂₅(SR)₁₈ cluster as a model and simulated their absorption by performing time-dependent DFT calculations. Figure 9(a) shows the Kohn-Sham molecular orbitals, energies, and atomic orbital contributions in the cluster. The highest occupied molecular orbital (HOMO) and the lowest three lowest unoccupied molecular orbitals (LUMOs) are mainly composed of 6sp atomic orbitals of Au, and these orbitals constitute the sp-band. The HOMO-1 to HOMO-5 orbitals are constructed from the 5d atomic orbitals of Au and form the d-band. In addition, the s 3p orbitals make contributions to both sets of HOMO and LUMO orbitals. The multiband absorption of

metal nanoclusters suggests their multiple emission peaks and broad fluorescence spectra (Figure 9(b)). Figure 9(c) shows the simulated absorption spectrum. The multiple absorptions are attributed to the intraband (sp) HOMO \rightarrow LUMO transition, the sinterband transition $(d \rightarrow sp)$, or mixed $sp \rightarrow sp$ intraband and $d \rightarrow sp$ interband transitions. Figure 9(d) shows two fluorescence bands with the maxima at around 1.5 and 1.15 eV observed in $Au_{28}(GSH)_{16}$ nanoclusters by Link *et al.* These two bands are separated from a broad luminescence in the range of 2.0 - 0.8 eV, and are ascribed to the radiative interband recombination between the *sp* and *d* bands, and intraband transitions (*sp* bands) between the HOMO and LUMO.

Despite the good agreement between simulated data and experimental observations, the origin of metal fluorescence is not completely understood. Most reported atomically precise $\mathrm{Au}_n(\mathrm{SR})_m$ nanoclusters show very weak luminescence. Recently, Yu and co-workers identified that $\mathrm{Au}_{22}(\mathrm{SR})_{18}$ has two RS-[Au-SR]₃ and two RS-[Au-SR]₄ motifs that are interlocked and capped on a prolate Au8 core. ¹³³ These $\mathrm{Au}_{22}(\mathrm{SR})_{18}$ nanoclusters exhibited an emission at ~665 nm with a QY of ~8%. Their results show that the luminescence of these core-shell nanoclusters was generated by the aggregation-induced emission of Au(I)-thiolate complexes on the nanocluster surface.

The fluorescence of metal nanoclusters is very sensitive to the 25 cluster size, surface ligands, solvents, etc., so it is thus necessary to modify them in order to maintain their bright fluorescence in addition to improving their stability and biocompatibility. There are few reports, however, on the post-modification of fluorescent metal nanoclusters in comparison with large nanoparticles, due to 30 their tiny size and sensitivity to external conditions. Lin and coworkers prepared DHLA-protected fluorescent Au nanoclusters by etching large nanoparticles and replacing surface ligands. 119 They took advantage of carboxylic acid groups from DHLA to conjugate PEG-NH₂ or biotin-PEG-NH₂ with Au nanoclusters. 35 The gel electrophoresis and the cell labelling indicate the successful conjugation. Samanta et al. prepared fluorescent Au nanoclusters using a novel quaternary ammonium as the ligand, and then coated them with silica. 134 Similar to fluorescent QDs, surface modification can lead to the fluorescence quenching of 40 metal nanoclusters. It is still a great challenge to obtain robust fluorescent metal nanoclusters through surface modification.

3.3 Application of fluorescent metal nanoclusters in bioimaging

Similar to other fluorophores, fluorescent metal nanoclusters have also been tested for *in vitro* and *in vivo* bioimaging. In the early reports, Zheng *et al.* prepared fluorescent Au, Ag nanoclusters in the presence of dendrimers, DNA, and proteins, and used them as labels for cell imaging. ^{89, 94-97} Baskov *et al.* prepared fluorescent Ag nanoclusters in the presence of thioflavin T with remarkable fluorescent properties, ¹³⁵ and then used them to label amyloid fibrils produced from recombinant mammalian prion proteins and non-prion proteins. The labelled amyloid fibrils exhibited a time-dependent increase in fluorescence with no photobleaching after 24-h illumination, while those stained with thioflavin T showed a rapid decay in fluorescence. Their results demonstrate the higher stability of Ag nanoclusters than that of organic fluorophore.

Recently, we prepared fluorescent Au nanoclusters stabilized with PTMP-PMAA, and then compared them with CdTe ODs in

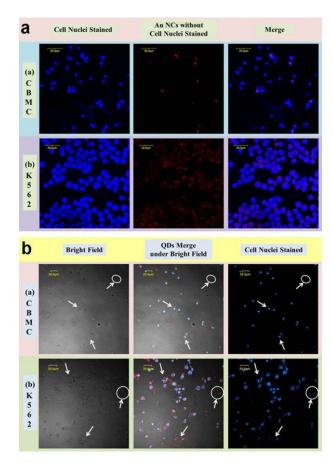


Figure 10. Comparison of cell labeling by using fluorescent Au nanoclusters and CdTe QDs. Reproduced from Ref. 100.

labelling suspended and adherent hematopoietic relatively normal cord blood mononuclear (CBMC) cells and cancer K562 cells 60 (Figure 10). 100 The results show that the cancer cells took up more Au nanoclusters than the normal cells, even though they were from the same hematopoietic system. There was no difference, however, in the uptake of CdTe QDs between the two kinds of cells. The selective uptake of Au nanoclusters by cancer 65 cells could be attributed to the unique properties of Au nanoclusters or the nature of the cells. In addition, CdTe QDs destroyed the nuclei of some cells. We also compared the cytotoxicity of Au nanoclusters with that of CdTe QDs through MTT and apoptosis assay. The results show that our fluorescent 70 Au nanoclusters had lower toxicity than QDs, and did not induce acute toxicity. These advantages make them very attractive in selective bio-labelling of cancer cells. Retnakumari et al. conjugated folic acid with BSA-stabilized Au nanoclusters and then used them for targeted imaging. 136 The receptor-targeted 75 cancer detection was demonstrated on FR+ve oral squamous cell carcinoma (KB) and breast adenocarcinoma MCF-7 cells, where the FA-conjugated Au₂₅ clusters were found to be internalized in significantly higher concentrations compared to the negative control cell lines.¹³⁶ Apart from routine utilization of cell 80 labelling, fluorescent metal nanoclusters can be used as intracellular sensors. For example, Shang and co-workers demonstrated the use of Au nanoclusters for intracelluar thermometry by taking advantage of the temperature sensitivity of their fluorescence lifetime and emission intensity (Figure

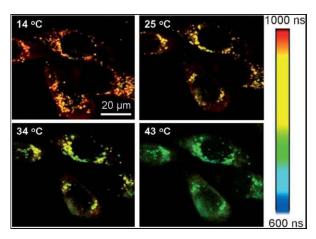


Figure 11. FLIM images of HeLa cells with internalized Au nanoclusters at four different temperatures. Reproduced from Ref. 15.

11). 15 Using fluorescence lifetime imaging microscopy (FLIM), they observed the considerable variation of fluorescence lifetime of nanoclusters internalized in HeLa cells with the temperature increasing from 15 to 45 $^{\circ}$ C.

In addition to the above *in vitro* cell labelling and imaging, Wu *et al.* investigated *in vivo* imaging through the tail vein administration of near infrared (NIR) fluorescent Au nanoclusters in live mice, ¹³⁷ and found that the uptake of NIR Au nanoclusters by the reticuloendothelial system was relatively low in comparison with other nanoparticles due to their ultra-small hydrodynamic size (~2.7 nm). They then used MDA-MB-453 and xenografted HeLa tumour cells as models to do *in vivo* and *ex vivo* imaging studies, and found that the ultra-small Au nanoclusters were highly accumulated in the tumour areas due to the enhanced permeability and retention (EPR) effect. ¹³⁷

Zhou and co-workers studied the renal clearance of 2 nm glutathione (GSH)-coated fluorescent Au nanoclusters (Figure 12), ¹³⁸ and found that only ~4% of the particles were accumulated in the liver, while more than 50% of the particles were found in ²⁰ urine within 24 h after intravenous injection, which is comparable to the QDs with the best renal clearance efficiency.⁸³ They also used computed tomography (CT) to visualize real time accumulation of luminescent GS-AuNPs in the bladder, and demonstrated that fluorescent Au nanoclusters can serve as 25 contrast agents for CT imaging (Figure 12). Recently, they compared GSH-coated fluorescent Au nanoclusters (2.5 nm) with small dye molecules IRDye 800CW, 139 and found that they both have similar physiological stability and renal clearance, but Au nanoclusters exhibited a much longer tumour retention time and 30 faster normal tissue clearance (Figure 12). These merits enabled the Au nanoclusters to detect the tumour more rapidly than the dye molecules without severe accumulation in reticuloendothelial system organs. 139

Besides the above *in vivo* passive targeting, fluorescent metal nanoclusters can be tagged with bioactive molecules for targeting, imaging, and therapy. For example, Liu *et al.* synthesized fluorescent Au nanoclusters $(0.92 \pm 0.03 \text{ nm})$ using insulin as a template. The resulting Au-insulin nanoclusters retain the insulin bioactivity and biocompatibility, and have been used to regulate the *in vivo* glucose level in Wistar rats. The results show

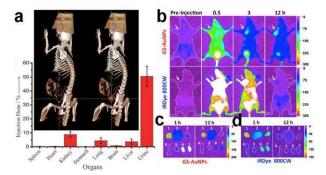


Figure 12. (a) Biodistibution of 2-nm GS-Au nanoclusters. The inset shows CT images of a live mouse before and 30 min after injection of Au nanoclusters; (b-d) comparison of biodistribution of GS-Au nanoclusters and IRDye 800CW. Reproduced from Ref. 138 and Ref. 139.

that an injection of insulin-Au nanoclusters into the rats tended to reduce the blood glucose in a similar way to commercial insulin. Fluorescent insulin-Au nanoclusters can also be used as contrast agents for CT imaging. These studies indicate that ultra-small fluorescent Au nanoclusters could simultaneously serve as very promising contrast agents for *in vivo* fluorescence imaging and CT imaging.

In summary, fluorescent metal nanoclusters as emerging fluorophores have attracted considerable attention due to their tuneable emissions, ultra-small size, fast renal clearance, and low toxicity. There are a few obstacles to be overcome, however, including (i) low fluorescence QY, which is usually about ~10% and less than that of QDs and many organic dyes; (ii) polydispersity in size and components, which makes it very difficult to fundamentally study their novel properties and mechanisms; (iii) difficulty in modifying their surface to introduce other functions due to their tiny size and lower stability; (iv) complicated interactions with biological environments.

60 4 Fluorescent carbon nanomaterials

4.1 Fluorescent carbon dots

Fluorescent carbon dots are also used as alternatives to QDs for bioimaging, ^{17, 140} because they not only exhibit several favourable attributes of traditional semiconductor-based QDs (namely, size- and wavelength-dependent emission, resistance to photobleaching, ease of bioconjugation), but also show chemical inertness, low toxicity, and biocompatibility. Fluorescent carbon dots were accidently discovered in 2004 during the purification of single-wall carbon nanotubes (SWCNTs) fabricated by the arc- discharge approach. ¹⁴¹ Two new classes of nanomaterials were isolated from the crude soot. One was short, tubular carbon, and the other a mixture of fluorescent nanoparticles derived from the SWCNTs.

In 2006, Sun *et al.* obtained 5-nm non-fluorescent carbon dots via laser ablation of a carbon target, and then modified them with PEG to get fluorescent carbon dots with a fluorescence QY of 4% – 10%. ¹⁴² The photoluminescence of these carbon dots was broad and strongly dependent on the excitation wavelength, which could be attributed to the different sizes in the sample and different emission sites on the passivated particle surfaces. After fractionation with gel column chromatography, most of the

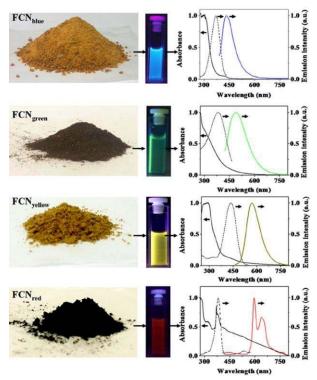


Figure 13. Digital images of solid fluorescent carbon dots, aqueous solutions, and their absorption, excitation and emission spectra. Reproduced from Ref. 144.

fluorescent fractions could achieve emission yields close to 60 %. 143 Interestingly, their optical properties resemble band-gap transitions, which are found in nanoscale semiconductors, suggesting that carbon dots have essentially semiconductor-like characteristics. Recently Bhunia *et al.* prepared hydrophobic and hydrophilic carbon dots with tuneable size and visible emissions, 144 by dehydrating carbohydrate in octadecene in the presence of octadecylamine, or in concentrated sulphuric acid (Figure 13). Their method produced gram-scale fluorescent carbon dots with a QY of 6 – 30%. Zhu and co-workers also reported a rapid and high-output hydrothermal approach to prepare polymer-like carbon dots with QYs as high as 80 %. 145

In addition to solid fluorescent carbon dots, there are some reports on hollow fluorescent carbon dots. 146, 147 For example, 15 Fang et al. simply mixed acetic acid, water and diphosphorus pentoxide to obtain cross-linked hollow fluorescent carbon nanoparticles. By reducing the release of heat, they also obtained solid fluorescent nanoparticles. So far, many approaches, such as arc-discharge, laser ablation, electrochemical oxidation, 20 combustion/pyrolysis, and hydrothermal and microwave methods, have been developed to prepare solid and hollow fluorescent carbon dots. 148 The preparation is inexpensive on a large scale without the need for stringent, intricate, tedious, costly, or inefficient steps. 149 The recent advances in the synthesis 25 and characterization of fluorescent carbon dots have been reviewed. 17, 148, 149

The first study of fluorescent carbon dots in bioimaging was reported by the Sun group in 2007. The authors used poly-(propionylethylenimine-*co*-ethylenimine) (PPEI-EI, with EI fraction ~20%) to modify the carbon dots, and then applied them to label human breast cancer MCF-7 cells. These labelled cells

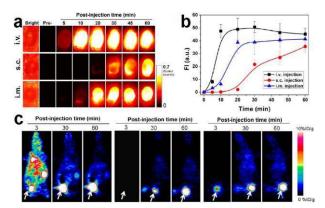


Figure 14. (a) NIR images of mouse bladders acquired before and after injection of carbon dots through intravenous injection, subcutaneous injection, and intramuscular injection; **(b)** quantification of the ZW800 fluorescence signal in (a); **(c)** representative coronal images from 1 h dynamic positron emission tomography (PET) imaging. Reproduced from Ref. 152.

exhibited bright fluorescence in both cell membrane and cytoplasm regions under an excitation of 800-nm laser pulses. The results demonstrate that carbon dots exhibit strong 35 luminescence with two-photon excitation in the near-infrared, and moreover, large two-photon absorption cross-sections, those of available high-performance comparable to semiconductor QDs. 150 The authors further demonstrated the in vivo imaging of fluorescent carbon dots. 151 They compared the 40 imaging capability of carbon dots and ZnS-doped carbon dots, and found that the later dots emitted more strongly than the former dots both in solution and in mice. The fluorescence from the bladder area was observed, and 3 h after injection, the fluorescence could be detected in the urine, but it completely faded 24 h after injection. They analysed the biodistribution of carbon dots and found that the carbon dots accumulated in the kidney and, to a small extent, in the liver. 151 This is attributed to the surface PEG, which likely reduces the protein adsorption.

Recently Huang and co-workers investigated the effects of injection routes on the biodistribution, clearance, and tumour uptake of carbon dots (Figure 14). They prepared fluorescent carbon dots through a laser ablation approach, and then functionalized carbon dots with the NIR dye ZW-800 and the isotope 64Cu. They injected the conjugates into mice in three 35 different manners, i.e. intravenous, intramuscular, and subcutaneous injection. The results show that the carbon dots were efficiently and rapidly excreted from body after injection, and the clearance rate of carbon dots decreased when the administration was varied from intravenous, to intramuscular, and 60 then to subcutaneous injection (Figure 14). Different injection routes also showed different blood clearance patterns and different tumour uptake of carbon dots.

4.2 Fluorescent graphene quantum dots

It should be noted that fluorescent graphene quantum dots (GQDs), the analogues of carbon dots, have also attracted considerable attention. Similar to carbon dots, GQDs can be prepared by top-down and bottom-up approaches, and their fluorescence can be enhanced via surface modification. The top-down methods usually refer to cutting larger size carbon

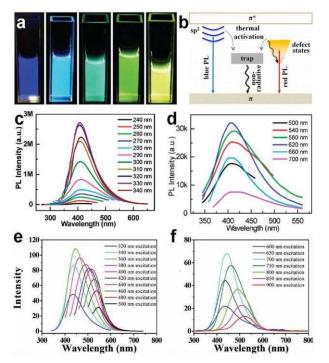


Figure 15. (a) Image of GQD solution under UV-light; (b) schematic emissions in GQDs; (c-d) excitation-independent down-conversion and up-conversion spectra of GQDs; (e-f) excitation-dependent down-conversion and up-conversion spectra of GQDs. Reproduced from Refs. 155, 157, 160, and 161.

materials such as carbon nanotubes, graphene or graphene oxide sheets, and carbon fibres into small GQDs, through strong acid oxidation, hydrothermal or solvothermal treatment, or microwave and sonication treatment. 154 For example, Zhu et al. dispersed 5 graphene oxide in dimethyl formamide (DMF) under sonication, and then transferred the suspension into Teflon autoclaves and treated them at high temperature for a few hours to get GQDs with a OY of 11%. 155, 156 Tetsuka and co-workers used the hydrothermal approach to treat graphene oxide sheets in ammonia 10 solution to get GQDs with a QY between ~19 - 29%. 157 The emission of GQDs can be tuned by controlling the hydrothermal temperature (Figure 15(a)), and the QYs can be further enhanced to ~46% after modification with PEG. Wu et al. used a one-step pyrolysis of a natural amino acid (i.e. glutamic acid) to prepare 15 fluorescent GODs with a OY of 54.5%. 158 Recently, Dong and co-workers used L-cysteine as precursor to prepare S,N-co-doped GQDs with a QY up to 73%, 159 which is the highest value reported so far.

The preparation process significantly influences the optical properties of GQDs. There are two types of emissions in GQDs, i.e. intrinsic state emission and defect state emissions (Figure 15(b)). The competition between these two states could be changed during preparation or post surface modification. For example, Zhuo and colleagues oxidized graphene in concentrated 25 H₂SO₄ and HNO₃, and then sonicated the mixture and calcinated it at 350 °C to remove acid. The resultant fluorescent GQDs did not exhibit excitation-dependent fluorescence [Figure 15(c-d)]. However, Zhu *et al.* prepared green fluorescent GQDs through the hydrothermal approach.

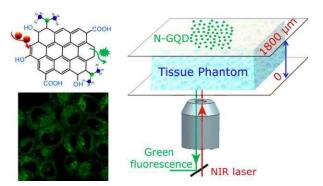


Figure 16. Nitrogen-doped GQDs for cellular and deeptissue imaging. Reproduced from Ref. 164.

30 was changed into blue after the GQDs were modified with alkylamines or reduced with NaBH₄ (referred to as m-GQDs and r-GQDs respectively), while the particle size was similar. The fluorescence shift was attributed to the suppression of nonradiative processes and to the enhanced integrity of the π 35 conjugated system. These three types of GQDs exhibited strong excitation-dependent down-conversion and up-conversion emissions, demonstrated by that of the m-GQDs in Figure 15(ef), 155 which is in contrast to Zhuo's report [Figure 15(c-d)]. In addition to the preparation, the optical properties of GODs are 40 also influenced by the solution pH, solvent, and concentration. 154 Recently, Xu et al. studied the fluorescence of GQDs on a substrate at the single particle level. 160 All the GQDs investigated had the same spectral lineshapes and peak positions, despite notable differences in particle size and the number of layers. 45 GQDs with more layers were brighter than those with fewer layers, but were associated with shorter fluorescence lifetimes.

Although there are some debates on the fluorescence mechanisms of GQDs, their unique properties afford many applications in cellular and deep-tissue imaging. Sun and coworkers demonstrated the first bioapplication of nanographene oxide (NGO), 162 i.e., single-layer graphene oxide sheets a few nanometers in lateral width. The PEGylated NGO sheets used were soluble in buffers and serum without agglomeration, and showed photoluminescence in the visible and infrared regions.

These NGO sheets had low background photoluminescence in the near-infrared (NIR) window. In addition, simple physisorption through π-stacking was used to load the anticancer drug doxorubicin onto NGO functionalized with antibody for selective killing of cancer cells *in vitro*. 162

Compared with fluorescent carbon dots, GQDs can be used for two-photon or multi-photon luminescence imaging. 163, 164 Qian *et al.* used PEGylated graphene oxide nanoparticles to label HeLa cells, 163 and observed that graphene oxide nanoparticles were mainly localized in the mitochondria, endoplasmic reticulum, 65 Golgi apparatus, and lysosomes of HeLa cells with a two-photon scanning microscope. They intravenously injected graphene oxide nanoparticles into mouse bodies from the tail vein, and observed their flow, distribution, and clearance in the blood vessels, utilizing a deep-penetrating two-photon imaging 170 technique. These nanoparticles were also injected into the brains of gene transfected mice, and the *in vivo* two-photon luminescence imaging results showed that graphene oxide nanoparticles were located at 300 μm depth in the brain,

demonstrating the advantage of QGDs for deep imaging in tissues. Recently, Liu *et al.* prepared nitrogen-doped GQDs as efficient two-photon fluorescent probes. ¹⁶⁴ These N-GQDs exhibited the highest two-photon absorption cross-section (up to 5 48000 Göppert-Mayer units) among the carbon-based materials. They also demonstrated a large imaging depth of 1800 µm by a study of penetration depth in tissue phantom (Figure 16).

In summary, surface-modified fluorescent carbon nanomaterials (carbon dots and GQDs) have small size, distinctive photoluminescence properties, low toxicity, and low cost. These advantages offer them great potential for optical imaging and biomedical applications, as they might gradually replace conventional semiconductor QDs in these aspects.

15 5 Ultra-small up-conversion nanocrystals

Compared with previously mentioned fluorescent nanomaterials, up-conversion nanostructures, especially lanthanide-doped nanocrystals, have distinct advantages in fluorescence bioimaging, such as low autofluorescence background, large anti
Stokes shifts, sharp emission bandwidth, high resistance to photobleaching, and high penetration depth and temporal resolution, 165-171 In addition, they can be used for multimodal bioimaging and therapy (Figure 17). More bioapplications of up-conversion nanoparticles can be found in recent reviews. 165, 169-171

However, they usually have a larger size in comparison with those nanoprobes described previously (i.e. QDs, metal nanoclusters, carbon dots, and GQDs). There are few reports on ultra-small up-conversion nanoparticles, especially those below 5 nm. ¹⁷²⁻¹⁷⁷ Herein we mainly introduce the fundamentals of up-30 conversion nanoparticles and the progress in preparation and imaging application of ultra-small nanoparticles.

For up-conversion nanocrystals, their emission process involves the sequential absorption of two or more photons, which is fundamentally different from the multi-photon process, where 35 the absorption of photons takes place simultaneously. There are three types of up-conversion mechanisms, i.e. excited state absorption (ESA), energy transfer up-conversion (ETU), and photon avalanche. 166 The up-conversion nanocrystals usually consist of activators, sensitizers, and the host matrix [Figure 40 17(a)]. The activators should have more excited energy levels, and the energy difference between each excited level and the ground level should be close enough to facilitate photon absorption and energy transfer in the up-conversion process. Lanthanide ions such as Er³⁺, Tm³⁺, and Ho³⁺ have such ladder-45 like energy levels and are usually selected as activators. In order to improve the luminescence efficiency, sensitizers are introduced. Yb3+ is usually chosen as sensitizer because it has only one excited energy level (${}^{2}F_{5/2}$), and the transition between the ground level (${}^{2}F_{7/2}$) and excited level is strongly resonant with 50 many f-f transitions of lanthanide ions. The concentration of activators, and the molar ratio between activators and sensitizers is usually kept low to avoid the quenching effect. 166 Zhao et al., however, showed that up-conversion luminescence can be significantly enhanced by using much higher activator 55 concentrations (e.g. 8 mol% Tm³⁺ in NaYF₄) under relatively high-irradiance excitation. 178 The authors attributed the high brightness to a combination of high excitation intensity, increased

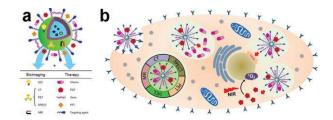


Figure 17. (a) Schematic structure of multifunctional upconversion nanoparticles; and (b) their potential applications in bioimaging and therapy. Reproduced from Ref. 165.

activator concentration, and accelerated sensitizer-activator energy transfer rate arising from the decreased average minimum distance between adjacent lanthanide ions. The high brightness makes it possible to remotely track a single nanocrystal with a microstructured optical-fibre dip sensor. 178

Ideal host materials should have low lattice phonon energy and the minimum lattice mismatch with dopants (activators and 65 sensitizers). Rare-earth fluorides are generally chosen as host materials, as rare-earth ions have similar ionic size and chemical properties to lanthanide ions, and their fluorides exhibit low phonon energy and high chemical stability. 166 In particular, NaGdF₄ is extensively used as it can serve as a positive contrast 70 agent for magnetic resonance imaging (MRI). Johnson et al. prepared four different sizes of β-NaGdF₄ nanoparticles between 2.5 nm and 8.0 nm. ¹⁷² They found that the longitudinal relaxivity of nanoparticles increased from 3.0 mM⁻¹s⁻¹ to 7.2 mM⁻¹s⁻¹ with decreasing particle size from 8.0 nm to 2.5 nm. The authors 75 doped Yb³⁺ and Tm³⁺ into β-NaGdF₄ to form 3.5 nm particles, which exhibited an emission at 800 nm under the excitation of a 980-nm laser.¹⁷² Their results highlight the importance of preparation of ultra-small nanoparticles in order to achieve large

The fluorescence of up-conversion nanoparticles can be engineered through modulation of activators, sensitizers, host materials, and their crystal phase, particle size, and surface coating. Hasse and co-workers demonstrated the first example of multicolour emission of Yb³⁺/Er³⁺, and Yb³⁺/Tm³⁺ co-doped α -85 NaYF₄ colloidal solution. 179 In 2008, Wang et al. developed a general and versatile approach to fine-tune the multicolour emissions over a broad range with single wavelength excitation. 180 By introducing Gd3+ during preparation, the authors simultaneously controlled the crystal phase, particle size, and 90 optical properties of the resultant nanocrystals. 181 Recently, a core-shell structure with a set of lanthanide ions incorporated into separated layers was designed. The core-shell structure can minimize the deleterious effects of cross-relaxation. The bright up-conversion emission was achieved through Gd 3+ mediated 95 energy migration without long-lived intermediate energy states. 182, 183

In up-conversion nanoparticles, minimizing the depletion of excitation energy is the key to tuning their luminescence. The excitation energy can randomly migrate from an atom to its neighbouring atoms that are isotropically distributed in a 3D structured crystal sublattice (type I in Figure 18). This energy can also migrate in a crystal with a 2D layer structure (type II), or in a crystal featuring a 1D atomic chain structure (type III). 184

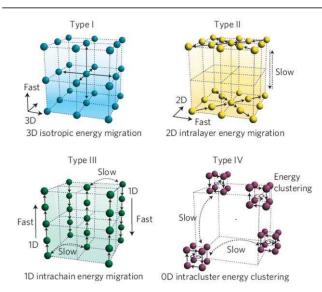


Figure 18. Schematic illustration of the topological energy migration pathways in different types of crystal sublattice. Reproduced from Ref. 184.

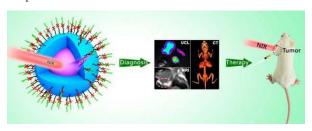


Figure 19. Multifunctional upconversion nanoparticles for diagnosis and treatment of cancer through imaging-guided therapy. Reproduced from Ref. 186.

Recently, Wang *et al.* proposed that migration of the excitation energy can be effectively minimized through use of a type IV (Figure 18) lattice containing arrays of isolated atomic clusters. ¹⁸⁴ This allows to minimize the concentration quenching of the luminescence, and generates an unusual four-photon-promoted violet up-conversion emission from KYb₂F₇:Er (2 mol%) with an intensity more than eight times higher than that previously reported. ¹⁸⁴ The approach of enhancing up-conversion through energy clustering at the sublattice level may provide new opportunities to engineer up-conversion nanoparticles for diverse applications.

The good understanding of the energy migration, luminescence mechanism, and the recent advances in wet chemistry have enabled the fine-tuning of particle size (even in the small size 15 range), crystal structure, surface functionalities, and optical properties of up-conversion nanocrystals for bioimaging, drug delivery, and sensing. 165, 167, 169 Their fluorescence has been applied to image cells and small animals. 185 As mentioned previously, Gd-based up-conversion nanoparticles are particularly 20 interesting as they can serve as fluorescent nanoprobes and contrast agents of MRI simultaneously. Recently, a multifunctional drug delivery system combining up-conversion luminescence/magnetic resonance/computer tomography trimodality imaging and NIR-activated platinum pro-drug 25 delivery has been developed by Dai and co-workers (Figure 19).186 Organic-soluble core-shell

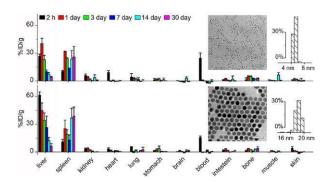


Figure 20. Biodistribution of 5.1nm (NaGdF4) and 18.5nm (NaGdF4:Yb,Er) nanoparticles in different organs and tissues of mice. Reproduced from Ref. 189.

NaYF₄:Yb³⁺/Tm³⁺@NaGdF₄:Yb³⁺ nanoparticles were first prepared by complex thermal decomposition method, followed by surface modification and conjugation with the trans-platinum (IV) pro-drug. The up-conversion nanoparticles can not only deliver the platinum (IV) pro-drugs into the cells effectively, but convert near-infrared light into UV to activate pro-drug as well. Meanwhile, they can further serve as contrast agents for multimodality imaging to guide cancer treatment. The pro-drug-conjugated nanoparticles under near-infrared irradiation led to better inhibition of tumor growth than that under direct UV irradiation in the mouse test. Such multifunctional up-conversion nanoparticles have been a subject of intensive research due to their potential in disease diagnosis and treatment. Such as the properties of the potential in disease diagnosis and treatment. Such as the properties of the pr

For in-vivo bioapplications, one of the major issues for upconversion nanocrystals is the fate of nanoparticles and potential toxicity of lanthanide ions. 187, 188 Liu et al. prepared 5.1 nm NaGdF₄ and 18.5 nm NaGdF₄:Yb/Er nanoparticles with the same 45 surface modification and investigated their biodistributions in different organs and tissues of mice (Figure 20). 189 The accumulation of both types of nanoparticles in liver decreased with the circulation time. In contrast, their accumulation in spleen increased with the circulation time. This suggests that both of 50 these nanoparticles may be eliminated through the biliary pathway. Analysis of urine collected at different time points indicates that renal clearance was one of the major elimination pathways for 5.1-nm particles, but not for 18.5-nm particles. Further analysis on faces by TEM shows that these particles do 55 not change in shape and size, suggesting the high stability of upconversion nanoparticles in vivo. 189

Although up-conversion nanoparticles do not show acute toxicity at the cell or animal level, it is necessary to investigate their long-term toxicity. Another drawback of up-conversion nanocrystals is the low quantum yield (usually less than 1%) in comparison with other fluorescent agents. 165 Preparation of highly efficient up-conversion nanocrystals remains a great challenge.

65 6 Fluorescent silicon nanoparticles

Fluorescent silicon nanoparticles (Si NPs) have also attracted considerable attention in bioapplications due to their excellent biocompatibility as silicon naturally exists in human body as a

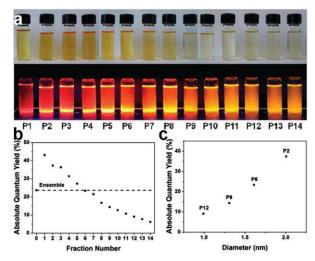


Figure 21. (a) Silicon nanoparticle fractions under ambient light and under photoexcitation at 365 nm; (b-c) size-dependent absolute QYs of Si nanoparticles. Reproduced from Ref. 191.

trace element. 18, 20, 190 More importantly, they have tunable fluorescence from visible to near-infrared window. Compared with other semiconducting QDs, the preparation of high-quality water-soluble and biocompatible Si QDs is devious and 5 laborious. Colloidal Si NPs are conventionally prepared by etching annealed SiO_x with HF, plasma approach, electrochemical method, laser ablation, reduction of SiCl₄, and solvothermal reaction of sodium silicide with NH₄Br. 18, 190-193 These Si NPs are usually functionalized with hydrophobic ligands 10 such as styrene, alkyl, and octene. They are photochemically stable in non-polar solvents up to 1 year. 192 For example, hightemperature thermal processing of the sol-gel precursor derived from trichlorosilane (HSiCl₃) produced Si NPs embedded within the SiO₂ matrix. 191, 194 Si NPs were released after etching the SiO₂ 15 matrix with HF, and then passivated with allylbenzene through the thermally initiated hydrosilylation reaction. The resultant colloidal Si NPs were fractionated by size selective precipitation to obtain monodisperse nanoparticles, which showed strong quantum confinement effects and size-dependent absolute QYs 20 (Figure 21). 191 The absolute QYs increased with particle size up to 43%.

During preparation, Si NPs can be chemically doped to introduce other functions. $^{195,\ 196}$ Paramagnetic fluorescent Si NPs were prepared by solvothermal decomposition of Mn-doped sodium silicide. 195 The resultant Mn-doped Si NPs showed a longitudinal relaxivity (r_l) of $25.50 \pm 1.44 \, \mathrm{mM^{-1}s^{-1}}$ and a transverse relaxivity (r_2) of $89.01 \pm 3.26 \, \mathrm{mM^{-1}s^{-1}}$ under a magnetic field of 1.4 T at 37 °C. Similarly, Fe-doped Si NPs were prepared and exploited as bimodal imaging agents. 196 The use of reactive sodium silicide makes their control preparation difficult and could limit their broad applications, and thus development of novel preparation approaches is necessary.

Similar to carbon dots and GQDs, Si NPs produced from different methods seem identical, but their optical properties are dramatically different. For example, the Si NPs prepared with high-temperature method routinely exhibit photoluminescence agreeing with the effective mass approximation (EMA), while those prepared via solution methods exhibit blue emission

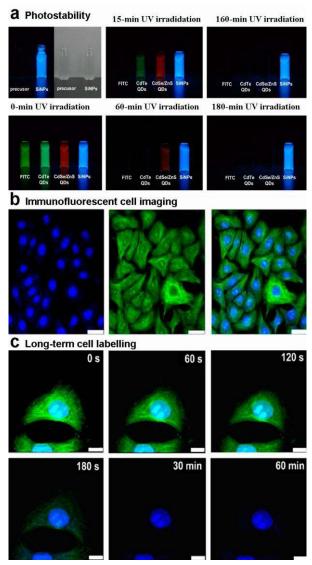


Figure 22. (a) Photostability of Si NPs in comparison with FITC, CdTe QDs and CdSe/ZnS QDs; (b) cell nuclei are labeled by Si NPs (Left), microtubules are labeled by FITC (middle); and superposition of the two fluorescence images (right); (c) time-dependent stability comparison of fluorescence signals of Hela cells labeled by Si NPs (blue) and FITC (green). Reproduced from Ref. 204.

somewhat independent of particle size. ^{197, 198} Recently, Dasog *et al.* prepared Si NPs using three most widely cited procedures (i.e. etching of annealed SiO_x, reduction of SiCl₄, and solvothermal reaction of sodium silicide with NH₄Br), ¹⁹⁷ and found their conversion of red-fluorescence to blue emission. Their findings suggest that the presence of trace nitrogen and oxygen even at the ppm level in Si NPs gives rise to the blue emission, and support the hypothesis that the nitrogen defect or impurity site contributes to the blue emission. ¹⁹⁷

In order to apply Si NPs to bioimaging, tremendous efforts have been made to prepare water-soluble and biocompatible Si 50 NPs through simple and efficient methods. 199-205 For example, Si NPs with excellent aqueous dispersibility, robust photo- and pH-stability, strong fluorescence (~15%), and favorable size (~4 nm) are facilely and rapidly prepared from Si nanowires and glutaric

acid in a short reaction time (e.g., 15 min) by He and coworkers. 202 These Si NPs are particularly suitable for long-term and real-time cellular imaging due to their higher photostability than II-VI QDs and dyes (e.g. CdTe QDs and FITC). Distinctive 5 red fluorescence of Si NPs can be retained throughout 240-min irradiation. In contrast, the green fluorescence of FITC rapidly diminishes in 3 min due to severe photo bleaching, and the red signals of CdTe QDs nearly vanishes after 25-min irradiation. The MTT assays showed negligible cellular toxicity to HeLa 10 cells, demonstrating the excellent biocompatibility of Si NPs. Furthering their research, the authors aminopropyl)trimethoxylsiliane as precursor and prepared 2.2 nm Si NPs by the similar method.²⁰⁴ The obtained Si NPs exhibited strong green fluorescence with a OY of 20-25%, 15 biocompatibility, and robust photo- and pH-stability. As shown in Figure 22, FITC, CdTe and CdSe/ZnS QDs, and Si NPs exhibited distinct fluorescence behaviors during initial UV irradiation.²⁰⁴ The fluorescent signals of the former three samples were gradually reduced with increasing irradiation time. The 20 fluorescence of FITC was completely quenched within 15 min irradiation. In contrast, the Si NPs preserved stable and bright fluorescence during long-time (e.g., 180 min) irradiation under the same conditions. Figure 22c also shows that Si NPs-labelled nuclei (blue) and the FITC-labelled cellular microtubules (green) 25 were intense and clearly spectrally resolved, respectively. The Si NP-labelled nuclei retained stable fluorescence during observation for 60 min, however the fluorescence from FITC labels drastically decreased in 3 min due to severe photo bleaching.

In addition to in vitro labelling cells, Si NPs can also be used in multiple cancer-related in vivo applications, including tumor vasculature targeting, sentinel lymph node mapping, and multicolor NIR imaging in live mice. 206 Erogbogbo and coworkers demonstrated that Si NPs can overcome dispersibility 35 and functionalization challenges for in vivo imaging through surface functionalization, PEGylated micelle encapsulation, and bioconjugation process, which produced bright, targeted nanospheres with stable luminescence and long (>40 h) tumor accumulation time in vivo. Recently, the biodistribution and 40 toxicity of Si NPs in mice and monkeys have been assessed (Figure 23).²⁰⁷ The top images in Figure 23 show the biodistribution of Si NPs in mice, the fluorescence image of frozen tissue sections, and the confocal images, which clearly reveals particles localized in the liver, spleen, and kidneys after 45 injection. The ICP-MS data show notable increase of silicon levels in the liver, spleen, lung, kidneys, and lymph. The concentration of silicon in the lymph and kidneys declined over the 14-week time period, while the liver and spleen retained a significant fraction of the silicon injected, even after 14 weeks.²⁰⁷ 50 There is no evidence of the biodegradability of silicon NPs. The bottom images in Figure 23 display the histological images of the brain, cerebellum, atrium, ventricle, heart muscle, lung, kidney, liver, spleen, renal tubule, intestine, lymph nodes, and skin of the rhesus macaques.²⁰⁷ There was no sign of nanoparticle-induced 55 changes in these organs and tissues. This research indicates neither mice nor monkeys showed overt signs of toxicity reflected in their behavior, body mass, or blood chemistry. The biodistribution of Si NPs in mice was also quantitatively

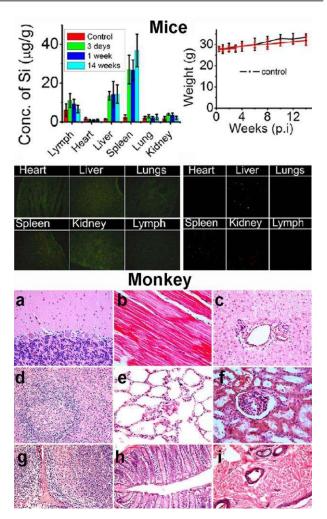


Figure 23. (**Top**) biodistributions of Si NPs in mice assessed by ICP-MS analysis, fluorescence images of frozen tissue sections, confocal microscopy images; (**bottom**) histological images of (a) brain, (b) heart, (c) liver, (d) spleen, (e) lung, (f) kidney, (g) lymph, (h) intestine, and (i) skin harvested from rhesus macaques administrated with Si NPs. Reproduced from Ref. 207.

evaluated by *in vivo* positron emission tomography (PET) imaging and *ex vivo* gamma counting.²⁰⁸ A new macrocyclic ligand-⁶⁴Cu²⁺ complex was conjugated with dextran-coated Si NPs and served as PET agent. The results show that conjugates were excreted via renal filtration shortly post injection and also accumulated in the liver, again demonstrating the stability and biocompatibility of Si NPs.

In summary, Si NPs have tuneable fluorescence from visible to near-infrared window, excellent biocompatibility, chemical, and photostability. These properties make them very attractive in bioimaging, however long-term studies on their safety and adverse effects are still needed for their clinical applications.

7 Summary and outlook

We have summarized the current state of the art on ultra-small inorganic nanoparticles for fluorescence bioimaging. These ultra-small nanoparticles bridge the gap between big particles and molecules. They have unique properties and great potential in

molecular imaging for diagnosis and treatment of cancer and other diseases, as they could escape from macrophages, pass biological barriers, and be easily degraded or excreted in comparison with large particles.

The ultra-small fluorescent probes we addressed include the conventional QDs, fluorescent metal nanoclusters, carbon-based up-conversion nanocrystals, nanomaterials, and silicon nanoparticles. The fluorescence mechanisms in metal nanoclusters, carbon dots, and graphene quantum dots are not 10 completely clear as yet. Although they can be prepared by various wet chemistry methods, it remains a challenge to prepare robust fluorescent probes with high photostability (i.e. nonblinking), chemical stability, high quantum yield, and tunable emissions in the visible to NIR window. From the applications 15 perspective, some of them face the issue of toxicity, especially semiconducting QDs and up-conversion nanocrystals, as they have toxic elements such as cadmium and lanthanides. Various approaches and coatings have been developed to modify and functionalize their surfaces to overcome these shortcomings. In 20 addition to the issues of fluorescent nanoprobes themselves, there are some important issues that have to be considered for practical applications, including their interactions with proteins and other biomolecules, their interactions with cells, their endocytosis and intracellular stability and behaviour, and their metabolism and 25 excretion. These issues have not been well understood, despite some progresses on bio-interface interactions in the biological environments have been made in recent years. 61-63

The use of fluorescence imaging alone could lead to inaccurate diagnosis or misdiagnosis, due to the low spatial and temporal resolution, and the sensitivity of fluorescence to external environments. Simultaneous use of multi-modal imaging (e.g. magneto-fluorescence) could overcome the disadvantages of individual methods. There are increasing reports on the combination of fluorescence with other imaging methods such as MRI, CT, and PET. One 209-212 There are few commercial multifunctional instruments for multimodal imaging, however. Relocating biological samples between different imaging instruments could lead to inaccuracy. Development of multimodal imaging that employs a single instrument is an attractive solution. In addition, incorporation of therapeutics into multimodal nano-agents for early detection and treatment will be an important feature of future nanotheranostics.

Notes and references

- Institute of Superconducting & Electronic Materials, The University of
 Wollongong, NSW 2500, Australia. Fax: +61-2-42215731; E-mail: zhenl@uow.edu.au
 - ^b Australian Institute for Bioengineering and Nanotechnology, The University of Queensland, Queensland 4072, Australia. Fax: +61-7-33463973; E-mail: gordonxu@uq.edu.au
- 50 School of Chemistry and Chemical Engineering, Huazhong University of Science and Technology, Wuhan, 430074, China.
- † Z. Li acknowledges support from the Australian Research Council (ARC) through the Discovery Projects DP130102699 and DP130102274, and the Australian Institute of Innovative Materials (AIIM) for a 55 collaborative grant.
 - 1. P. Alivisatos, Science, 1996, 271, 933-937.

- Z. Li, Q. Sun and M. Y. Gao, Angew. Chem. Int. Ed., 2005, 44, 123-126.
- M. F. Yu, O. Lourie, M. J. Dyer, K. Moloni, T. F. Kelly and R. S. Ruoff, Science, 2000, 287, 637-640.
 - 4. A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183-191.
- K. N. Shinde, S. J. Dhoble, H. C. Swart and K. Park, eds., *Phosphate Phosphors for Solid-State Lighting*, Springer-Verlag, Berlin, 2012.
- 65 6. B. N. G. Giepmans, S. R. Adams, M. H. Ellisman and R. Y. Tsien, Science, 2006, 312, 217-224.
- 7. R. Weissleder and M. J. Pittet, Nature, 2008, 452, 580-589.
- 8. A. Policard, C R Séances Soc Biol Fil., 1924, 91, 1423-1424.
- 9. P. Alivisatos, Nat. Biotechnol., 2004, 22, 47-52.
- 70 10. X. Michalet, F. F. Pinaud, L. A. Bentolila, J. M. Tsay, S. Doose, J. J. Li, G. Sundaresan, A. M. Wu, S. S. Gambhir and S. Weiss, *Science*, 2005, 307, 538-544.
 - I. L. Medintz, H. T. Uyeda, E. R. Goldman and H. Mattoussi, *Nat. Mater.*, 2005, 4, 435-446.
- 75 12. V. Biju, T. Itoh and M. Ishikawa, Chem. Soc. Rev., 2010, 39, 3031-3056.
- 13. H. Koo, M. S. Huh, J. H. Ryu, D. E. Lee, I. C. Sun, K. Choi, K. Kim and I. C. Kwon, *Nano Today*, 2011, **6**, 204-220.
- I. L. Medintz, M. H. Stewart, S. A. Trammell, K. Susumu, J. B.
 Delehanty, B. C. Mei, J. S. Melinger, J. B. Blanco-Canosa, P. E.
 Dawson and H. Mattoussi, *Nat. Mater.*, 2010, 9, 676-684.
- L. Shang, F. Stockmar, N. Azadfar and G. U. Nienhaus, *Angew. Chem. Int. Ed.*, 2013, 52, 11154-11157.
- A. W. Zhu, Q. Qu, X. L. Shao, B. Kong and Y. Tian, *Angew. Chem. Int. Ed.*, 2012, 51, 7185-7189.
- P. G. Luo, S. Sahu, S.-T. Yang, S. K. Sonkar, J. Wang, H. Wang, G. E. LeCroy, L. Cao and Y.-P. Sun, *J. Mater. Chem. B*, 2013, 1, 2116-2127.
- M. L. Mastronardi, E. J. Henderson, D. P. Puzzo and G. A. Ozin, *Adv. Mater.*, 2012, 24, 5890-5898.
- 19. D. D. V. II and R. E. Schaak, Chem. Soc. Rev., 2013, 42, 2861-2879.
- F. Peng, Y. Y. Su, Y. L. Zhong, C. Fan, S.-T. Lee and Y. He, Acc. Chem. Res., 2014, 47, 10.1021/ar400221g.
- 21. P. M. Allen and M. G. Bawendi, *J. Am. Chem. Soc.*, 2008, **130**, 9240-
- H. Z. Zhong, Z. L. Bai and B. S. Zou, J. Phys. Chem. Lett., 2012, 3, 3167-3175.
- 23. S. L. Shen and Q. B. Wang, Chem. Mater., 2013, 25, 1166-1178.
- C. B. Murray, D. J. Norris and M. G. Bawendi, J. Am. Chem. Soc., 1993, 115, 8706-8715.
- 25. Y. Yin and A. P. Alivisatos, Nature, 2005, 437, 664-670.
- 26. Z. A. Peng and X. G. Peng, J. Am. Chem. Soc., 2001, 123, 183-184.
- D. V. Talapin, A. L. Rogach, A. Kornowski, M. Haase and H. Weller, Nano Lett., 2001, 1 207-211.
- 105 28. Z. A. Peng and X. G. Peng, J. Am. Chem. Soc., 2002, 124, 3343-3353.
 - 29. X. G. Peng, L. Manna, W. D. Yang, J. Wickham, E. Scher, A. Kadavanich and A. P. Alivisatos, *Nature*, 2000, **404**, 59-61.
 - 30. X. G. Peng, Adv. Mater., 2003, 15, 459-463.
- 110 31. D. J. Milliron, S. M. Hughes, Y. Cui, L. Manna, J. B. Li, L. W. Wang and A. P. Alivisatos, *Nature*, 2004, 430, 190-195.
 - X. Peng, J. Wickham and A. P. Alivisatos, J. Am. Chem. Soc., 1998, 120, 5343-5344.

- 33. L. H. Qu and X. G. Peng, J. Am. Chem. Soc., 2002, 124, 2049-2055.
- 34. J. J. Li, Y. A. Wang, W. Z. Guo, J. C. Keay, T. D. Mishima, M. B. Johnson and X. G. Peng, *J. Am. Chem. Soc.*, 2003, **125**, 12567-12575.
- R. G. Xie, U. Kolb, J. X. Li, T. Basche and A. Mews, J. Am. Chem. Soc., 2005, 127, 7480-7488.
- O. Chen, J. Zhao, V. P. Chauhan, J. Cui, C. Wong, D. K. Harris, H. Wei, H.-S. Han, D. Fukumura, R. K. Jain and M. G. Bawendi, *Nat. Mater.*, 2013, 12, 445-451.
- 37. H. Y. Qin, Y. Niu, R. Y. Meng, X. Lin, R. C. Lai, W. Fang and X. G. Peng, *J. Am. Chem. Soc.*, 2014, **136**, 179-187.
- X. Y. Wang, X. F. Ren, K. Kahen, M. A. Hahn, M. Rajeswaran, S. Maccagnano-Zacher, J. Silcox, G. E. Cragg, A. L. Efros and T. D. Krauss, *Nature*, 2009, 459, 686-689.
- 39. Z. Li, Q. Sun, Z. Zhu, S. C. Smith and G. Q. M. Lu, in *One-Dimensional Nanostructures: Principles and Applications*, eds. T. Y. Zhai and J. N. Yao, Wiley, 2013, pp. 65-102.
- Z. Li, L. N. Cheng, Q. Sun, Z. H. Zhu, M. J. Riley, M. Aljada, Z. X. Cheng, X. L. Wang, G. R. Hanson, S. Z. Qiao, S. C. Smith and G. Q. Lu, *Angew. Chem. Int. Ed.*, 2010, 49, 2777-2781.
- 20 41. Z. Li, A. Kornowski, A. Myalitsin and A. Mews, Small, 2008, 4, 1698-1702.
 - Z. Li, Ö. Kurtulus, F. Nan, A. Myalitsin, Z. Wang, A. Kornowski, U. Pietsch and A. Mews, Adv. Funct. Mater., 2009, 19, 3650-3661.
- 43. Z. Wang, Z. Li, A. Kornowski, X. D. Ma, A. Myalitsin and A. Mews, *Small*, 2011, **7**, 2464-2468.
- N. Fu, Z. Li, A. Myalitsin, M. Scolari, R. T. Weitz, M. Burghard and A. Mews, Small, 2010, 6, 376-380.
- Z. Li, A. J. Du, Q. Sun, M. Aljada, L. N. Cheng, Z. H. Zhu, M. J. Riley, Z. X. Cheng, X. L. Wang, J. Hall, E. Krausz, S. Z. Qiao, S. C. Smith and G. Q. Lu, *Chem. Commun.*, 2011, 47, 11894-11896.
- A. Myalitsin, C. Strelow, Z. Wang, Z. Li, T. Kipp and A. Mews, ACS Nano, 2011, 5, 7920-7927.
- 47. Z. Li, A. J. Du, Q. Sun, M. Aljada, Z. H. Zhu and G. Q. Lu, *Nanoscale*, 2012, **4**, 1263-1266.
- 35 48. Z. Li, Q. Sun, X. D. Yao, Z. H. Zhu and G. Q. M. Lu, J. Mater. Chem., 2012, 22, 22821-22831.
- 49. C. Han, Z. Li and S. X. Dou, Chin. Sci. Bulletin, 2014, In press.
- O. Kurtulus, Z. Li, A. Mews and U. Pietsch, *Phys. Stat. Sol. A*, 2009, 206, 1752-1756.
- K. P. Kandel, U. Pietsch, Z. Li and O. Kurtulus, *Phys. Chem. Chem. Phys.*, 2013, 15, 4444-4450.
 - Z. Li, X. Ma, Q. Sun, Z. Wang, J. Liu, Z. Zhu, S. Z. Qiao, S. C. Smith, G. M. Lu and A. Mews, *Eur. J. Inorg. Chem.*, 2010, 27, 4325-4331.
- 45 53. A. Henglein, Ber. Bunsen-Ges. Phys. Chem., 1982, 86, 301-305.
 - 54. U. Resch, H. Weller and A. Henglein, Langmuir, 1989, 5, 1015-1020.
 - A. L. Rogach, L. Katsikas, A. Kornowski, I. G. Popovic, D. Su, A. Eychmuller and H. Weller, *Ber. Bunsen-Ges. Phys. Chem.*, 1996, 100, 1772-1778.
- 50 56. Y. L. Li, L. H. Jing, R. R. Qiao and M. Y. Gao, *Chem. Comm.*, 2011, 105 47, 9293-9311.
 - H. B. Bao, Y. J. Gong, Z. Li and M. Y. Gao, *Chem. Mater.*, 2004, 16, 3853-3859.
- 58. Y. Zhu, Z. Li, M. Chen, H. M. Cooper, G. Q. Lu and Z. P. Xu, *J. Colloid Interface Sci.*, 2013, **390**, 3-10.

- S. R. Sturzenbaum, M. Hockner, A. Panneerselvam, J. Levitt, J. S. Bouillard, S. Taniguchi, L. A. Dailey, R. A. Khanbeigi, E. V. Rosca, M. Thanou, K. Suhling, A. V. Zayats and M. Green, *Nat. Nanotechnol.*, 2013, 8, 57-60.
- 60 60. Y. A. Zhu, H. Hong, Z. P. Xu, Z. Li and W. B. Cai, *Curr. Mol. Med.*, 2013, 13, 1549-1567.
 - L. C. Cheng, X. M. Jiang, J. Wang, C. Y. Chen and R. S. Liu, Nanoscale, 2013, 5, 3547-3569.
- 62. L. Shang and G. U. Nienhaus, Mater. Today, 2013, 16, 58-66.
- 65 63. Y. C. Wang, R. Hu, G. M. Lin, I. Roy and K. T. Yong, ACS Appl. Mater. Interfaces, 2013, 5, 2786-2799.
- 64. W. C. W. Chan and S. M. Nie, Science, 1998, 281, 2016-2018.
- C. X. C. Lin, Z. Li, S. Brumbley, L. Petrasovits, R. McQualter, C. Yu and G. Q. M. Lu, J. Mater. Chem., 2011, 21, 7565-7571.
- 70 66. A. M. Derfus, W. C. W. Chan and S. N. Bhatia, *Nano Lett.*, 2004, 4, 11-18.
- M. Bruchez, M. Moronne, P. Gin, S. Weiss and A. P. Alivisatos, Science, 1998, 281, 2013-2016.
- 68. W. Stöber and A. Fink, J Colloid Interf Sci., 1968, 26, 62-29.
- 75 69. Y. H. Yang and M. Y. Gao, Adv. Mater., 2005, 17, 2354-2357.
 - T. Nann and P. Mulvaney, Angew. Chem. Int. Ed., 2004, 43, 5393-5396.
 - L. Jing, C. Yang, R. Qiao, M. Niu, M. Du, D. Wang and M. Gao, *Chem. Mater.*, 2010, 22, 420-427.
- 80 72. Y. Zhu, Z. Li, M. Chen, H. M. Cooper, G. Q. Lu and Z. P. Xu, *Chem. Mater.*, 2012, **24**, 421-423.
 - W. Jiang, B. Y. S. Kim, J. T. Rutka and W. C. W. Chan, *Nat. Nanotechnol.*, 2008, 3, 145-150.
- 74. Y. Zhu, Z. Li, M. Chen, H. M. Cooper and Z. P. Xu, *J. Mater. Chem. B*, 2013, **1**, 2315-2323.
- Z. L. Cheng, A. Al Zaki, J. Z. Hui, V. R. Muzykantov and A. Tsourkas, Science, 2012, 338, 903-910.
- X. H. Gao, Y. Y. Cui, R. M. Levenson, L. W. K. Chung and S. M. Nie, *Nat. Biotechnol.*, 2004, 22, 969-976.
- 90 77. E. I. Sega and P. S. Low, Cancer Metast Rev, 2008, 27, 655-664.
 - B. Xue, D. W. Deng, J. Cao, F. Liu, X. Li, W. Akers, S. Achilefu and Y. Q. Gu, *Dalton T*, 2012, 41, 4935-4947.
 - 79. D. A. Ossipov, Expert opinion on drug delivery, 2010, 7, 681-703.
 - 80. S. H. Bhang, N. Won, T. J. Lee, H. Jin, J. Nam, J. Park, H. Chung, H. S. Park, Y. E. Sung, S. K. Hahn, B. S. Kim and S. Kim, *Acs Nano*,
 - 81. K. Bourzac, Nature, 2013, 493, 283-283.

2009, 3, 1389-1398.

- 82. R. Hardman, Environ. Health Perspect., 2006, 114, 165-172.
- H. S. Choi, W. Liu, P. Misra, E. Tanaka, J. P. Zimmer, B. I. Ipe, M.
 G. Bawendi and J. V. Frangioni, *Nat. Biotechnol.*, 2007, 25, 1165-1170.
- 84. H. S. Choi, W. H. Liu, F. B. Liu, K. Nasr, P. Misra, M. G. Bawendi and J. V. Frangioni, *Nat. Nanotechnol.*, 2010, 5, 42-47.
- L. Ye, K. T. Yong, L. W. Liu, I. Roy, R. Hu, J. Zhu, H. X. Cai, W. C.
 Law, J. W. Liu, K. Wang, J. Liu, Y. Q. Liu, Y. Z. Hu, X. H. Zhang,
 M. T. Swihart and P. N. Prasad, *Nat. Nanotechnol.*, 2012, 7, 453-458.
- S. Link and M. A. El-Sayed, Annu. Rev. Phys. Chem., 2003, 54, 331-366.
- 87. L. Shang, S. J. Dong and G. U. Nienhaus, *Nano Today*, 2011, **6**, 401-
- 88. Y. Z. Lu and W. Chen, Chem. Soc. Rev., 2012, 41, 3594-3623.

- S. Choi, R. M. Dickson and J. H. Yu, Chem. Soc. Rev., 2012, 41, 1867-1891.
- Y. H. Li, J. Xing, Z. J. Chen, Z. Li, F. Tian, L. R. Zheng, H. F. Wang,
 P. Hu, H. J. Zhao and H. G. Yang, *Nat. Commun.*, 2013, 4.
- 5 91. H. F. Qian, M. Z. Zhu, Z. K. Wu and R. C. Jin, Acc. Chem. Res., 2012, 45, 1470-1479.
- 92. A. Mooradian, Phys. Rev. Lett., 1969, 22, 185-187.
- J. P. Wilcoxon, J. E. Martin, F. Parsapour, B. Wiedenman and D. F. Kelley, *J. Chem. Phys.*, 1998, 108, 9137-9143.
- 10 94. J. Zheng, J. T. Petty and R. M. Dickson, J. Am. Chem. Soc., 2003, 125, 7780-7781.
 - J. Zheng, C. W. Zhang and R. M. Dickson, *Phys. Rev. Lett.*, 2004, 93, 077402.
- J. Zheng, P. R. Nicovich and R. M. Dickson, *Annu. Rev. Phys. Chem.* , 2007, 58, 409-431.
- J. Zheng and R. M. Dickson, J. Am. Chem. Soc., 2002, 124, 13982-13983.
- 98. Z. Shen, H. Duan and H. Frey, Adv. Mater., 2007, 19, 349-352.
- F. Aldeek, M. A. H. Muhammed, G. Palui, N. Zhan and H. Mattoussi,
 ACS Nano, 2013, 7, 2509-2521.
- 100.X. Huang, Y. Luo, Z. Li, B. Li, H. Zhang, L. Li, I. Majeed, P. Zou and B. Tan, J. Phys. Chem. C, 2011, 115, 16753-15763.
- 101.Z. Li, B. Tan, M. Allix, A. I. Cooper and M. J. Rosseinsky, *Small*, 2008, 4, 231-239.
- 25 102.Z. Li, P. W. Yi, Q. Sun, H. Lei, H. L. Zhao, Z. H. Zhu, S. C. Smith, M. Lan and G. M. Lu, Adv. Funct. Mater., 2012, 22, 2387-2393.
 - 103.Z. Li, S. X. Wang, Q. Sun, H. L. Zhao, H. Lei, M. Lan, Z. X. Chen, X. L. Wang, S. X. Dou and G. M. Lu, *Adv. Healthcare Mater.*, 2013, 2 958-964
- 30 104.M. I. Majeed, Q. W. Lu, W. Yan, Z. Li, I. Hussain, M. N. Tahir, W. Tremel and B. Tan, J. Mater. Chem. B, 2013, 1, 2874-2884.
 - 105.H. Zhang, X. Huang, L. Li, G. W. Zhang, I. Hussain, Z. Li and B. Tan, Chem. Comm., 2012, 48, 567-569.
 - 106.L. Shang and S. Dong, Chem. Commun., 2008, 1088-1090.
- 35 107.L. Li, Z. Li, H. Zhang, S. C. Zhang, I. Majeed and B. Tan, Nanoscale, 2013, 5, 1986-1992.
 - 108.R. C. Jin, Nanoscale, 2010, 2, 343-362.
 - 109.Z. Luo, X. Yuan, Y. Yu, Q. Zhang, D. T. Leong, J. Y. Lee and J. Xie, J. Am. Chem. Soc., 2012, 134, 16662-166670.
- 40 110.J. T. Petty, J. Zheng, N. V. Hud and R. M. Dickson, J. Am. Chem. Soc., 2004, 126, 5207-5212.
 - 111.C. I. Richards, S. Choi, J. C. Hsiang, Y. Antoku, T. Vosch, A. Bongiorno, Y. L. Tzeng and R. M. Dickson, *J. Am. Chem. Soc.*, 2008, **130**, 5038-+.
- 45 112.J. P. Xie, Y. G. Zheng and J. Y. Ying, J. Am. Chem. Soc., 2009, 131, 888-889.
 - 113.C.-L. Liu, H.-T. Wu, Y.-H. Hsiao, C.-W. Lai, C.-W. Shih, Y.-K. Peng, K.-C. Tang, H.-W. Chang, Y.-C. Chien, J.-K. Hsiao, J.-T. Cheng and P.-T. Chou, *Angew. Chem. Int. Ed.*, 2011, **50**, 7056-7060.
- 50 114.J. L. Wang, G. Zhang, Q. W. Li, H. Jiang, C. Y. Liu, C. Amatore and X. M. Wang, Sci. Rep., 2013, 3, 1157.
 - 115.M. A. H. Muhammed, S. Ramesh, S. S. Sinha, S. K. Pal and T. Pradeep, *Nano Res.*, 2008, 1, 333-340.
- 116.T. U. B. Rao and T. Pradeep, *Angew. Chem. Int. Ed.*, 2010, **49**, 3925-5 3929.

- 117.T. Udayabhaskararao, Y. Sun, N. Goswami, K. B. Samir K. Pal and T. Pradeep, *Angew. Chem. Int. Ed.*, 2012, **51**, 2155-2159.
- 118.H. W. Duan and S. M. Nie, *J. Am. Chem. Soc.*, 2007, **129**, 2412-2413.
- 60 119.C. A. J. Lin, T. Y. Yang, C. H. Lee, S. H. Huang, R. A. Sperling, M. Zanella, J. K. Li, J. L. Shen, H. H. Wang, H. I. Yeh, W. J. Parak and W. H. Chang, *ACS Nano*, 2009, 3, 395-401.
- 120.X. Yuan, Z. Luo, Q. Zhang, X. Zhang, Y. Zheng, J. Y. Lee and J. Xie, ACS Nano, 2011, 5, 8800-8808.
- 65 121.H. Xu and K. S. Suslick, ACS Nano, 2010, 4, 3209-3214.
 - 122.L. Shang, L. Yang, F. Stockmar, R. Popescu, V. Trouillet, M. Bruns, D. Gerthsen and G. U. Nienhaus, *Nanoscale*, 2012, 4, 4155-4160.
 - 123.Z. Wu and R. Jin, Nano Lett, 2010, 10, 2568-2573.
 - 124.H. Tsunoyama, Y. Negishi and T. Tsukuda, *J. Am. Chem. Soc.*, 2006, **128**, 6036-6037.
 - 125.Y. Negishi, K. Nobusada and T. Tsukuda, J. Am. Chem. Soc., 2005, 127, 5261-5270.
 - 126.C. E. Briant, B. R. C. Theobald, J. W. White, L. K. Bell, D. M. P. Mingos and A. J. Welch, *Chem. Commun.*, 1981, 201-202.
- 75 127.P. D. Jadzinsky, G. Calero, C. J. Ackerson, D. A. Bushnell and R. D. Kornberg, *Science*, 2007, 318, 430-433.
 - 128.M. W. Heaven, A. Dass, P. S. White, K. M. Holt and R. W. Murray, J. Am. Chem. Soc., 2008, 130, 3754-3755.
- 129.H. Xiang, S.-H. Wei and X. Gong, *J. Am. Chem. Soc.*, 2010, **132**, 7355-7360.
- 130.O. M. Bakr, V. Amendola, C. M. Aikens, W. Wenseleers, R. Li, L. D. Negro, G. C. Schatz and F. Stellacci, *Angew. Chem. Int. Ed.*, 2009, 48, 5921-5926.
- 131.M. Zhu, C. M. Aikens, F. J. Hollander, G. C. Schatz and R. Jin, *J. Am. Chem. Soc.*, 2008, **130**, 5883-5885.
- 132.S. Link, A. Beeby, S. FitzGerald, M. A. El-Sayed, T. G. Schaaff and R. L. Whetten, *J. Phys. Chem. B*, 2002, **106**, 3410-3415.
- 133.Y. Yu, Z. Luo, D. M. Chevrier, D. T. Leong, P. Zhang, D.-e. Jiang and J. Xie, J. Am. Chem. Soc., 2014, 136, ASAP.
- 90 134.A. Samanta, B. B. Dhar and R. N. Devi, J. Phys. Chem. C, 2012, 106, 1748-1754.
- 135.N. Makarava, A. Parfenov and I. V. Baskakov, *Biophys. J.*, 2005, 90, 572-580.
- 136.A. Retnakumari, S. Setua, D. Menon, P. Ravindran, H. Muhammed,
 T. Pradeep, S. Nair and M. Koyakutty, *Nanotechnology*, 2010, 21, 055103.
- 137.X. Wu, X. He, K. Wang, C. Xie, B. Zhou and Z. Qing, *Nanoscale*, 2010, **2**, 2244-2249.
- 138.C. Zhou, M. Long, Y. Qin, X. Sun and J. Zheng, *Angew. Chem. Int.* 60 Ed., 2011, **50**, 3168-3172.
- 139.J. Liu, M. Yu, C. Zhou, S. Yang, X. Ning and J. Zheng, J. Am. Chem. Soc., 2013, 135, 4978-4981.
- 140.J. Shen, Y. Zhu, X. Yang and C. Li, Chem. Commun., 2012, 48, 3686-3699.
- 105 141.X. Xu, R. Ray, Y. Gu, H. J. Ploehn, L. Gearheart, K. Raker and W. A. Scrivens, J. Am. Chem. Soc., 2004, 126, 12736-12737.
 - 142.Y.-P. Sun, B. Zhou, Y. Lin, W. Wang, K. A. S. Fernando, P. Pathak, M. J. Meziani, B. A. Harruff, X. Wang, H. Wang, P. G. Luo, H. Yang, M. E. Kose, B. Chen, L. M. Veca and S.-Y. Xie, *J. Am. Chem. Soc.*, 2006, **128**, 7756-7757.

- 143.X. Wang, L. Cao, S.-T. Yang, F. Lu, M. J. Meziani, L. Tian, K. W. Sun, M. A. Bloodgood and Y.-P. Sun, *Angew. Chem. Int. Ed.*, 2010, 49, 5310-5314.
- 144.S. K. Bhunia, A. Saha, A. R. Maity, S. C. Ray and N. R. Jana, Sci. Rep., 2013, 3, 1473.
- 145.S. Zhu, Q. Meng, L. Wang, J. Zhang, Y. Song, H. Jin, K. Zhang, H. Sun, H. Wang and B. Yang, *Angew. Chem. Int. Ed.*, 2013, **52**, 3953-3957
- 146.Y. X. Fang, S. J. Guo, D. Li, C. Z. Zhu, W. Ren, S. J. Dong and E. K. Wang, *ACS Nano*, 2012, **6**, 400-409.
 - 147.Q. L. Wang, X. X. Huang, Y. J. Long, X. L. Wang, H. J. Zhang, R. Zhu, L. P. Liang, P. Teng and H. Z. Zheng, *Carbon*, 2013, **59**, 192-199.
- 148.C. Ding, A. Zhu and Y. Tian, Acc. Chem. Res., 2013, 46, 10.1021/ar400023s.
- 149.S. N. Baker and G. A. Baker, Angew. Chem. Int. Ed., 2010, 49, 6727-6744.
- 150.L. Cao, X. Wang, M. J. Meziani, F. Lu, H. Wang, P. G. Luo, Y. Lin, B. A. Harruff, L. M. Veca, D. Murray, S.-Y. Xie and Y.-P. Sun, J. Am. Chem. Soc., 2007, 129, 11318-11319.
- 151.S. T. Yang, L. Cao, P. G. J. Luo, F. S. Lu, X. Wang, H. F. Wang, M. J. Meziani, Y. F. Liu, G. Qi and Y. P. Sun, J. Am. Chem. Soc., 2009, 131, 11308-11309.
- 152.X. L. Huang, F. Zhang, L. Zhu, K. Y. Choi, N. Guo, J. X. Guo, K.
 Tackett, P. Anilkumar, G. Liu, Q. M. Quan, H. S. Choi, G. Niu, Y. P.
 Sun, S. Lee and X. Y. Chen, ACS Nano, 2013, 7, 5684-5693.
- 153.J. H. Shen, Y. H. Zhu, X. L. Yang and C. Z. Li, Chem. Commun., 2012, 48, 3686-3699.
- 154.L. L. Li, G. H. Wu, G. H. Yang, J. Peng, J. W. Zhao and J. J. Zhu,Nanoscale, 2013, 5, 4015-4039.
- 155.S. J. Zhu, J. H. Zhang, S. J. Tang, C. Y. Qiao, L. Wang, H. Y. Wang, X. Liu, B. Li, Y. F. Li, W. L. Yu, X. F. Wang, H. C. Sun and B. Yang, Adv. Funct. Mater., 2012, 22, 4732-4740.
- 156.S. J. Zhu, J. H. Zhang, C. Y. Qiao, S. J. Tang, Y. F. Li, W. J. Yuan,
 B. Li, L. Tian, F. Liu, R. Hu, H. N. Gao, H. T. Wei, H. Zhang, H. C. Sun and B. Yang, *Chem. Commun.*, 2011, 47, 6858-6860.
- 157.H. Tetsuka, R. Asahi, A. Nagoya, K. Okamoto, I. Tajima, R. Ohta and A. Okamoto, *Adv. Mater.*, 2012, **24**, 5333-5338.
- 158.X. Wu, F. Tian, W. X. Wang, J. Chen, M. Wu and J. X. Zhao, *J. Mater. Chem. C*, 2013, **1**, 4676-4684.
- 159.Y. Dong, H. Pang, H. B. Yang, C. Guo, J. Shao, Y. Chi, C. M. Li and T. Yu, Angew. Chem. Int. Ed., 2013, 52, 7800-7804.
- 160.Q. Xu, Q. Zhou, Z. Hua, Q. Xue, C. Zhang, X. Wang, D. Pan and M. Xiao, ACS Nano, 2013, 7, 10654-10661.
- 45 161.S. J. Zhuo, M. W. Shao and S. T. Lee, ACS Nano, 2012, 6, 1059-1064.
 - 162.X. Sun, Z. Liu, K. Welsher, J. T. Robinson, A. Goodwin, S. Zaric and H. Dai, *Nano Res.*, 2008, 1, 203-212.
- 163.J. Qian, D. Wang, F.-H. Cai, W. Xi, L. Peng, Z.-F. Zhu, H. He, M.-L. Hu and S. He, *Angew. Chem. Int. Ed.*, 2012, **51**, 10570-10575.
- 164.Q. Liu, B. D. Guo, Z. Y. Rao, B. H. Zhang and J. R. Gong, *Nano Lett.*, 2013, **13**, 2436-2441.
- 165.Z. Gu, L. Yan, G. Tian, S. Li, Z. Chai and Y. Zhao, Adv. Mater., 2013, 25, 3758-3779.
- 55 166.F. Wang and X. G. Liu, Chem. Soc. Rev., 2009, 38, 976-989.

- 167.F. Wang, D. Banerjee, Y. S. Liu, X. Y. Chen and X. G. Liu, *Analyst*, 2010, **135**, 1839-1854.
- 168.C. Bouzigues, T. Gacoin and A. Alexandrou, ACS Nano, 2011, 5, 8488-8505.
- 60 169.G. Chen, C. Yang and P. N. Prasad, Acc. Chem. Res., 2013, 46, 1474-1486.
- 170.L. Cheng, C. Wang and Z. Liu, *Nanoscale*, 2013, **5**, 23-37.
- 171.L. D. Sun, Y. F. Wang and C. H. Yan, Acc. Chem. Res., 2014, 47, ASAP.
- 65 172.N. J. J. Johnson, W. Oakden, G. J. Stanisz, R. S. Prosser and F. C. J. M. van Veggel, *Chem. Mater.*, 2011, 23, 3714-3722.
 - 173.G. Chen, T. Y. Ohulchanskyy, R. Kumar, H. Agren and P. N. Prasad, ACS Nano, 2010, 4, 3163-3168.
- 174.L. Gong, J. Yang, Y. Li, M. Ma, C. Xu, G. Ren, J. Lin and Q. Yang, *J. Mater. Sci.*, 2013, **48**, 3672-3678.
- 175.J. Ryu, H.-Y. Park, K. Kim, H. Kim, J. H. Yoo, M. Kang, K. Im, R. Grailhe and R. Song, J. Phys. Chem., 2010, 114, 21077-21082.
- 176.C.-f. Xu, M. Ma, L.-w. Yang, S.-j. Zeng and Q.-b. Yang, J. Colloid Interface Sci., 2012, 368, 49-55.
- 75 177.A. D. Ostrowski, E. M. Chan, D. J. Gargas, E. M. Katz, G. Han, P. J. Schuck, D. J. Milliron and B. E. Cohen, ACS Nano, 2012, 6, 2686-2692.
 - 178.J. Zhao, D. Jin, E. P. Schartner, Y. Lu, Y. Liu, A. V. Zvyagin, L. Zhang, J. Dawes, P. Xi, J. A. Piper, E. Goldys and T. M. Monro, *Nat. Nanotechnol.*, 2013, 8, 729-734.
 - 179.S. Heer, K. Kompe, H. U. Gudel and M. Haase, Adv. Mater., 2004, 16, 2102-2105.
 - 180.W. Feng and L. Xiaogang, J. Am. Chem. Soc., 2008, 130, 5642-5643.
- 181.F. Wang, Y. Han, C. S. Lim, Y. H. Lu, J. Wang, J. Xu, H. Y. Chen,
 C. Zhang, M. H. Hong and X. G. Liu, *Nature*, 2010, 463, 1061-1065.
- 182.F. Wang, R. R. Deng, J. Wang, Q. X. Wang, Y. Han, H. M. Zhu, X.
- Y. Chen and X. G. Liu, *Nat. Mater.*, 2011, **10**, 968-973.
 183.H. Wen, H. Zhu, X. Chen, T. F. Hung, B. Wang, G. Zhu, S. F. Yu and F. Wang, *Angew. Chem. Int. Ed.*, 2013, **52**, 13419-13423.
- 90 184.J. Wang, R. R. Deng, M. A. MacDonald, B. L. Chen, J. K. Yuan, F. Wang, D. Z. Chi, T. S. A. Hor, P. Zhang, G. K. Liu, Y. Han and X. G. Liu, *Nat. Mater.*, 2013, 12, 10.1038/nmat3804.
 - 185.Q. Liu, W. Feng, T. Yang, T. Yi and F. Li, Nat. Protoc., 2013, 8, 2033-2044.
- 95 186.Y. Dai, H. Xiao, J. Liu, Q. Yuan, P. a. Ma, D. Yang, C. Li, Z. Cheng, Z. Hou, P. Yang and J. Lin, J. Am. Chem. Soc., 2013, 135, 18920-18929.
 - 187.L. Q. Xiong, T. S. Yang, Y. Yang, C. J. Xu and F. Y. Li, Biomaterials, 2010, 31, 7078-7085.
- 100 188.L. Cheng, K. Yang, M. W. Shao, X. H. Lu and Z. Liu, *Nanomedicine*, 2011, 6, 1327-1340.
 - 189.C. Liu, Z. Gao, J. Zeng, Y. Hou, F. Fang, Y. Li, R. Qiao, L. Shen, H. Lei, W. Yang and M. Gao, ACS Nano, 2013, 7, 7227-7240.
 - 190.S. Chinnathambi, S. Chen, S. Ganesan and N. Hanagata, *Adv. Healthcare Mater.*, 2014, **3**, 10-29.
 - 191.M. L. Mastronardi, F. Maier-Flaig, D. Faulkner, E. J. Henderson, C. Kubel, U. Lemmer and G. A. Ozin, *Nano Lett*, 2012, **12**, 337-342.
 - 192.J. Zou, R. K. Baldwin, K. A. Pettigrew and S. M. Kauzlarich, *Nano Lett.*, 2004, 4, 1181-1186.
- 110 193.D. Neiner, H. W. Chiu and S. M. Kauzlarich, J. Am. Chem. Soc., 2006, 128, 11016-11017.

- 194.M. L. Mastronardi, F. Hennrich, E. J. Henderson, F. Maier-Flaig, C. Blum, J. Reichenbach, U. Lemmer, C. Kuebel, D. Wang, M. M. Kappes and G. A. Ozin, J. Am. Chem. Soc., 2011, 133, 11928-11931.
- 195.C. Tu, X. Ma, P. Pantazis, S. M. Kauzlarich and A. Y. Louie, J. Am. Chem. Soc., 2010, 132, 2016-2023.
- 196.M. P. Singh, T. M. Atkins, E. Muthuswamy, S. Kamali, C. Tu, A. Y. Louie and S. M. Kauzlarich, *ACS Nano*, 2012, 6, 5596-5604.
- 197.M. Dasog, Z. Yang, S. Regli, T. M. Atkins, A. Faramus, M. P. Singh, E. Muthuswamy, S. M. Kauzlarich, R. D. Tilley and J. G. C. Veinot, ACS Nano, 2013, 7, 2676-2685.
- 198.J. Fuzell, A. Thibert, T. M. Atkins, M. Dasog, E. Busby, J. G. C. Veinot, S. M. Kauzlarich and D. S. Larsen, J. PHys. Chem. Lett., 2013, 4, 3806-3812.
- 199.Z. F. Li and E. Ruckenstein, Nano Lett., 2004, 4, 1463-1467.
- 15 200.F. Erogbogbo, K.-T. Yong, I. Roy, G. Xu, P. N. Prasad and M. T. Swihart, ACS Nano, 2008, 2, 873-878.
 - 201.J. H. Warner, A. Hoshino, K. Yamamoto and R. D. Tilley, *Angew. Chem. Int. Ed.*, 2005, 44, 4550-4554.
- 202.Y. He, Y. Zhong, F. Peng, X. Wei, Y. Su, Y. Lu, S. Su, W. Gu, L. Liao and S.-T. Lee, *J. Am. Chem. Soc.*, 2011, **133**, 14192-14195.
- 203.Y. Zhong, F. Peng, X. Wei, Y. Zhou, J. Wang, X. Jiang, Y. Su, S. Su, S.-T. Lee and Y. He, *Angew. Chem. Int. Ed.*, 2012, **51**, 8485-8489.
- 204.Y. Zhong, F. Peng, F. Bao, S. Wang, X. Ji, L. Yang, Y. Su, S.-T. Lee and Y. He, *J. Am. Chem. Soc.*, 2013, **135**, 8350-8356.
- 25 205.H. Sugimoto, M. Fujii, Y. Fukuda, K. Imakita and K. Akamatsu, *Nanoscale*, 2014, 6, 122-126.
 - 206.F. Erogbogbo, K.-T. Yong, I. Roy, R. Hu, W.-C. Law, W. Zhao, H. Ding, F. Wu, R. Kumar, M. T. Swihart and P. N. Prasad, ACS Nano, 2011, 5, 413-423.
- 30 207.J. W. Liu, F. Erogbogbo, K. T. Yong, L. Ye, J. Liu, R. Hu, H. Y. Chen, Y. Z. Hu, Y. Yang, J. H. Yang, I. Roy, N. A. Karker, M. T. Swihart and P. N. Prasad, *ACS Nano*, 2013, 7, 7303-7310.
 - 208.C. Tu, X. Ma, A. House, S. M. Kauzlarich and A. Y. Louie, ACS Med. Chem. Lett., 2011, 2, 285-288.
- 35 209.N. Mitchell, T. L. Kalber, M. S. Cooper, K. Sunassee, S. L. Chalker, K. P. Shaw, K. L. Ordidge, A. Badar, S. M. Janes, P. J. Blower, M. F. Lythgoe, H. C. Hailes and A. B. Tabor, *Biomaterials*, 2013, 34, 1179-1192.
- 210.M. Braddock, ed., Biomedical Imaging: The Chemistry of Labels, Probes and Contrast Agents, Royal Society of Chemistry, 2012.
- 211.N. Lee, H. R. Cho, M. H. Oh, S. H. Lee, K. Kim, B. H. Kim, K. Shin, T. Y. Ahn, J. W. Choi, Y. W. Kim, S. H. Choi and T. Hyeon, *J. Am. Chem. Soc.*, 2012, **134**, 10309-10312.
- 212.H. Y. Xing, W. B. Bu, S. J. Zhang, X. P. Zheng, M. Li, F. Chen, Q. J.
 He, L. P. Zhou, W. J. Peng, Y. Q. Hua and J. L. Shi, *Biomaterials*, 2012, 33, 1079-1089.
- 213.J.-S. Choi, J.-H. Lee, T.-H. Shin, H.-T. Song, E. Y. Kim and J. Cheon, J. Am. Chem. Soc., 2010, 132, 11015-11017.