

VIBRATIONAL SPECTRA AND STRUCTURE OF UNDOPED AND DOPED POLYPARAPHENYLENE

Z. Iqbal, H. Bill, R. Baughman

► To cite this version:

Z. Iqbal, H. Bill, R. Baughman. VIBRATIONAL SPECTRA AND STRUCTURE OF UNDOPED AND DOPED POLYPARAPHENYLENE. Journal de Physique Colloques, 1983, 44 (C3), pp.C3-761-C3-764. 10.1051/jphyscol:19833151. jpa-00222666

HAL Id: jpa-00222666 https://hal.archives-ouvertes.fr/jpa-00222666

Submitted on 1 Jan 1983

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers. L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

VIBRATIONAL SPECTRA AND STRUCTURE OF UNDOPED AND DOPED POLYPARAPHENY-LENE

Z. Iqbal**, H. Bill** and R.H. Baughman***

*Institute of Inorganic Chemistry, University of Zürich, Zürich, Switzerland **Department of Physical Chemistry, University of Geneva, Geneva, Switzerland ***Corporate Research Center, Allied Chemical Corporation, Morristown, New Jersey, U.S.A.

<u>Résumé</u>- Nous présentons des résultats par spectroscopie infrarouge et Raman sur du poly(paraphénylène) (PPP) non dopé et dopé avec AsF₅. Les résultats sont interprétés à la lumière des calculs de modes normaux de Rakovic et al. et du modèle par défauts de type bipolaron du polymère dopé proposé par Brédas et al.

Abstract- Infrared and Raman spectroscopic data on undoped and AsF₅-doped polyparaphenylene (PPP), are presented. The results are discussed in terms of the normal mode calculation of Rakovic et al and the bipolaron defect model of the doped polymer proposed by Bredas et al.

Introduction- In recent years a number of conjugated organic polymers have been discovered which undergo a transition toaconductive state on doping with electron acceptors or donors. The most recently discovered member of this group is polyparaphenylene (PPP) /1,2/, which shows a conductivity increase as high as 18 orders of magnitude/3/ on doping. The demonstration that rechargeable batteries based on doped polyacetylene (PA) and PPP are feasible /4,5/, have further stimulated both fundamental and applicative interest in these polymers.

Doped PPP and related polymers show behaviour reminiscent of conventional semiconductors like silicon. However, the presence of highly anisotropic interactions would lead to low dimensionality of electron motion and collective instabilities such as the Peierls transition would occur. In addition disorder initiated during the doping process would give rise to localized electronic states which would affect tranport phenomena in these materials. Structurally, topological kinks or so-called solitons are created on doping, and strong evidence has been put forward that in doped all-trans-PA transport occurs via spinless charged solitons /6,7/. In PPP, contrary to trans-PA, the ground state consists of the non-degenerate benzoid and quinoid forms. PPP is, therefore, not expected to accomodate solitons; nevertheless, the transport properties of doped trans-PA and PPP are very similar. A theoretical study of the problem by Bredas, Chance and Silbey/8/has rationalized this situation by showing that polaron defect formation occurs at low doping levels both in trans-PA and PPP, but at higher doping levels the defects interact to give uncorrelated charged solitons in PA and spinless bipolarons in PPP.

In this paper we present data obtained via infrared and Raman spectros-

+address of Z.I. ca. mid-1983: Corporate Research Center, Allied Chemical Corporation, Morristown, N.J., USA copy, on doped and undoped PPP, and suggest an interpretation which would be relevant to the structural nature of doping in PPP. In what follows, we present first a structural summary and some details regarding the vibrational model of Rakovic et al /9/ for PPP; we then proceed to our experimental results and a discussion and interpretation of the data.

<u>Theoretical and Structural Considerations</u> - PPP consists of benzene rings linked in the para-position. Based on crystallographic data on oligomers , the C-C bond lengths within the rings are 1.4 Å and between the rings 1.50 Å. In the solid two successive rings are tilted with respect to each other by 23° - the result of a compromise among effects associated with conjugation, crystal packing and steric repulsion of orthohydrogen atoms. To a first approximation then, the C-atom network of PPP is isostructural to that of graphite. On doping, it has been shown by Bredas et al /8/ via an adiabatic Hückel calculation that a (spinless) quinoid-bipolaron

with a defect extension of 5 rings, can result. The strong deformation of the rings in the bipolaron pushes the bonding state into the gap and conductivity could arise from the motion of these spinless bipolarons.

The vibrational model of Rakovic et al /9/ for pristine PPP on which we base some of our following interpretations assumes a planar PPP periodic structure and a valence force field transferred from the toluene molecule. In terms of line groups, the symmetry of the PPP model is L2/mmm and modes can be assigned to irreducible one dimensional representations A, B, A₁ and B₁. Each representation decomposes into a + and a - component, distinguishing those which are even and odd with respect to the plane orthogonal to the benzoid ring. At k = 0, the A₀, A₁ and B₁ components are infrared active with eigenvectors parallel to the y (chain direction), x and z (perpendicular to the benzoid plane) axes respectively, while the A₀, B₀, A₁ and B₁ components are Raman active. Thus the Raman and infrared active modes in pristine PPP are expected by symmetry to be mutually exclusive.

Experimental details, results and discussion - PPP used in the pres-ent study was prepared at Allied Chemical by the oxidative cationic polymerization of benzene. Acceptor doping was accomplished by exposing PPP pellets to 450 torr AsF, while donor doping was achieved via reaction of PPP with alkali metal napthalide in tetrahydrofuran. Furdetails are to be found in refs /1, 2/. The infrared spectra were measured with a Perkin Elmer 983 spectrometer which was purged for a minimum of 24 hrs with dry air before each measurement. The sample chamber was in addition flushed with pure N, during each measurement. The Raman measurements were made using a Spéx double monochromator and photon counting detection. The infrared spectra of pure PPP were measured in the specular reflection geometry using pressed pellets of the polymer. Raman measurements were made on the same samples. Measurements on the doped polymer were made on samples sealed in an inert atmosphere in glass capillaries. For infrared measurements, the sealed capillaries were opened in a glove box, quickly mixed with dried KBr and pelleted in a vacuum of ca. 0.1 torr, and then rapidly transferred to the spectrometer sampling chamber.

The infrared spectra of PPP (in reflection) and AsF_5 and PF_5 doped PPP (in absorption) are shown in Fig. 1 together with the calculated frequencies/9/. The Raman spectrum of AsF_5 doped PPP is displayed in Fig. 2. Sizable flourescence background was observed from pure PPP samples, precluding a careful study of its Raman spectrum. The peaks which could be separated from the background are listed in Table 1

~ . ~ .	C3	-7	63
---------	----	----	----

together with		Tabl			
those observed		frequencies (cm ⁻¹)		$PPP:AsF_(cm^{-1})$	
independently by Lefrant (private communication).	IR	Raman This work	Lefrant	IR	Raman
Also listed in Table 1 are the IR		1643 1606	1608	1635	
and Raman freq- encies in AsF ₅ - doped PPP. Att- empts to obtain	1470	1570		1580 1520 1480 1470	1595
donor doped samp- les have failed	1400	1332	1337	1400 1340 1270	1279
so far, while IR measurements on these samples were not attempted because of their extreme sensitiv- ity to air.	1000 800		1252.	1270 1240 1180 1085 1000 990 830 805	1278
The three main TR lines at 1470,1000 and 800 cm ⁻¹ in pure PPP can be	725			742 700 660 484	
assigned to A_0^{+}, A_1^{+} and B_1^{+} modes in- volving largely					328 277 218

stretching, CCC stretching + CH bend and CCC deformation motions, and are in good agreement with the calculated frequencies. In the Raman spectra lines in the 1600 cm⁻¹ region, and between 1200 and 1350 cm⁻¹ can be assigned to the symmetric stretching (A⁺ and A⁻) modes of the polymer backbone which distort PPP towards the biradical (1600 cm⁻¹) and quinoid (1300 cm⁻¹) structures. The observed frequencies are again in good agreement with the calculated frequencies. This agreement with the calculated frequencies that the toluene molecule suggests that the charge density in PPP is largely localized within the benzoid rings.

The IR spectra of AsF, and PF, doped PPP show a large number of absorptions in the 2000-600 cm region. These are buried under the free-carrier continuum in "over-doped"samples,as has been observed previously/1/ The dominant feature is the appearance of two broad and strong absor-ptions at 1520 and 1180 cm⁻¹. In addition maxima are observed near 1580, 1270 and 1240cm⁻¹ both in the IR and Raman spectra of PPP:AsF₅, suggesting a breakdown of the mutual exclusivity of the IR and Raman modes as a result of the statistical nature of defect formation.At the defect site itself localized charge oscillation would make the defect modes strongly infrared-active, consistent with our observations. Also, the downward frequency shifts of the stretching Raman-active modes in the doped samples, suggests increased delocalization of the charge-density probably induced by a quinoid-like deformation of the chain as suggestby the model of Bredas et al/8/.The narrow bands at 1480,1000 and 800 $_{
m 1}$ correspond to the undoped polymer regions and those below 700 cm сm are most likely due to AsF, ions. In conclusion then; the results ind-icate a signature of the défect in doped PPP consistent with a quinoid -like distortion of the chain-the detailed assignments have however to await a calculation of the defect mode frequencies.

REFERENCES - 1. SHACKLEMEE. L(1949) C308., IVORY D., MILLER G., BAUGHMAN R.H.,



2.SHACKLETTE L., ELSENBAUMER R., CHANCE R., ECKHARDT H., FROMMER J. and BAUGHMAN R.H., J.Chem.Phys. 75(1981)1919 3.IVORY D., MILLLER G., SOWA J., SHACKLETTE L., CHANCE R. and BAUGHMAN R.H., J.Chem.Phys. 71(1979)1506 4.McINNES D., DRUY M.A., NIGREY P., NAIRNS D., MACDIARMID A. and HEEGER A., J.Chem.Soc.Chem. Commun.(1981)317 5. SHACKLETTE L., ELSENBAUMER R., SOWA J., CHANCE R., IVORY D., MILLER G and BAUGHMAN R.H., J.Chem.Soc.Chem.Commun.(1982)361 6.SU W., SCHRIEFFER J. and HEEGER A., Phys.Rev.Lett. 42(1979)1698 7.IKEHATA S., KAUFER J., WOERNER T., PRON A., DRUY M., SIVAK A., HEEGER A., MACDIARMID A.G., Phys. Rev. Lett. 45(1980)1123 8.BREDAS J.L., CHANCE R.R. and SILBEY R., Phys.Rev. B26(1982)5843 9.RAKOVIC D., BOZOVIC I., STEPANYAN S. and GRIBOV L., Solid State Comm. 43(1982)127