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Wannier-like functions and tight-binding parametrization for the manganese bands in CaMnO_3

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Abstract

We study the electronic band structure of CaMnO_3 , in order to understand the origin of the dispersion of the $\text{Mn}(e_g)$ bands, which is in contrast with the predicted dispersionless bands within the Anderson–Hasegawa double-exchange model with infinite Hund’s-rule coupling. A downfolding technique within the newly developed muffin-tin orbital-based method is used to analyse the density-functional band structure obtained in the local spin density approximation. The finite Hund’s coupling parameter in realistic situations allows the same-spin bands on the two manganese sublattices to mix producing a large dispersion. The calculated Wannier functions for the $\text{Mn}(e_g)$ bands also show large oxygen character at sites further away from nearest oxygen sites causing long-ranged Mn–Mn hopping processes.

1. Introduction

Knowledge about perovskite manganese oxides dates back to 1950s, with the pioneering work of Jonker and van Santen [1] establishing the close relation between the magnetism and electrical transport. The discovery of a very large negative magnetoresistance, namely the colossal magnetoresistance (CMR), has resulted in a great deal of attention being paid to these perovskite materials in recent years, particularly in the light of the exchange mechanism, and the manner in which charge, spin, and lattice degrees of freedom are coupled [2]. $\text{La}_{1-x}\text{D}_x\text{MnO}_3$ (D being a divalent alkaline earth: Ca, Sr, or Ba) systems provide a prime example of such class of compounds. The end-members ($x = 0$ and 1) are antiferromagnetic (AF) insulators. With increase in x , they become ferromagnetic (FM) with well-defined Curie temperatures. Around the same temperature, they also exhibit metal-like conductivity. This simultaneous occurrence of ferromagnetism and conductivity in these doped compounds was interpreted quite early on as being caused by the *double-exchange* mechanism [3], where the magnetic coupling between localized spins on neighbouring atoms is mediated via the conduction electrons connecting the parallel alignment of Mn moments with the hopping of carriers. Many of the studies are

therefore based on the Anderson–Hasegawa double-exchange model [4] with infinite Hund’s-rule coupling, J_H , giving rise to an effective hopping integral for neighbouring spins at the angle θ of $-t \cos(\theta/2)$ (t is the nearest-neighbour (NN) hopping integral for hopping between orbitals of same spin), implying zero kinetic energy gain in the AF Néel state and maximum kinetic energy gain in the FM state.

To investigate the scope and limitations of such commonly used infinite-Hund’s-rule-coupling models, we have considered in the present paper a particular end-member of this family of compounds, namely the AF CaMnO_3 . We have studied its electronic structure in terms of a first-principles Wannier-like description of the Mn e_g bands using the newly formulated and implemented NMTO technique [5]. This compound deserves attention in its own right, being a simple case of an AF compound which can be electron doped. CaMnO_3 , being less prone to chemical defects than LaMnO_3 [6], provides an ideal subject for detailed doping studies. Furthermore, recent conjectures about magnetic polaron formation [7–9] in these compounds with low levels of electron doping has attracted new interest. Our study provides Wannier functions and the corresponding hopping integrals of the tight-binding (TB) Mn e_g bands derived from *ab initio* density-functional calculations. This should prove useful for realistic modelling of the manganites.

2. Crystal structure

The structure of CaMnO_3 is relatively simple compared to those of other members of the $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$ family. It forms in the cubic perovskite ABX_3 structure, with a lattice constant of 3.73 Å [10]. Each Mn atom is sixfold coordinated by oxygen atoms, thereby providing the octahedral environment at the Mn site, resulting in the splitting of the degenerate d levels into t_{2g} and e_g states. Considering the nominal chemical formula of $\text{Ca}^{2+}\text{Mn}^{4+}\text{O}_3^{2-}$ and the fact that Mn^{4+} is not a Jahn–Teller-active ion, the Jahn–Teller distortion of the MnO octahedron observed in other members of the family is absent in CaMnO_3 . Ca ions are twelvefold coordinated by oxygen atoms lying along its $\langle 110 \rangle$ directions. The AF insulator CaMnO_3 is observed to be in the G-type AF state, so the spins at neighbouring Mn sites are antiferromagnetically coupled in all three directions. Figure 1 shows the AF structure of CaMnO_3 .

3. LDA band structure

The LDA calculation for AF CaMnO_3 has been carried out with linear muffin-tin orbital (LMTO) method within the framework of the atomic-sphere approximation (ASA) [11], with two formula units to account for the G-type magnetic unit cell. We have used the TB form of the LMTO method for the present calculation, as discussed in [12]. The orbital-projected AF band structure is shown in figure 2. The calculated band structure agrees with the previous calculations [13, 14]³. In figure 2 (and also in figure 3 discussed below), the k -points correspond to the BZ of face-centred cubic (FCC) structure appropriate for AF CaMnO_3 , namely, $\Gamma = (0, 0, 0)$, $X = (2\pi/a)\left(\frac{1}{2\sqrt{2}}, \frac{1}{2\sqrt{2}}, 0\right)$, $W = (2\pi/a)\left(\frac{3}{4\sqrt{2}}, \frac{1}{4\sqrt{2}}, 0\right)$, and $L = (2\pi/a)\left(\frac{1}{2\sqrt{2}}, 0, 1/4\right)$ with the x - and y -coordinate axes chosen along the diagonal Mn–Mn direction and the z -axes chosen along the perpendicular Mn–O–Mn direction (as shown in figure 1); a is the cubic lattice constant equal to 3.73 Å. Shown in the figure are projections

³ In this paper, the W point for AF CaMnO_3 was different from the conventional W point used in the present paper. Also, the X point was approached from Γ using a direction different from the Δ one. The k -points used were $X = (2\pi/a)\left(0, \frac{1}{\sqrt{2}}, 0\right)$, $W = (2\pi/a)\left(\frac{1}{4\sqrt{2}}, \frac{5}{4\sqrt{2}}, -1/4\right)$, and $L = (2\pi/a)\left(\frac{1}{2\sqrt{2}}, \frac{1}{\sqrt{2}}, 0\right)$.

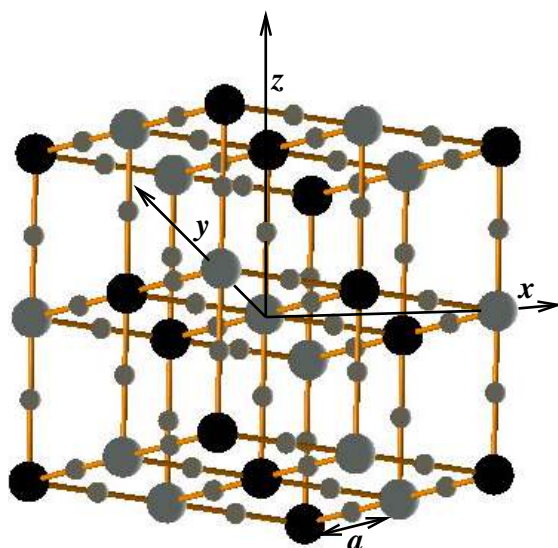


Figure 1. The G-type AF structure of CaMnO_3 . Large black and grey balls denote the Mn atoms belonging to two different AF sublattices, while the small balls denote the oxygen sites. Ca sites have not been shown, for clarity.

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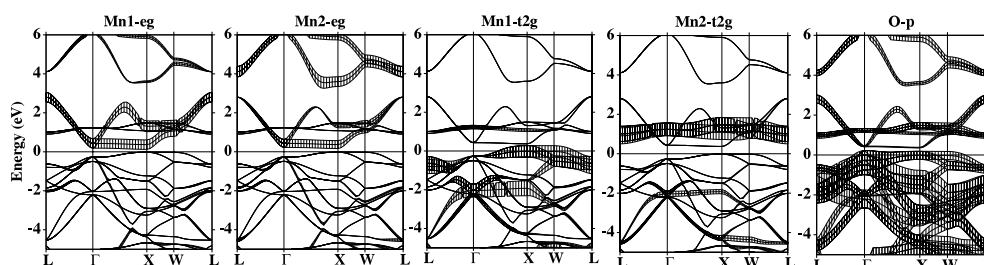


Figure 2. The orbital-decomposed band structure of AF CaMnO_3 .

to e_g -, t_{2g} -derived states at the two Mn sites (denoted as Mn1 and Mn2) belonging to the two different sublattices of the AF structure and that of the oxygen p states. The width of the bands in each panel is proportional to the weight of the indicated orbital in the wavefunction. By AF construction, the state in the up-spin channel at the Mn1 site is equivalent to that in the down-spin channel at the Mn2 site and vice versa. While e_g -derived states at the Mn1 site span the energy range of about 0.2–2.5 eV, the e_g -derived states at the Mn2 site span the energy range of about 3.5–6 eV. The t_{2g} -derived states at the Mn2 site appear at an energy of about 1.5 eV, in the middle of the range of the e_g -derived Mn1 states. The significant feature to notice is the appreciable dispersion of the e_g -derived bands and the non-negligible oxygen contributions in these bands. We note that the Anderson–Hasegawa model of infinite Hund’s-rule coupling with effective hopping of $-t \cos(\theta/2)$ would give rise to dispersionless Mn e_g bands, since the term $-t \cos(\theta/2)$ would prohibit any processes of hopping between spins that are antiferromagnetically aligned.

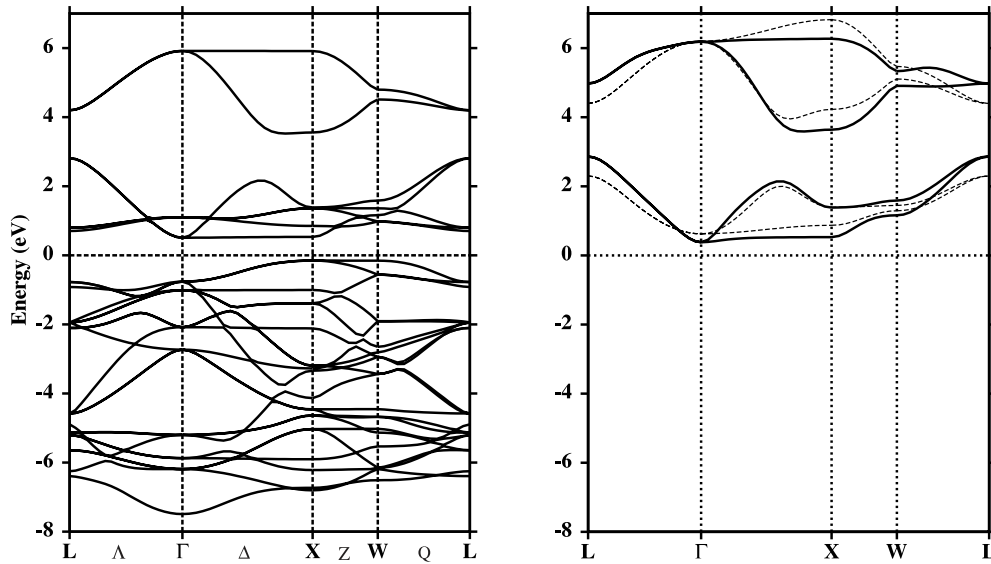


Figure 3. The full LDA band structure (left panel) and the downfolded Mn e_g -only bands (right panel) for AF CaMnO_3 . The right panel also shows, as dashed lines, the best-fit TB bands for first- and second-NN hoppings, with the parameters given in table 1.

4. Downfolding: Wannier-like orbitals and the tight-binding Hamiltonian

In recent years, the downfolding technique within the framework of an improved LMTO method, namely the N th-order MTO (NMTO) method [5], has been developed, a method that goes beyond the standard LMTO method in describing an energetically accurate basis set with a consistent description throughout the space of MT spheres and the interstitials. The new method provides a useful way of deriving the few-band Hamiltonians starting from complicated LDA band structure. It has been successfully applied to systems such as low-dimensional magnetic compounds [15, 16], high- T_c compounds [17], and double perovskites [18]. This method, which relies on integrating out the other degrees of freedom, keeping only the relevant degrees of freedom or degrees of freedom of interest, naturally takes into account the renormalization effect due to the integrated orbitals in the effective orbitals defining the few-band Hamiltonian. Fourier transformation of this few-orbital downfolded Hamiltonian provides the hopping matrix elements of the corresponding TB Hamiltonian defined in the effective orbital basis, while the effective orbitals themselves provide the Wannier-like functions of the corresponding bands. It is to be noted here that this method provides a first-principles way of deriving the single-particle model Hamiltonian and directly generating Wannier-like functions without any fitting procedure. This way of generating Wannier-like functions may be contrasted with that starting from the Bloch-function outputs of some standard band-structure method and then constructing them demanding that the maximum localization property be exhibited [19].

In the present study we have applied this technique to derive an effective Hamiltonian with only the Mn(e_g) orbitals, which will help with the understanding of the origin of the appreciable dispersional width of the Mn e_g bands as provided by LDA calculations bringing in the important interaction pathways. In the right-hand panel of figure 3 we show (as solid curves) the bands obtained by downfolding all the other channels except the Mn e_g channels. For comparison, in the left-hand panel of figure 3 the full LDA band structure is shown. We

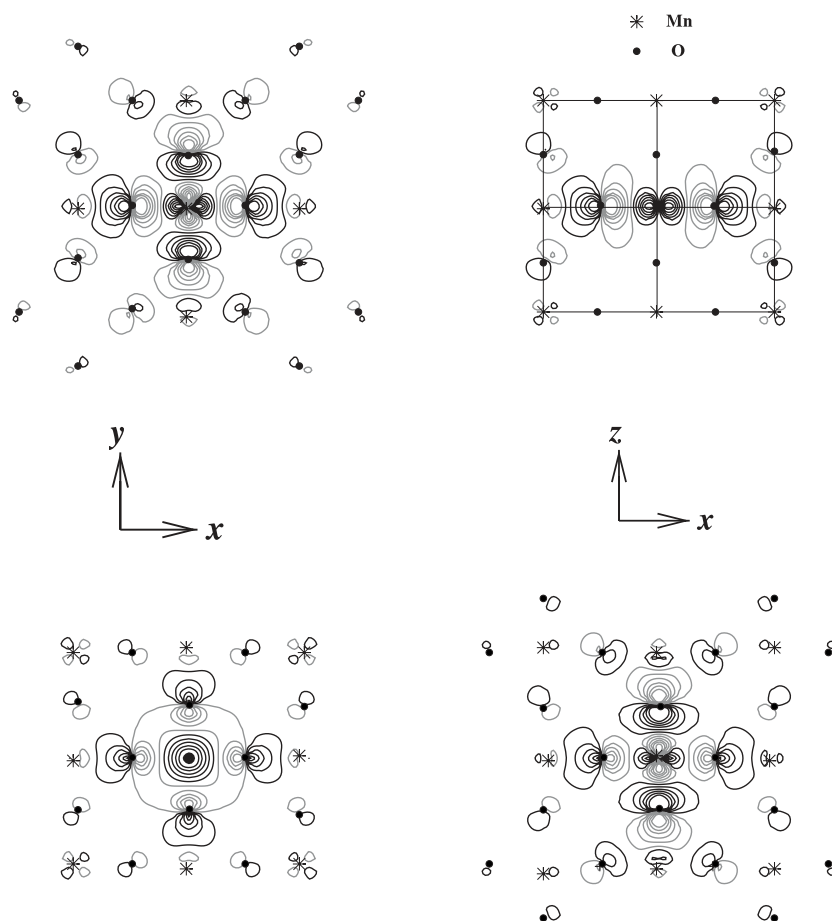


Figure 4. Downfolded Mn $x^2 - y^2$ (upper panel) and $3z^2 - 1$ (lower panel) orbitals plotted in xy - and xz -planes. The darker and lighter curves represent positive and negative lobes of the orbitals respectively. The underlying Mn–O lattice is also shown in the upper right panel.

note the efficiency of the downfolding method in picking out the correct Mg e_g -like bands from the overlapping Mn1 e_g and Mn2 t_{2g} band complex. The effective Mn e_g MTOs which form the Wannier-like functions corresponding to downfolded Mn e_g bands are shown in figure 4. The e_g orbitals with $x^2 - y^2$ and $3z^2 - 1$ symmetry at the Mn1 site have been plotted in the xy - and xz -planes. The tails of these effective orbitals are shaped according to the renormalization effect due to the contribution of the orbitals being downfolded. We note the significant presence of oxygen character in the tails of the effective orbitals further away from NN oxygen sites, giving rise to the long-range nature of these orbitals. We also note the presence of t_{2g} character at the Mn sites due to the overlap of the t_{2g} - and e_g -derived states at the two Mn sublattices.

The effective TB Hamiltonian is of dimension 4×4 , defined on the basis of two effective e_g orbitals for each of the two Mn sites belonging to two magnetic sublattices. Fourier transformation of the downfolded Hamiltonian ($H(k) \rightarrow H(R)$) gives a TB electronic Hamiltonian in real space consisting of hopping over up to five NNs. For many studies, it is desirable to have a shorter-range Hamiltonian. In order to arrive at a short-ranged Hamiltonian, restricted to second NN, and still keeping the advantage of the downfolding method in properly

Table 1. The Slater–Koster parameters for the best fit corresponding to the 2NN TB Hamiltonian for Mn e_g bands of AF CaMnO_3 . For the NN interaction in a simple cubic lattice, there is no $dd\pi$ interaction with e_g orbitals and $V_{dd\pi}$ does not appear [16].

On-site:	Distance = $0.00a$	$\epsilon_{e_g}^{\text{Mn1}} = 2.30 \text{ eV}$	$\epsilon_{e_g}^{\text{Mn2}} = 4.40 \text{ eV}$	
1NN:	Distance = $1.00a$			
	Mn1–Mn2	$dd\sigma = -0.90 \text{ eV}$		$dd\delta = 0.07$
2NN:	Distance = $1.41a$			
	Mn1–Mn1	$dd\sigma = -0.25 \text{ eV}$	$dd\pi = 0.04 \text{ eV}$	$dd\delta = 0.00 \text{ eV}$
	Mn2–Mn2	$dd\sigma = -0.20 \text{ eV}$	$dd\pi = 0.10 \text{ eV}$	$dd\delta = -0.01 \text{ eV}$

Table 2. The Slater–Koster parameters for the best fit corresponding to 2NN TB Hamiltonian for Mn e_g bands of FM CaMnO_3 .

On-site:	Distance = $0.00a$	$\epsilon_{e_g}^{\text{Mn}} = 2.00 \text{ eV}$		
1NN:	Distance = $1.00a$			
		$dd\sigma = -0.75 \text{ eV}$		$dd\delta = 0.08$
2NN:	Distance = $1.41a$			
		$dd\sigma = -0.27 \text{ eV}$	$dd\pi = 0.04 \text{ eV}$	$dd\delta = 0.03 \text{ eV}$

taking into account the renormalization effect, we have used a combined method where we have used the optimization procedure to reproduce the downfolded bands starting from the hopping matrix elements provided by the Fourier transform of the downfolded Hamiltonian as the input. The second-NN Hamiltonian that best fits the downfolded bands expressed in terms of the Slater–Koster integrals $dd\sigma$, $dd\pi$, $dd\delta$ (by turning the z -axis along the line connecting two interacting sites) are shown in table 1. The corresponding TB bands are shown by dashed curves in the right-hand panel of figure 3.

The on-site energy difference between the e_g states at two Mn sites, Mn1 and Mn2, belonging to two magnetic sublattices, which is caused by the exchange splitting between the spin channels at a given Mn site (by AFM construction), turns out to be finite, and moderate in value ($\approx 2.1 \text{ eV}$), in contrast with the assumption of infinite Hund’s-rule coupling. This estimate of the on-site energy difference between two effective Mg e_g levels at two magnetic sublattices may be considered as a lower bound for the Hund’s-rule coupling in this material. In the absence of the strong renormalization effect from oxygen p levels, this could have been considered as an estimate of the Hund’s-rule coupling. However, the hybridization from the oxygen p levels pushes the Mn e_g levels in the effective Mn e_g model, giving rise to a lower value of this energy difference compared to that obtained from the constrained LDA kind of approach [14].

At the end, we also carried out similar analysis with CaMnO_3 in the hypothetical FM phase, since this is of relevance for the study of electron-doped systems, where local FM order may be induced by the doped electrons (magnetic polarons). The FM phase has higher energy compared to the ground-state AF phase. Figure 5 shows the LDA bands (left-hand panel) and the downfolded e_g -only bands (as solid curves, right-hand panel) for the majority spin channel. The k -points are: $\Gamma = (0, 0, 0)$, $X = (2\pi/a)(0, 1/2, 0)$, and $M = (2\pi/a)(1/2, 1/2, 0)$. The right-hand panel of the figure also shows the TB bands with nearest- and next-nearest-neighbour hoppings, as dashed curves. The TB parameters obtained are given in table 2.

The NN hopping obtained by Chen and Allen [8] is consistent with ours, although, as seen from table 2, we also find a significant next-nearest-neighbour hopping.

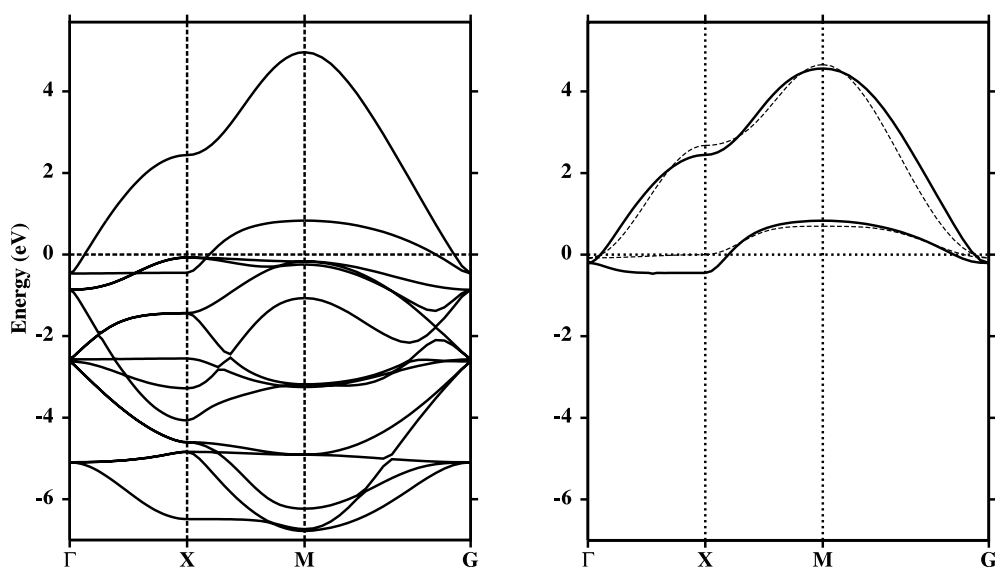


Figure 5. The full LDA band structure (left panel) and the downfolded Mn e_g -only bands (right panel) in the majority spin channel for FM CaMnO_3 . The right-hand panel also shows, as dashed curves, the best-fit TB bands with first- and second-NN hoppings with the parameters given in table 2.

5. Conclusions

In summary, by using the downfolding technique within the new formulation of the NMTO method, we have analysed the nature of the Mn e_g -derived bands in CaMnO_3 in terms of Wannier-like functions and obtained the parameters for the TB Hamiltonian. The effective Mn e_g orbitals for CaMnO_3 obtained by integrating out the other degrees of freedom show significant long-ranged oxygen characters, and Mn t_{2g} characters at other sublattices, caused by the moderate value of the Hund's-rule coupling parameter J_H allowing mixing between the two sublattices. This in turn gives rise to the appreciable width of the Mn e_g bands in CaMnO_3 . We have also carried out a downfolding calculation to derive the Mn e_g bands in the hypothetical FM phase of CaMnO_3 and obtained the corresponding parameters for the TB Hamiltonian.

Acknowledgments

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References

- [1] Jonker G H and van Santen J H 1950 *Physica* **16** 377
- [2] Rao C N R, Cheetham A K and Mahesh R 1996 *Chem. Mater.* **8** 2421
- [3] Zener C 1951 *Phys. Rev.* **82** 403
- [4] Anderson P W and Hasegawa H 1955 *Phys. Rev.* **100** 675
- [5] Andersen O K and Saha-Dasgupta T 2000 *Phys. Rev. B* **62** R16219 and references therein
- [6] van Roosmalen A M and Cordfunke E H P 1994 *J. Solid State Chem.* **110** 109
- [7] Neumeier J J and Cohn J L 2000 *Phys. Rev. B* **61** 14319

-
- [8] Chen Y and Allen P B 2001 *Phys. Rev. B* **64** 064401
 - [9] Meskine H *et al.*, unpublished
 - [10] Elemans J B A A, van Laar B, van der Veen K R and Loopstra B O 1971 *J. Solid State Chem.* **3** 238
 - [11] Andersen O K 1975 *Phys. Rev. B* **12** 3060
 - [12] Andersen O K and Jepsen O 1984 *Phys. Rev. Lett.* **53** 2571
 - [13] Pickett W and Singh D 1996 *Phys. Rev. B* **53** 1146
 - [14] Satpathy S, Popović Z and Vukajlović F 1996 *Phys. Rev. Lett.* **76** 960
 - [15] Müller T, Anisimov V, Rice T M, Dasgupta I and Saha-Dasgupta T 1998 *Phys. Rev. B* **57** R12655
 - [16] Valentí R, Saha-Dasgupta T, Alvarez J V, Pozgajcic K and Gros C 2001 *Phys. Rev. Lett.* **86** 5381
 - [17] Pavarini E, Dasgupta I, Saha-Dasgupta T, Jepsen O and Andersen O K 2001 *Phys. Rev. Lett.* **87** 047003
 - [18] Sarma D D, Mahadevan P, Saha-Dasgupta T, Ray S and Kumar A 2000 *Phys. Rev. Lett.* **85** 2549
 - [19] Marzari N and Vanderbilt D 1997 *Phys. Rev. B* **56** 12847
 - [20] Walter A Harrison 1980 *Electronic Structure and the Properties of Solids* (San Francisco, CA: Freeman) p 481