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Ultrafast charge and discharge biscrolled yarn supercapacitors for textiles and microdevices

Abstract

Flexible, wearable, implantable and easily reconfigurable supercapacitors delivering high energy and power densities are needed for electronic devices. Here we demonstrate weavable, sewable, knottable and braidable yarns that function as high performance electrodes of redox supercapacitors. A novel technology, gradient biscrolling, provides fastion-transport yarn in which hundreds of layers of conducting-polymer-infiltrated carbon nanotube sheet are scrolled into similar to 20 mu m diameter yarn. Plying the biscrolled yarn with a metal wire current collector increases power generation capabilities. The volumetric capacitance is high (up to similar to 179 Fcm(-3)) and the discharge current of the plied yarn supercapacitor linearly increases with voltage scan rate up to similar to 80 Vs(-1) and similar to 20 Vs(-1) for liquid and solid electrolytes, respectively. The exceptionally high energy and power densities for the complete supercapacitor, and high cycle life that little depends on winding or sewing (92%, 99% after 10,000 cycles, respectively) are important for the applications in electronic textiles.

Keywords

supercapacitors, discharge, textiles, charge, microdevices, ultrafast, yarn, biscrolled

Disciplines

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Ultrafast Charge and Discharge Biscrolled Yarn Supercapacitors for Textiles and Microdevices

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Flexible, wearable, implantable, and easily reconfigurable supercapacitors delivering high energy and power densities are needed for electronic devices. Here we demonstrate weavable, sewable, knottable and braidable yarns that function as high performance electrodes of redox supercapacitors. A novel technology, gradient biscrolling, provides fast-ion-transport yarn in which hundreds of layers of conducting-polymer-infiltrated carbon nanotube sheet are scrolled into ~20 μ m diameter yarn. Plying the biscrolled yarn with a metal wire current collector increases power generation capabilities. The volumetric capacitance is high (up to ~179 F cm⁻³) and the discharge current of the plied yarn supercapacitor linearly increases with voltage scan rate up to ~80 V s⁻¹ and ~20 V s⁻¹ for liquid and solid electrolytes, respectively. The exceptionally high energy and power densities for the complete supercapacitor, and high cycle life that little depends on winding or sewing (92%, 99% after 10,000 cycles, respectively) are important for application in electronic textiles.

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Supercapacitor electrodes having engineered hierarchical structures on nano- and micro-scales can provide high energy and power densities as a result of large surface-to-volume ratios, appropriately inter-connected porosity, and exploitation of both non-Faradaic and Faradaic charge storage processes¹. Important recent advances in the micro-supercapacitor field have exploited novel active materials, such as carbon nanotubes (CNTs)², reduced graphene oxide³, activated carbon⁴, conducting polymers^{5,6}, and metal oxides⁷. Many methods for fabricating miniature energy storage devices, such as rolling thin films into nanotubes^{8,9} and micro-patterning^{10,4-6,11,12} require micro-electromechanical system (MEMS) technology, which is expensive and difficult to up-scale. Also, maintaining high energy and power densities under conditions where micro-supercapacitors are subjected to various bending and stretching deformations remains a challenge.

Strong, flexible yarn-based supercapacitors are attractive as power sources for miniaturized electronic devices¹³⁻¹⁶ such as micro-robots, wearable electronic textiles, and implantable medical devices, since they can have small volumes and could be easily integrated into variously shaped structures. However, technical challenges have limited the development of strong, flexible, and weavable yarns and fibers having attractive supercapacitor performance. Previously developed yarn supercapacitors have been based on solution-spun CNT/polymer fibers¹⁷ and dry-spun CNT fibers¹⁸, which have lower energy storage capabilities than redox materials such as conducting polymers and metal oxides.

Pseudocapacitors (redox supercapacitors) combine non-Faradic double-layer charge storage processes with redox reactions to achieve high effective capacitances and high energy storage densities¹⁹. However, while pseudocapacitors can provide higher energy storage capabilities than double-layer supercapacitors when drawn power is low, delivery of high energy at high power is a special problem unless the active material layer is very thin^{20,21}. The use of very thin active material depositions on much thicker current collectors can provide

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high energy and power densities based on active material volume or weight that are unrealistic predictors of the gravimetric or volumetric performance of the complete supercapacitor²².

We here describe weavable, braidable, sewable, and knottable yarn electrodes for redox supercapacitors that provide the combination of high energy storage densities and high rate capacities based on total cell volume. These yarns are made by a process called biscrolling, which involves inserting twist in a host sheet that is overlaid with guest. In previous work on making biscrolled yarns that are superconductors or electrodes for either batteries or fuel cells²³, a carbon nanotube sheet host was overcoated with a layer of guest particles or nanofibers, and then twist spin to make a strong hybrid yarn that can contain 95 wt% of powder. These guest particles are trapped in scroll corridors of the nanotube host that are formed by inserting twist in the bilayer stack, but can remain highly electrochemically accessible because of yarn porosity and hundred micron or smaller yarn diameters.

Results

Preparation of two-ply yarn/metal wire assemblies.

In the present work, carbon multiwalled nanotube (MWNT) bundles in the host sheet are conformally overcoated with guest before twist insertion, which decreases the degree of segregation of host and guest into layers. Even though the guest interpenetrates the host nanotube sheet (or sheet stack), there is a gradient in guest concentration from sheet outer surface to sheet bottom (**Fig. 1a**). By using this modified biscrolling method (presently named gradient biscrolling), increased yarn strength resulted, since both host and guest structurally reinforce the yarn. The high mechanical strength and flexibility of the biscrolled yarns enable fabrication of weavable two-ply yarn/metal wire assemblies that exhibit high volumetric and areal energy and power densities in both liquid and solid electrolytes. While other configurations can also be usefully exploited, the present yarn supercapacitors use identical anodes and cathodes.

More specifically, the biscrolled yarns were prepared by twist insertion in hundred nanometer thick conducting polymer infiltrated MWNT sheets (**Fig. 1a** and **Supplementary Fig. S1**). The guest MWNT aerogel sheet strips were drawn from special carbon nanotube forests that are drawable²⁴, and the conducting polymer yarn guest was subsequently deposited by vapor phase polymerization (VPP), which is a well-known method for providing uniform coatings of highly conducting polymer layers on substrates²⁵. In the present case the coated substrate structure is three-dimensional, since the conducting polymer, poly(3,4-ethylenedioxythiophene) (PEDOT), is well known to be a useful as the active material for supercapacitor electrodes, since it is highly conducting, can charge both Faradaically and non-Faradaically, has high stability in the oxidized state²⁶, and can be conveniently deposited.

Scanning electron microscope (SEM) images indicate that the oriented fibular structure of the nanotube sheet is retained after VPP (**Fig. 1b**). The thickness of membranes comprising aligned PEDOT-coated MWNTs was controlled by varying the PEDOT loading level resulting from VPP; for 75 wt% PEDOT the average nanomembrane thickness was ~102 nm (**Supplementary Fig. S2**). PEDOT-coated nanotubes have been previously investigated as supercapacitor electrodes²⁷, but complete supercapacitors were not demonstrated (so performance metrics based on total capacitor volume could not be obtained). Also, no work has been done using solid-state electrolytes, which are prerequisite for textile applications.

The PEDOT/MWNT nanomembrane was biscrolled into yarn by using a novel method for twist insertion. This method provides one solution to the problem of exploiting wet-based processing of nanotube sheets into bilayers and then into biscrolled yarn without producing surface-tension-caused collapse of nano-scale thick membranes into highly irregular yarns before twist insertion. Two nanotube sheets are successively drawn from a nanotube forest and mechanically attached to a glass slide to provide a two-layer sheet stack. Without disturbing this attachment, the catalyst-containing solution was deposited on the sheet stack and dried, the PEDOT conducting polymer was deposited by VPP and the film was washed in ethanol. Afterward, in a key step, an ethanol/water solution was used to float the hybrid nanomembrane from the substrate (except for a substrate-attached end that serves as an anchor, thereby enabling twist insertion in this membrane using a rotational electric motor (**Fig. 1c**).

A wedge structure forms as the hybrid nanomembrane is twist spun into yarn (**Fig. 1d**). Opposite lateral sides of the wedge scroll in opposite directions to produce dual-Archimedean scrolls (the dark lines at spinning wedge edges in the **Fig. 1c** illustration and the **Fig. 1d** photograph). These interconnected scrolls, which are the three-dimensional analogue of a Cornu spiral (**Fig. 1d** inset), finally twist together to form an exotic type of two-ply yarn, which has only recently been discovered²³. SEM images of the biscrolled yarn provided the yarn bias angle α , which is the angle between the yarn direction and the orientation direction of helically wrapped nanotubes on the outer yarn surface, which ranged in our experiments between ~22° and ~40°. For supercapacitor evaluations, biscrolled yarns with a bias angle of ~30° were usually used. About 8000 turns/meter of inserted twist was needed to obtain this bias angle for a biscrolled 20 µm diameter yarn.

Biscrolling generates high compressive forces²³. As a result, compact PEDOT/MWNT multilayer structures having an important degree of porosity were formed within the biscrolled yarn (**Fig. 1e**). The surface porosity of the biscrolled yarn was found to decrease approximately proportionately with increase of PEDOT content (**Supplementary Fig. S3**). The high mechanical strength (367 \pm 113 MPa), useful modulus (5.9 \pm 1.4 GPa) (**Supplementary Fig. S4**), and flexibility of biscrolled yarns enabled fabrication of two-ply

yarns and 32-yarn braids (Fig. 1f, g).

Fast charge/discharge of two-ply electrodes in liquid and solid electrolytes.

A biscrolled PEDOT/MWNT yarn was plied with a Pt wire current collector to fabricate the identical electrodes used as both anode and cathode (**Fig. 1h**). Measurements of electrochemical performance were conducted using either a 1 M H₂SO₄ liquid electrolyte or a solid poly(vinyl alcohol) (PVA-H₂SO₄) electrolyte. Unless otherwise noted the investigated biscrolled yarn has ~20 μ m diameter and a two-ply electrode means a biscrolled yarn plied with a ~25 μ m diameter Pt wire current collector.

The solid-state supercapacitors were fabricated by separately infiltrating Pt-wire-plied PEDOT/MWNT yarn anode and cathode with ~10 μ m average thickness of PVA-H₂SO₄. Two such electrodes were then plied together (**Supplementary Fig. S5**) and then dried for three days at room temperature to provide fiber supercapacitors that are several centimeters in length. **Figure 2a** show volumetric and areal capacitances as a function of scan rate from 0.01 V s⁻¹ to 100 V s⁻¹ for symmetric-electrode supercapacitors containing electrodes obtained by plying the Pt wire current collector and a ~20 μ m biscrolled yarn. Here and elsewhere, unless otherwise noted, capacitances are single-electrode values that are normalized with respect to the area or volume of the electrochemically active material (presently the PEDOT/MWNT yarn). At a scan rate of 0.01 V s⁻¹, the volumetric capacitance of the liquid electrolyte (~167 F cm⁻³) is lower than that in the solid electrolyte (~180 F cm⁻³). However, the capacitance in the liquid electrolyte exceeded the capacitance in the solid electrolyte for scan rates above 1 V s⁻¹. Volumetric capacitances of ~147 F cm⁻³ (liquid electrolyte) and ~145 F cm⁻³ (solid electrolyte) were realized for a potential scan rate of 1 V s⁻¹, which decreased to ~129 F cm⁻³ and ~97 F cm⁻³, respectively, when the scan rate was increased to 10 V s⁻¹. Using the total volume of

both electrodes (including the Pt wire current collector) and the 10 V s⁻¹ scan rate, the volumetric capacitances were 13 F cm⁻³ and 10 F cm⁻³ for liquid and solid electrolytes, respectively. These capacitances are at least 6 times higher than obtained at this scan rate for micro-supercapacitors using onion-like carbon electrodes¹⁰ (1.1 F cm⁻³) or activated carbon¹⁰ (1.5 F cm⁻³). Length-normalized capacitances for yarn supercapacitors, which are important for textile applications, were essentially the same at a 1 V s⁻¹ scan rate for liquid (0.47 mF cm⁻¹) and solid electrolytes (0.46 mF cm⁻¹). Increasing scan rate to 10 V s⁻¹ slightly decreased these capacitance (to 0.41 mF cm⁻¹ and 0.31 mF cm⁻¹ for the liquid and solid electrolytes, respectively). Absolute capacitance scaled linearly with supercapacitor length to a maximum capacitance of 10 mF for the longest device produced to date (**Supplementary Fig. S6**).

The performance of the biscrolled yarn supercapacitor was also higher than for conducting polymer and carbon-ink-based solid-electrolyte supercapacitors^{6,15,28}. Since recent high performance supercapacitors are made of nanometer-thick active materials on much larger volume substrates, normalization based on electrode geometrical area is here used to provide fair performance comparisons. Meng and Ding²⁸ reported that polypyrrole-decorated nanoporous gold membrane supercapacitors with a thickness of ~100 nm (HClO₄/PVA gel electrolyte) had an areal capacitance of 1.8 mF cm⁻² at 0.1 V s⁻¹. Wang et al.⁶ showed that polyaniline nanowire micro-supercapacitors with a thickness of ~400 nm (H₂SO₄/PVA gel electrolyte) had a capacitance of 23.5 mF cm⁻² at 0.1 mA cm⁻². Also Fu et al.¹⁵ reported areal capacitance (9.5 mF cm⁻²) for a fiber supercapacitor with ~21 µm thickness of ink-deposited carbon (H₂SO₄/PVA gel electrolyte) on a 150 µm diameter nickel wire, which was packaged using a plastic tube for a flexible/wearable energy device. The two-ply electrode supercapacitor presented here has a capacitance of 73 mF cm⁻² at 1 V s⁻¹ that is at least 3 times higher than for the previously described supercapacitors.

Although the PEDOT undergoes changes in volume changes and electrical conductivity during redox reaction²⁹, cyclic voltammograms (CVs) were rectangular in shape even at low scan rates, indicating capacitive behavior (**Supplementary Fig. S6a**). Little reduction in capacitance was observed even for scan rates as high as ~25 V s⁻¹ and ~5 V s⁻¹ in liquid and solid electrolytes, respectively (**Supplementary Fig. S7a,b**). This low dependence of capacitance on scan rate is similar to that for conventional double-layer supercapacitors having much lower volumetric capacitances. Discharge current for the two-ply electrodes was a linear function of potential scan rate at up to ~80 and ~20 V s⁻¹ in liquid and solid electrolyte, respectively (**Fig. 2b**).

The discharge rate capability of the two-ply electrode supercapacitor can also be evaluated from the relaxation time constant (τ_0), which is about 17 ms for the two-ply electrode supercapacitor in the aqueous electrolyte (**Fig. 2c**). This time constant is smaller than that for the onion-like carbon (26 ms) and activated carbon (700 ms) supercapacitors¹⁰. The time constant for solid-electrolyte supercapacitors (**Fig. 2c**) was significantly longer (~80 ms), which can be explained by an increased equivalent series resistance (ESR). This increases ESR is due mostly to an increased equivalent distributed resistance (EDR), which relates to the diffusion of ions within the porous electrode³⁰ (**Fig. 2d**).

High cyclic life and stability of two-ply electrodes in solid electrolyte.

Biscrolled yarn supercapacitors (comprising two-ply anode and cathode separated by solid electrolyte) were stable in ambient atmosphere and showed high cycle life even when bent, helically wound, or woven into a glove. The results in **Figure 3a** show that the desired rectangular dependence of current density on applied potential was retained for scan rates up to 1 V s^{-1} . The **Figure 3b-d** results show little change in capacitance when the capacitor was bent (~2% for 2000 cycles), helically wound on a glass capillary tube (~8% for 10,000 cycles), or woven into a glove (~1% for 10,000 cycles), respectively. Bending did increase

the ESR compared to that for the unbent state, which may limit charge/discharge rates (Supplementary Fig. S8b).

Energy and average power densities for the complete system for commercialization.

For a given constant scan rate v (V s⁻¹), the average power during discharge (P_{av} , in W) was calculated by integrating the current density (I) versus voltage (V) curves:

$$P_{\rm av} = V_{\rm i}^{-1} \int_{V_{\rm i}}^{0} I \cdot V dV, \tag{1}$$

where V_i is the initial voltage during discharge. Integration to obtain average power during charge and discharge provided nearly identical results. The discharged energy (*E*, in Wh) was obtained using equation (3):

$$E = \frac{1}{3600\nu} \int_{V_{\rm i}}^{0} I \cdot V dV \quad . \tag{2}$$

The combination of exceptional energy storage and power densities for the complete supercapacitor (including both electrodes of plied yarn/Pt-wire and solid electrolyte) is indicated in the Ragone plots of **Fig. 4**, which compares performance with that for other micro-size energy storage devices. The yarn supercapacitor (using 20 µm diameter biscrolled yarn) had an average power density of 40 W cm⁻³, which is about four times higher than for commercial 25-mF supercapacitors¹⁰. The biscrolled yarn supercapacitor exhibited energy densities of up to 1.4 mWh cm⁻³, which is over two times higher than for commercial micro-supercapacitors^{10,31}. Although a commercial 500-µAh lithium thin-film battery¹⁰ delivers much high energy, it has three orders of magnitude lower power density than for the biscrolled yarn redox supercapacitor. Also the biscrolled yarn redox supercapacitor has two-orders of magnitude higher power density and a six times higher energy density than for a recent all-solid-state, flexible supercapacitor based on MnO₂-coated carbon fiber.³² These total cell volume comparisons avoid the pitfall²² of studies that report high active-material

energy and power densities by using small amounts of active material on comparatively massive current collectors. By plying the active yarn with current collecting wire and using the nanotubes to provide enhanced electronic connectivity down to the nanoscale, we decrease electrode weight and volume by eliminating the need for a conventional conducting binder, which can also interfere with electrochemical accessibility. Multifunctionality is deployed: the solid electrolyte functions as separator and connector, the scrolled nanotubes provide electrical, electrochemical, and mechanical functions, and the PEDOT provides both energy storage and mechanical reinforcement.

Discussion

Such attractive electrochemical performance is attributed to the good adhesion and resulting low contact resistance between the biscrolled yarn and the Pt wire current collector, as well as the topology of the biscrolled yarn. The strong physical contact caused by compressive forces generated during plying results in a low ESR³³. The ESR measured dramatically decreased from 110.4 to 8.5 ohm (at 1 kHz) when an unplied biscrolled electrode was replaced by an electrode of yarn plied with Pt wire, and this resistance decrease can also be seen in the decrease of the diameter of the Nyquist semi-circle (**Supplementary Fig. S8a**). The latter can be related to the decrease of charge transfer resistance between the yarn and the Pt wire.

We also demonstrated the effect of yarn geometry on supercapacitor performance. **Figure 5a** shows the dependence of volumetric capacitance on yarn diameter for two-ply yarn electrodes measured at 1 V s⁻¹ scan rate are nearly identical for liquid and solid electrolytes, and both monotonically increased by a factor of about 2 upon decreasing yarn diameter from 28 to 20 μ m. The yarn diameter was controlled by the hybrid nanomembrane width and the forest height. Otherwise the hybrid membranes were identically prepared with the same amount of inserted twist per yarn length and all yarns contained about 75 wt%

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PEDOT. The origin of this strong diameter dependence of volumetric capacitance is not completely understood. A number of effects might be important, increasing yarn diameter (while applying the same twist per yarn length) can cause yarn densification that closes pore space between biscrolled sheets and from one biscrolled layer to another, which when filled with electrolyte provide valuable ion transport pathways from yarn surface to yarn center. Increased yarn diameter would also increase ion diffusion distances²³ even if yarn pore structure is unaffected. Additionally, Hertzian contact pressure theory for cylindrical yarns held in parallel or in contact with metal wires of constant diameter, predicts lower contact pressure for large diameter yarns for the same contact force conditions³⁴. Decreased contact pressure can generate increased contact resistance, thereby resulting in an increase of the ESR. In fact, the ESR measured at 1 kHz decreased from 16.2 to 8.5 ohm for two-ply electrode supercapacitors when the diameter of the yarn decreased from 28 to 20 μ m (Fig. 5b). Evaluation of longer lengths of plied yarn electrodes showed only small reduction in capacitance (Supplementary Fig 9), suggesting the potential for scale up, although batch fabrication processes currently limit obtainable lengths.

The explanation for the extraordinary combination of high power and high energy storage capabilities lies in the novel structure of the biscrolled yarn (**Fig. 5c**). Even though there is up to 75 wt% PEDOT in the biscrolled yarn, the scroll structure provides a continuous pathway for both ion and electronic transport from yarn surface to yarn center, as well as sufficient porosity through scroll layers for cross-layer ion transport. The ease of ion transport is enhanced by the gradient in PEDOT concentration through the thickness of the nanomembrane that is twist spun to make the biscrolled yarn. As a result, one side of each of the hundreds of scroll layers in a 20 μ m diameter biscrolled yarn is largely free of PEDOT, and therefore able to provide fast ion transport via incorporated electrolyte.

In conclusion, this method for producing biscrolled yarn electrodes seems amenable to upscaling. Since drawable nanotube forest could be continuously grown on a flexible belt (like a flexible steel foil coated with oxide and catalyst)³⁵ and drawn at high rates into sheet, continuous transfer of nanotubes from forest on one belt to sheet on a second moving belt seems feasible. After liquid-based deposition of catalyst on the nanotube sheet and VPP deposition of the PEDOT on the second moving belt, a delivered liquid could be used to separate the bilayer sheet from the belt substrate for conventional twist insertion to make yarn.

The described technology of gradient biscrolling has produced supercapacitor yarns whose hierarchal structure leads to outstanding energy storage and power performance as a redox supercapacitor. These weavable, sewable, braidable, and knottable yarns are promising for electronic textile applications, since the fabrication process seems upscalable and supercapacitor performance is little affected by bending or weaving. The high strength of the biscrolled yarns means that textiles woven from these yarns could simultaneously provide structural reinforcement and energy storage, such as in automotive panels. On the opposite extreme, it was demonstrated that all-solid-state supercapacitors can be sewn as a single thread into textiles to power micro-electronic devices.

Methods

Materials

Drawable MWNT forests (~400 μ m high and consisting of ~12 nm diameter nanotubes that contain ~9 walls and form large bundles) were grown on a Si wafer by chemical vapor deposition, as previously described²³. Polyvinyl alcohol (PVA, M_w 146,000-186,000), iron(III) p-toluenesulfonate hexahydrate (Fe(III)PTS), pyridine (anhydrous, 99.8% purity), 1-butanol (\geq 99% purity), and 3,4 ethylenedioxythiophene (EDOT, 97% purity) were from Sigma-Aldrich Corporation and 1 M aqueous sulfuric acid solution was from Daejung Chemicals and Metals Company. SEM and AFM images of the PEDOT-infiltrated MWNT sheets are shown in Supplementary Figs. S1 and S2.

PEDOT/MWNT nanomembrane preparation.

A 20 wt% solution of iron (III) p-toluenesulfonate hexahydrate (Fe(III)PTS) in butanol (with 1.6 volume percent of subsequently added pyridine) was used as oxidizing agent for the polymerization of PEDOT (after diluting this stock solution to obtain 4 to 16 wt% of oxidant). The diluted solutions were added drop-wise over a two-layer stack of aligned carbon nanotube aerogel sheet strips that were supported by a glass substrate, and then the sheet stack was dried at 60 °C for 20 minutes. Densification during evaporation of the solution reduced the stack thickness to ~75 nm. VPP to provide PEDOT infiltration within the nanotube host sheets was accomplished by exposing these catalyst-containing sheets to 3,4-ethylenedixoythiophene (EDOT) vapor in a VPP chamber for 1 hour at 60 °C. After the VPP, the PEDOT-coated nanotube sheets were washed three times with ethanol.

Fabrication of yarn electrodes.

Biscrolled yarn was fabricated from a PEDOT-infiltrated nanotube sheet strip using the process illustrated in **Fig. 1c**. The hybrid nanomembrane sheet strips were released from a supporting glass substrate by using a 50/50 by volume ethanol/water solution. While one end of the 7.5 cm long sheet strip remained attached to the substrate, the opposite strip end was peeled from the substrate and attached on an electric motor shaft with a flat rectangular paddle. Then, the nanomembrane was slowly twisted using the motor to introduce ~8000 turns/meter of inserted twist per yarn length (**Supplementary Movie 1**). After twist insertion, both ends of the biscrolled yarn were fixed to the glass slide using carbon tape and the biscrolled yarn was dried at room temperature. Depending upon the thickness and width of the hybrid nanomembrane, the resulting yarn diameter ranged from ~20 to ~35 μ m. Two-ply electrodes for supercapacitors were made by twisting together a PEDOT/MWNT biscrolled

yarn and a ~25 μ m Pt wire using an electric motor (**Supplementary Movie 2**). The electrodes contained 5.7 μ g of active materials (PEDOT and MWNT) per cm length.

Preparation of solid electrolyte.

Water (30 ml) and sulfuric acid (1.67 ml) were mixed using a stirring bar for 10 minutes in a 50 ml glass bottle (1 M sulfuric acid solution). Three grams of PVA (M_w 146,000-186,000) was added in the solution, and then the mixture was stirred for 1 h at ~90 °C. This solution was deposited and dried, as described above, to obtain the solid electrolyte.

Calculation of yarn porosity

The fraction of void space in the yarn that is potentially available for electrolyte filling was calculated by using the following equation:

$$\phi = V_{\rm v}/V_{\rm t} = (V_{\rm t} - V_{\rm m})/V_{\rm t}$$
,

where V_v is the void-space volume, V_t is the total volume of the yarn, V_m (= V_p+V_c) is the volume of materials in the yarn, including PEDOT and MWNT. The MWNT volume (V_c) was calculated by dividing MWNT mass by the density of MWNT bundles (~1.67 g cm⁻³). The PEDOT volume (V_p) was calculated by dividing the PEDOT mass by PEDOT density, using 1.47 g cm⁻³ for the density of PEDOT³⁶. Sheet stack mass was measured before and after PEDOT coating using a microbalance in order to obtain the wt% of PEDOT in the yarn. Calculated porosities are given in **Supplementary Fig. S3**.

Apparatus for characterizations

Surface morphology and height profiles of hybrid nanomembranes were characterized using scanning electron microscopy (Hitachi S4700) and atomic force microscopy (Park Systems XE-100). Cyclic voltammetry and electrochemical impedance spectroscopy (EIS) measurements were obtained using electrochemical analyzers (CHI 627B from CH Instruments and Reference 600 from Gamry Instruments). Mechanical properties of the yarn

micro supercapacitor (**Supplementary Fig. S4**) were characterized using a Thermal Mechanical Analyzer (TMA/SS7100 from RT Instruments).

Construction of supercapacitors

Supercapacitors were constructed using two configurations. For tests in liquid electrolytes, two identically-prepared electrodes were immersed in a beaker of the electrolyte. The electrodes were constructed from biscrolled yarn plied with thin platinum wire. For solid state devices, two such electrodes were first individually coated with 1 M H₂SO₄/PVA gel electrolyte to avoid short circuit in the next step (**Supplementary Fig. S5a**). After a few hours of drying in air, the full two-electrode system was made by plying together the individual electrolyte and thoroughly drying the electrolyte in air, the solid-state supercapacitor could be directly woven into a textile structure without the need of further packaging (**Supplementary Fig. S5b**).

Calculation of electrochemical capacitance for two-ply electrode supercapacitors

Electrochemical characterizations of the supercapacitor yarns are given in **Supplementary Figs. S6-S9**. Cyclic voltammetry curves were obtained at the scan rate from 0.01 V s⁻¹ to 100 V s⁻¹. Electrochemical impedance spectroscopy (EIS) was conducted for frequencies between 100 kHz and 10 mHz using a 5 mV AC voltage and a 0.1 V DC voltage. EIS tests were made in 1 M sulfuric acid electrolyte using a symmetric two-electrode system. Except for the results in **Supplementary Fig. S7**, the capacitance for the two-electrode system (*C*) was derived using the time-integrated current and the voltage scan range from CV curves. For **Supplementary Fig. S7**, *C=1/(dV/dt)*, where *I* is the discharge current. Using these values of *C*, the single-electrode volumetric capacitance (C_{sp}) was calculated from the following equation³⁷:

$$C_{\rm sp} \left(\rm F \, \rm cm^{-3} \right) = 4 \cdot C/V_{\rm vol}, \tag{1}$$

where V_{vol} is the total volume of the electrochemically active material in both electrodes. The factor of 4 multiplier adjusts the capacitance of the two-electrode system and the combined volume of two-electrodes to the capacitance and volume of a single-electrode.

Calculation of the real and imaginary parts of electrochemical capacitance (C' and C'') for two-ply electrode supercapacitors

The frequency dependence of the real and imaginary parts (C' and C'') of electrochemical capacitance was obtained from electrochemical impedance spectroscopy measurements by using the following equations³⁸:

$$C'(\omega) = \frac{-Z'(\omega)}{\omega | Z(\omega) |^{2}}$$
$$C''(\omega) = \frac{Z'(\omega)}{\omega | Z(\omega) |^{2}},$$

where ω is the angular frequency and Z' and Z'' are the real and imaginary parts of the impedance, defined as $Z'(\omega)^2 + Z''(\omega)^2 = |Z(\omega)|^2$.

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Author Contributions

J.A.L., M.K.S., G.M.S., G.G.W., R.H.B. and S.J.K. conceived and designed the experiments. M.D.L. synthesized multiwall carbon nanotube forest and J.A.L., M.K.S., S.H.K., H.U.C. and X.L. carried out the experiments. J.A.L., M.K.S., G.M.S., G.G.W., M.E.K., R.H.B. and S.J.K. wrote the paper. All authors contributed to data analysis and scientific discussion.

Figure Legends

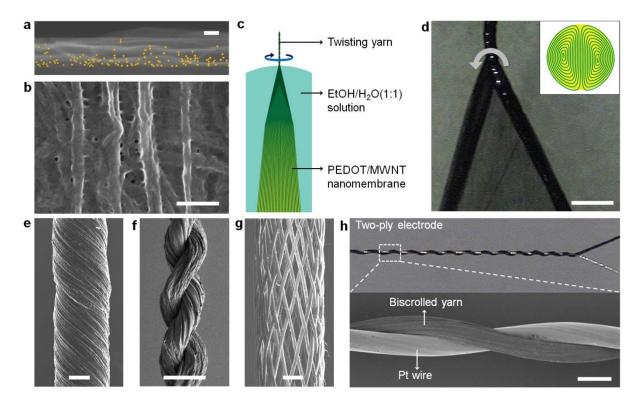


Figure 1. Images showing the biscrolling method and precursor and derived structures for PEDOT/MWNT yarns. (a) An EDAX (Energy Dispersive Analysis by X-ray) map of the cross-section of a PEDOT-infiltrated two-layer carbon nanotube sheet stack, where PEDOT location is indicated by yellow dots signifying that sulfur concentration in above a threshold. The scale bar equals 50 nm. (b) SEM image of the side of a PEDOT-coated MWNTs nanomembrane that was on top during PEDOT deposition. The scale bar equals 200 nm. (c) Schematic illustration showing the fabrication of a biscrolled PEDOT/MWNT yarn. (d) Optical microscope image of the spinning wedge, which shows the wedge edges being twisted to form a dual-Archimedean scroll yarn, which is schematically illustrated in the inset. The scale bar equals 200 μ m. (c) SEM images of a biscrolled yarn with ~37° bias angle. The scale bar equals 10 μ m. ; (f) SEM image of two PEDOT/MWNT yarns plied together. The scale bar equals 50 μ m. ; (g) SEM image of a braided structure containing 32 biscrolled yarns. The scale bar equals 100 μ m. ; and (h) Two SEM images of a PEDOT/MWNT biscrolled yarn that is plied with a 25 μ m Pt wire. The scale bar equals 40 μ m.

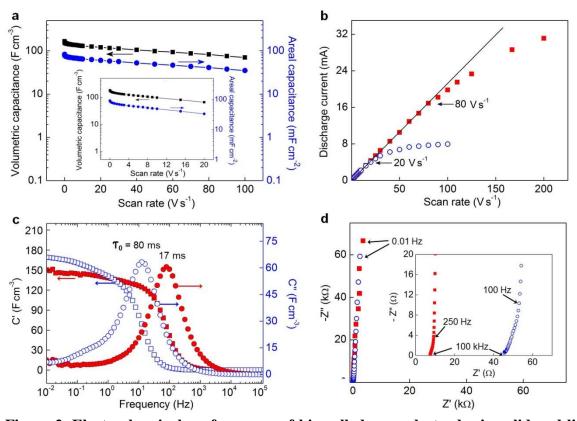


Figure 2. Electrochemical performance of biscrolled yarn electrodes in solid and liquid electrolytes: (a) Volumetric and areal capacitances as a function of scan rate for two-ply yarn/wire supercapacitor electrodes (20 μ m yarn diameter) in (a) liquid electrolyte, and **inset**, solid electrolyte, respectively. (b) Discharge current versus voltage scan rate for a two-ply supercapacitor electrode. The linear dependence of discharge current on scan rate up to high scan rates indicates ultra-high power ability in both liquid (filled red squares) and solid (open blue circles) electrolytes. (c) Frequency dependence of the real and imaginary parts (C' and C'') of the volumetric capacitance of a two-ply electrode supercapacitor using liquid (filled red circles) and solid (open blue circles) electrolytes. (d) Nyquist plots for two-ply yarn electrodes in liquid (filled red squares) and solid (open blue circles) electrolyte, showing the imaginary part versus the real part of impedance. Inset: magnified diagram for high frequencies.

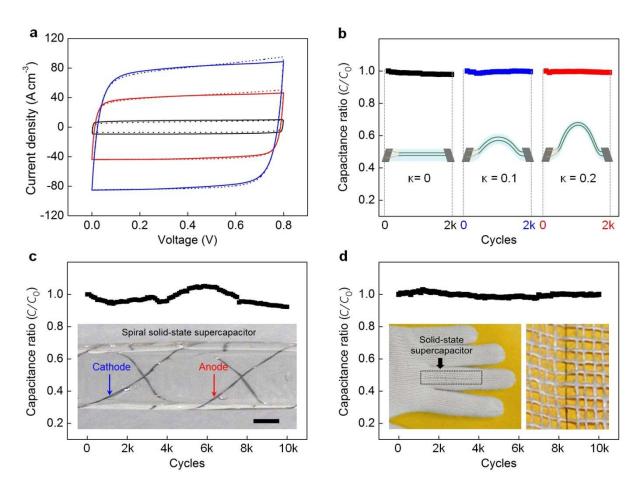


Figure 3. Cyclic life and stability of solid-electrolyte, biscrolled yarn supercapacitor:

(a) Cyclic voltammetry at different voltage scan rates of 0.1 V s⁻¹(black line), 0.5 V s⁻¹(red line), and 1 V s⁻¹(blue line) for the solid-electrolyte, biscrolled yarn supercapacitor in ambient (solid line). Exposure of the supercapacitor in ambient for three months (dotted line) did not significantly change in performance. Normalized capacitance in ambient (C/C_o, where C_o is the initial capacitance) versus cycle number for a two-ply, all-solid-state Pt/yarn supercapacitor after deformation by (**b**) bending on a flexible polyethylene terephthalate film (κ is curvature in mm⁻¹); (**c**) winding on a 1.4 mm diameter glass tube. The scale bar equals 500 µm. (the inset shows a photograph taken after solid electrolyte coating); and (**d**) weaving into a glove (the yarn supercapacitor was 5 cm long).

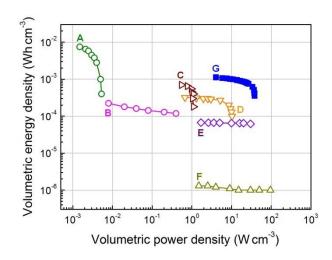


Figure 4. Volumetric energy and average power densities for complete supercapacitors: Ragone plots comparing volumetric energy and average power densities for the present allsolid-state, two-ply electrode supercapacitor (G) (yarn diameter: 20 μ m) with that for a commercially available 4V/500 μ Ah Li thin-film battery¹⁰, a 2.75 V/44 mF activated carbon electrochemical capacitor³¹ (C), a 3.5 V/25 mF supercapacitor¹⁰ (D), a 63 V/220 μ F electrolytic capacitor¹⁰ (E), a 3 V/300 μ F Al electrolytic capacitor³¹ (F), as well as an experimental MnO₂ carbon fiber supercapacitor³² (B).

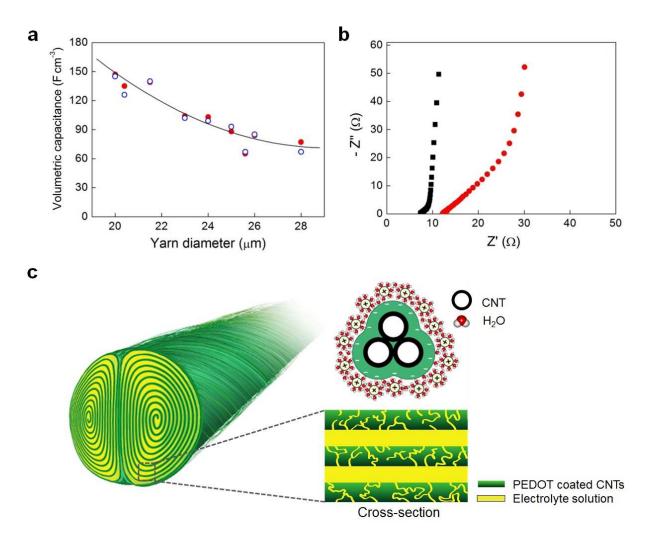


Figure 5. Yarn diameter effects and structure for PEDOT/MWNT yarn supercapacitors: (a) The dependence of volumetric capacitance (for a scan rate of 1 V s^{-1}) on yarn diameter for a two-ply electrode supercapacitor in solid (open blue circles) and liquid (filled red circles) electrolytes. (b) Nyquist plots for two-ply yarn/wire electrodes using different diameter PEDOT/MWNT yarn. The equivalent series resistance (ESR) measured at 1 kHz for a 20 µm diameter yarn (filled black squares) is 8.5 Ω , which is ~2 times smaller than that of a 28 µm diameter yarn (filled red circles) at this frequency (16.2 Ω). (c) Schematic illustration of the pore structure of MWNT/PEDOT biscrolled yarns and the electrochemical double layer of a PEDOT coated MWNT bundle.