Visible color-center laser in diamond

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Laser action at 530 nm using H3 centers in diamond was observed with an efficiency of 13.5% at room temperature. Optical double-resonance experiments on N3 centers provided direct evidence for a 2A metastable level. Its presence results in rapid decay of the excited 2E state and low quantum efficiency as well as significant excited-state absorption in the N3 luminescence region.

Since the pioneering spectroscopy on $F_A(II)$ centers in KCl by Fritz et al., which led to construction of the first color-center laser by Fritz and Menke,2 many efficient, widely tunable sources of coherent radiation based on defect structures in alkali halides have been developed.³ These ionic crystals have hitherto required cooling to cryogenic temperatures, however, both to prolong the shelf-life of laser-active centers and to achieve high quantum efficiency. While stable, tunable laser operation and Q switching using F-aggregate centers in LiF at room temperature have been achieved recently with special techniques,⁴ in the work reported in this Letter we have followed a different approach to secure hightemperature operation. We have investigated substitutional impurities in a covalent crystal and have obtained efficient, visible laser action using H3 centers in diamond at room temperature.

Diamonds contain a variety of color centers owing to the presence of impurity atom and vacancy complexes.⁵ Nitrogen is the most common impurity of natural and synthetic diamond, entering the face-centered cubic lattice substitutionally and often occurring with concentrations in the paired (A-aggregate) form as high as 0.1%. Two centers incorporating nitrogen are known to be thermally stable and exhibit high quantum efficiency up to temperatures in excess of 500 K. These centers are designated N3 and H3. They have $C_{3\nu}$ and $C_{2\nu}$ point symmetries, respectively. Their structures^{6,7} are pictured, together with the corresponding absorption and emission spectra, in Figs. 1 and 2. Although it is present in many natural crystals, the N3 center is difficult to produce artificially. The H3 center, on the other hand, may be formed abundantly by irradiation with 1-2-MeV electrons to doses of 10¹⁸-10²⁰ electrons/cm² and annealing at 1200 K for 20 h in vacuum.⁵ The exceptional properties of these centers, together with the high thermal conductivity,8 excellent hardness, and high damage threshold of the host crystal, contributed to our initial interest in the laser potential of diamond color centers.

Fluorescence linewidths and decay kinetics were measured in a total of 10 diamond samples to permit a comparison, given in Table 1, of gain coefficients expected for H3 and N3 centers in diamond with that of a representative center in alkali halides, namely, F_2^+ : KCl at 77 K. Fluorescence spectra were recorded using

chopped ultraviolet/visible excitation from an argon-ion laser and phase-sensitive detection with a photomultiplier to yield the center wavelength λ_0 and width $\Delta\nu$ of the diamond-center emission bands. Tunable excitation pulses of $\tau_p=7$ -nsec duration were generated at $\lambda=370$ nm for N3 decay studies by sum mixing in KD*P of 1.06- μ m radiation with the output at 570 nm from a R6G dye laser. H3-center kinetics were studied with 4-nsec pulses from a Coumarin 480 dye laser. Only the longest of the observed sample-dependent fluorescence decay times $\tau_{\rm fl}$ are included in Table 1. Reduced decay times as well as lowered quantum efficiency η

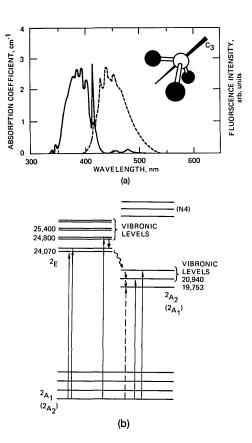
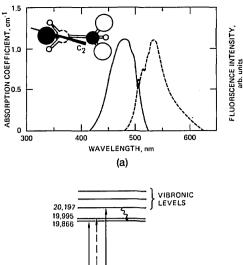


Fig. 1. (a) Absorption and emission spectra and structural model of the N3 center. Filled circles, nitrogen; open circles, carbon. (b) Energy levels of the N3 center.



(b)

Fig. 2. (a) Absorption and emission spectra and structural model of the H3 center. Dashed circle, vacancy. (b) Energy levels of the H3 center.

were found to be correlated with the concentration of nitrogen-pair impurities as reported in Refs. 6 and 7.

From Table 1 it is evident that the room-temperature gain coefficient for H3 in the green is only 17.5 times smaller than that for F_2 ⁺:KCl in the infrared at liquid-nitrogen temperature. The calculated gain for N3 centers is 350 times smaller.

Two diamond samples were prepared as laser crystals with plane-parallel surfaces. The first was 0.175 cm thick and, while containing no measurable concentration of H3 centers, showed a peak N3 absorption coefficient of 1.37 cm $^{-1}$ at 416.4 nm and a concentration of nitrogen pairs estimated from the 4.04-eV absorption peak to be roughly 0.03%. This sample revealed excited-state absorption (ESA), as depicted in Fig. 3. Transmission of a weak cw Ar $^{+}$ laser beam was monitored at several wavelengths in the N3 emission band from 450 to 515 nm, during and after pulsed excitation at $\lambda=370$ nm. Transmission decreased rapidly as

shown in Fig. 3(a) in proportion to the pump intensity [Fig. 3(b)] with a characteristic time of $\tau_1 = 25$ nsec > τ_p . Recovery occurred only after $\tau_2 = 0.4$ msec [Fig. 3(a)]. As explained below, this behavior can only be due to nonradiative decay to a metastable excited state followed by absorption of a second photon and not to ESA from the 2E state itself, which would have appeared with a rise time τ_p and decayed in a time $\tau_{\rm fl}$.

The measured fluorescence lifetime of the 2E state in this sample was 16 nsec, consistent with the calculated decay time $\tau^{-1} = \tau_1^{-1} + \tau_n^{-1}$, which gives 21 nsec when $\tau_n = \tau_{\rm fl}/\eta = 141$ nsec is used for the radiative lifetime. Also, the measured relaxation time of the transient absorption τ_2 greatly exceeded the 2E fluorescence lifetime. These two results showed (1) that decay occurred in a time τ_1 to an excited state below 2E , shortening the 2E population-decay time from the purely radiative rate τ_n to $\tau_{\rm fl}$, and (2) that the level $[^2A$ in Fig. 1(b)] was metastable, with a lifetime of τ_2 . The

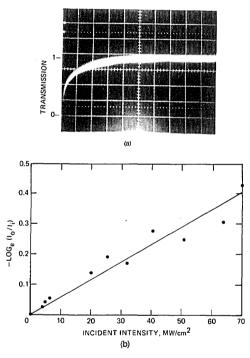


Fig. 3. (a) Sample transmission following pulsed excitation of the 2E level of the N3 center (0.5 msec/division). The time origin coincides with the first small division. (b) Intensity dependence of the N3 excited-state absorption coefficient.

Table 1. Comparison of Diamond and KCl Color-Center Gain Coefficients γ for Constant Inversion Density $N=10^{16}~{
m cm}^{-3}$

Color Center	n^a	T (K)	$ au_{ m fl}$ (nsec)	η	$\Delta \nu \ (10^{13} {\rm sec}^{-1})$	λ ₀ (μm)	$\gamma \text{ (cm}^{-1})$
N3	2.4468^{b}	295	41	0.29^{c}	8.40	0.445	0.010
H 3	2.4262^{b}	295	16	0.95^{d}	5.27	0.531	0.201
F ₂ +	1.4841e	77	$ au_{ m fl}/\eta = 80^e$		1.69	1.680	3.52

a n is the refractive index.

^b Ref. 10.

^c Ref. 6.

 $[^]d$ Ref. 7.

e Ref. 3.

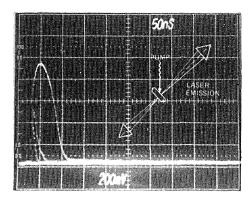


Fig. 4. H3 color-center laser pumping geometry and output-pulse shapes near threshold. Multiple traces are shown.

population that decayed to 2A gave rise to ESA to higher-lying levels. Although the appearance of ESA dimmed the prospect for cw N3 laser action, this double-resonance experiment revealed the $^2E-^2A$ relaxation rate and the nonradiative decay time of the 2A level itself. Previous results on enhancement of ESR signals during optical excitation of N3 centers did not probe this level directly nor temporally resolve the relaxation.

The second polished crystal contained H3 centers with a peak absorption of 1.05 cm⁻¹ at 480 nm. In transmission, surface flatness along the 100 crystal planes was only $\lambda/2$ over several millimeters, and the polished surfaces had a measured wedge angle of 0.36 deg. Nevertheless, laser action at 530 nm occurred owing to 18% Fresnel reflectivity at the uncoated diamond interfaces for pump intensities exceeding 30 MW/cm² at 490 nm, as indicated in Fig. 4. A sharp threshold was observed for emission in two beams that emerged perpendicular to the diamond faces with a divergence angle of 70 mrad. This emission was not due to amplified spontaneous emission since it was not coaxial with the pump beam in either the forward or the backward direction and had a divergence angle much smaller than that determined by considering extreme rays through the pumped region (460 mrad). Similarly, stimulated Raman scattering (SRS) was ruled out on the basis of beam direction, pulse width ($>\tau_p$ in Fig. 4), constant emission wavelength for excitation varying from 486-494 nm, and a threshold intensity well below that previously determined⁹ for SRS.

From the sample thickness, reflectivity, and the parameters of Table 1, the inversion density at threshold was calculated to be $\Delta N = 4.2 \times 10^{17}$ cm⁻³, where we have ignored losses resulting from misalignment, surface scattering, diffraction, and internal reabsorption. The minimum pump intensity necessary to achieve this ΔN within the excitation interval was estimated from $I_t = \Delta Nh\nu V/A\tau_p$ to be 8 MW/cm² using the measured crystal thickness V/A = 0.185 cm. Considering the neglect of losses other than reflectivity, this is in reasonable agreement with the observed value. If losses were reduced to 4% per pass by using an external reso-

nator, threshold inversion would be $\Delta N=9.8\times 10^{15}~\rm cm^{-3}$. Then Ar⁺ laser excitation at 488 nm in a 20- μm focal spot diameter should achieve continuous-wave threshold for a calculated incident power of 0.14 W. In the present pulsed oscillator, the output energy obtained for 1.27 mJ of absorbed input energy was 171 μJ , giving an efficiency of 13.5% just above threshold.

To summarize, we have verified the postulated energy-level structure of N3 centers in diamond and measured excited-state kinetics that leave little prospect for cw laser action using this structure. However, H3 centers have been shown to exhibit efficient, pulsed laser action at room temperature and should yield a cw solid-state laser tunable from 500 to 600 nm. Other defect structures based on substitutional atoms in covalent materials, including other color centers in diamond, can be expected to furnish stable color-center lasers operating at high temperatures.

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Note added in proof: J. A. Van Wyk has furnished evidence [J. Phys. C 15, L981 (1982)] that the shared carbon atom of the N3 structure shown in Fig. 1(a) may be a vacancy.

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